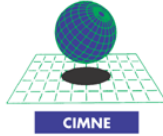


**Master Erasmus Mundus in Computational Mechanics  
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**Ecole Centrale de Nantes  
Institut de Recherche en Génie Civil et Mécanique**

**MASTER THESIS**

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**Prediction of Delamination in Composite  
Laminates under Quasi-Static and Fatigue  
Loadings with Local / Non-local  
Damage Evolution**

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## Abstract

Composites are materials that are composed of two or more distinct phases. Importantly, these phases shall not dissolve or blend with each other and remain distinct on a macroscopic level. However, mechanical and physical properties of each phase would differ from each other. Here, the challenge is to combine properties of each phase in a systematic manner to form the most efficient material for the intended application. Among many classes of advanced composite materials, fibrous composite laminates have been the most preferred option for structural applications. The inherent anisotropy and the brittle characteristics of composite laminates result in failure mechanisms that are very different from those of homogeneous monolithic materials. Among all different failure mechanisms, 'Delamination' is considered to be the most prominent mode of failure in fiber-reinforced laminates as a result of their relatively weak inter-laminar strength. When laminated composites are subjected to static, dynamic or cyclic loadings, the inter-laminar adhesion strength between individual plies tends to deteriorate significantly and act as the origin of the final failure. Therefore, an efficient and reliable design tool capable of predicting delamination would certainly improve the designs based on composite laminates. The present study is focused on taking a step forward in this respect.

The main objectives of this thesis work is to study an existing design tool [Allix O. and Ladevèze P., 1994; Gornet L., 1996; Ijaz H., 2009] and then to enhance it further to predict initiation and propagation of delamination under both quasi-static (i.e., monotonic) and fatigue (i.e., cyclic) loading conditions. The existing tool has been originally formulated within the framework of Damage Mechanics. Therefore, in the present work much attention has been devoted to improve its functionality and versatility by studying Damage Mechanics formulations. However, Fracture Mechanics also play a key role in determining the damage model. The key idea here is to use Fracture Mechanics test results to determine parameters of the Damage Mechanics model. Therefore, these damage models basically link Fracture Mechanics to Damage Mechanics.

The work presented in this report is organized in to four main parts. PART-I, also called 'Preamble' comprises two main chapters, Chapter 1 and Chapter 2. In Chapter 1, at first, a general introduction is given on Composites, focusing especially on their advantages and applications. Next, information on general failure mechanisms associated to composites laminates is also presented. In addition, a brief discussion is made on the delamination phenomena in laminated composites. Next, a synopsis of state-of-the-art modelling tools (based on Fracture and/or Damage Mechanics) for predicting delamination is also presented. In Chapter 2, a review has been made on the existing interface damage model which is based on meso-modelling concept. In addition, a detailed description of the constitutive equation and the methodology adopted for FE implementation is also included.

PART-II is dedicated for the study performed on 'Delamination under Quasi-static Loading'. It is also comprised of two major chapters, Chapter 3 and Chapter 4. Chapter 3 mainly contains formulations of the static damage model. Note that, the existing model is inherently local, meaning that it is mesh dependent. Therefore, to overcome such spurious localizations, existing theories on regularization methodologies were studied and reported. Chapter 4 includes detailed descriptions on FE simulations performed using static damage models (i.e., local and proposed nonlocal). At first, the local model's effectiveness was tested in all dimensional spaces with appropriate modifications for static loading condition. Identification of model parameters and preliminary investigation results followed by main investigation results are detailed out comprehensively. Next, a nonlocal integral-type regularization scheme was introduced to overcome the spurious localization problem associated to the existing local model. Special attention was devoted to improve the FE formulation of the nonlocal model to reduce computational cost. Details of complete formulation, simulations procedures and all investigations results with accompanying conclusions are included.

PART-III of the report is dedicated for the study on 'Delamination under Fatigue Loading'. Once more, it is comprised of two chapters, Chapter 5 and Chapter 6. Chapter 5 starts with an overview of fatigue and related theories associated to composite laminates. Then after, it includes an introduction of a new fatigue damage evolution law, its derivation and implementation. The existing local model was used as a platform to build the new fatigue model. Chapter 6 includes details of FE simulations performed to validate the proposed fatigue damage model. Versatility of the fatigue model was also checked for different mode-ratios. Procedure of identification of model parameters, simulation results and accompanying conclusions are also detailed out completely. Finally, a summary of conclusions of the present work is included in Chapter 7 under PART-IV, titled as 'Closure'.

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## Nomenclature

$U_1, U_2, U_3$	Interfacial displacements in orthotropic directions [mm]
$\sigma_{13}, \sigma_{23}, \sigma_{33}$	Shear and normal interfacial stresses [MPa] or [N/mm <sup>2</sup> ]
$k_1^0, k_2^0, k_3^0$	Shear and normal interfacial stiffnesses [MPa/mm]
$d_1, d_2, d_3$	Damage variables for Mode-II, Mode-III & Mode-I failures
$E_D$	Elastic strain energy of inter-laminar interface [N/mm]
$Y_{d_1}, Y_{d_2}, Y_{d_3}$	Damage energy release rates for Mode-II, Mode-III & Mode-I failures [N/mm]
$\bar{Y}$	Equivalent damage energy release rate [N/mm]
$\gamma_1, \gamma_2$	Coupling parameters in Mixed-Mode case for Mode-II and Mode-III failures
$\alpha$	Material parameter for damage evolution in Mixed-Mode case
$n$	Power of static damage evolution function
$w(\bar{Y})$	Static damage function
$Y_0$	Threshold damage energy release rate [N/mm]
$Y_C$	Critical damage energy release rate [N/mm]
$G_{IC}, G_{IIC}, G_{IIIC}$	Critical energy release rates for Mode-I, Mode-II & Mode-III failures [N/mm]
$G_I, G_{II}, G_{III}$	Energy release rates for Mode-I, Mode-II & Mode-III failure modes [N/mm]
$\bar{U}$	Averaged interfacial displacement [mm]
$\bar{Y}$	Averaged equivalent damage energy release rate [N/mm]
$\bar{d}$	Averaged damage variable
$\alpha_0, l$	Weighing function and internal length scale
$2L, b, 2h$	Span, width and height of specimen [mm]
$a$	Crack length [mm]
$da$	Crack extension [mm]
$P_I, P_{II}$	Force at load point for Mode-I & Mode-II failures
$U_I, U_{II}$	Displacement at load point for Mode-I & Mode-II failures
$\phi$	Mode-ratio (w.r.t, Mode-II)
$\lambda, C, \beta$	Fatigue damage model parameters
$G_F(d, N)$	Fatigue damage function for single cycle
$N, \Delta N$	Total number of fatigue cycles and number of Jump cycles
$G_{\max}$	Strain energy release rate at max. of cyclic load envelope ( $U_{\max}$ ) [N/mm]
$B, m$	Coefficient and exponent of Paris law for composites

# **PART – I**

## **Preamble**

Section comprises two main chapters, Chapter 1 and Chapter 2.

In Chapter 1, at first, a general introduction is given on Composites, focusing especially on their advantages and applications. Next, information on general failure mechanisms of composites laminates is also presented. In particular, a discussion is made on the delamination phenomena in laminated composites. Next, a synopsis of state-of-the-art modelling tools (based on Fracture and/or Damage Mechanics) for predicting delamination is also presented.

In Chapter 2, a review has been made on the existing interface damage model which is based on meso-modelling concept. In addition, a detailed description of the constitutive equation and the methodology adopted for FE implementation is also included.

# Chapter 1

## General Introduction

### 1.1 Composites and their Applications

A Composite is a material which is composed of two or more distinct phases. Unlike monolithic materials, one can easily distinguish these phases as they do not dissolve or blend with each other and remain distinct on a macroscopic level. They are merely connected with each other by means of interfacial bonding. Therefore, Composites are said to be 'Heterogeneous' by nature. Note that, the mechanical, physical and chemical properties of each phase would differ from one to the other. Here, the challenge is to combine properties of each phase in a systematic manner to form the most efficient material for the intended application. It is this particular characteristic which makes Composites as one of the most lucrative materials for engineering applications in the 21<sup>st</sup> century.

Composites are materials in which one phase acts as 'reinforcement' for the second phase, where the second phase is known as the 'matrix' [Herakovich C. T., 1997]. In other words, the matrix surrounds and binds together a cluster of reinforcing fibers or particles having a preferred orientation. Note that, reinforcements are much stronger / stiffer, and are responsible for composite's high structural properties. They mainly act as the primary load carrying component. On the other hand, matrix is responsible in transferring stresses between reinforcing fibers. Although fibers are strong, they can be brittle. The matrix is capable of absorbing energy by deforming under stress. In other words, the matrix adds toughness to the composite. In addition, matrix gives compression strength and thereby ensuring structural integrity upon compression.

Composites appeared in nature for a long time. A piece of Wood can be considered as a composite since it is composed of fibers of cellulose (reinforcement) held together by a weak binding substance called lignin (matrix). Human or animal bones are another good example for naturally occurring composites. Bones are mainly composed of fiber like osteons embedded in an interstitial bone matrix. A bird's wing, fins of a fish, etc. are few other examples of natural composites. The first man-made composite; straw-reinforced clay, were used to produce bricks, pottery, etc. and is still in use today.

At present, engineered composites are made up of metal, ceramic or polymer binders reinforced with different fibers (glass, carbon, polymer, etc) or particles (metal, ceramic, etc). By combining different matrix materials with different reinforcing materials it is possible to obtain a wide range of composites having different mechanical properties. More importantly, one can change the design parameters of the material during its manufacture to tailor its properties for the intended purpose. For example, during fabrication, the reinforcing fibers can be placed in the most preferred orientation to obtain desired properties in specified direction without overdesigning in other directions. In addition, the ratio between the fiber volume-fraction to matrix volume-fraction can be altered to change strength of the material according to design specifications of the structure. One can also instill desired physical properties (resistance to heat, thermal expansion, corrosion, etc.) into the composite by selecting an appropriate matrix material. Composite also exhibit high strength to weight ratios and high stiffness weight ratios, which make them ideal for structural applications. They can merely be fabricated in to the net-shape of the final desired product. As a result, composite materials find many applications compared to any other conventional materials.

Among many classes of advanced composite materials, fiber-reinforced polymer-matrix composites (Fibrous Composites) have been the most preferred option for structural applications. These fibrous composites can be found in many different forms, such as laminates, woven fabrics, etc. Laminated composites can be broadly classified under unidirectional laminates and multidirectional laminates. Unidirectional laminates are composed of layers of material having fibers in the same direction, where as multidirectional laminates are made of stacking unidirectional layers at different fiber orientations. The effective mechanical properties (i.e., strength, stiffness, etc) of the laminate vary with the orientation, thickness and stacking sequence of the individual layers.

The use of composite materials in our day-to-day life has increased dramatically over the last three decades. Initially, composites were exclusively used in technologically advanced applications, such as in aerospace and aeronautics industry, automotive industry, etc. Here, the main reason being high processing costs of composites compared to their counterparts. However, technological development and subsequent reduction of manufacturing costs, has allowed composites to be used in wide range of applications. These applications range from advanced aerospace and aircraft vehicles, bridges, boats, automotive parts, sportive items, household items to biomedical implants. The Figure 1.01 given below (Source - Toray Industries, Tokyo) illustrates statistics of the use of carbon-fiber reinforced composites for the period from 1985 to 2005. Here one can note the significant increase in use of composites in the industrial sector.

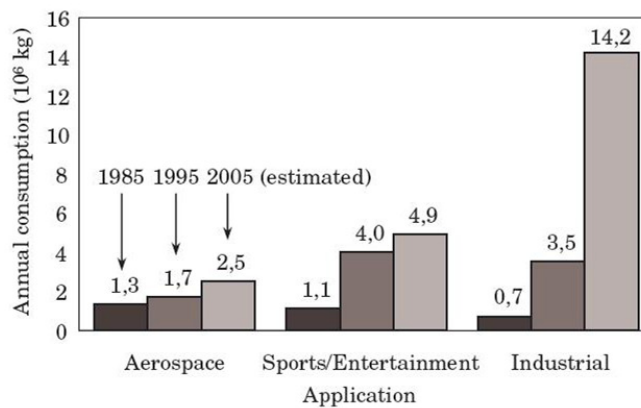


Figure 1.01 : Statistics on use of carbon-fiber composites in the industrial sector

Fiber composite structures are finding increasing use in the upcoming generation of aircrafts. The primary reason that has driven the increased use of composites in aircraft, particularly in airliners, is their reduced weight compared to equivalent metal structures. In addition, composites also exhibit greater specific strength and stiffness, aerodynamic smoothness, and resistance to fatigue and corrosion that is required for such applications. In terms of airliners, the Boeing 787 Dreamliner, Airbus A380 and A350XWB programs contain a large percentage (by weight) of composites, including many key structural components. Figure 1.02 given below (Source - Teal Group, Boeing, Airbus, Composite market reports), illustrates the use of composites (percentage by weight) on major aircraft programs for the period from 1975 to 2010.

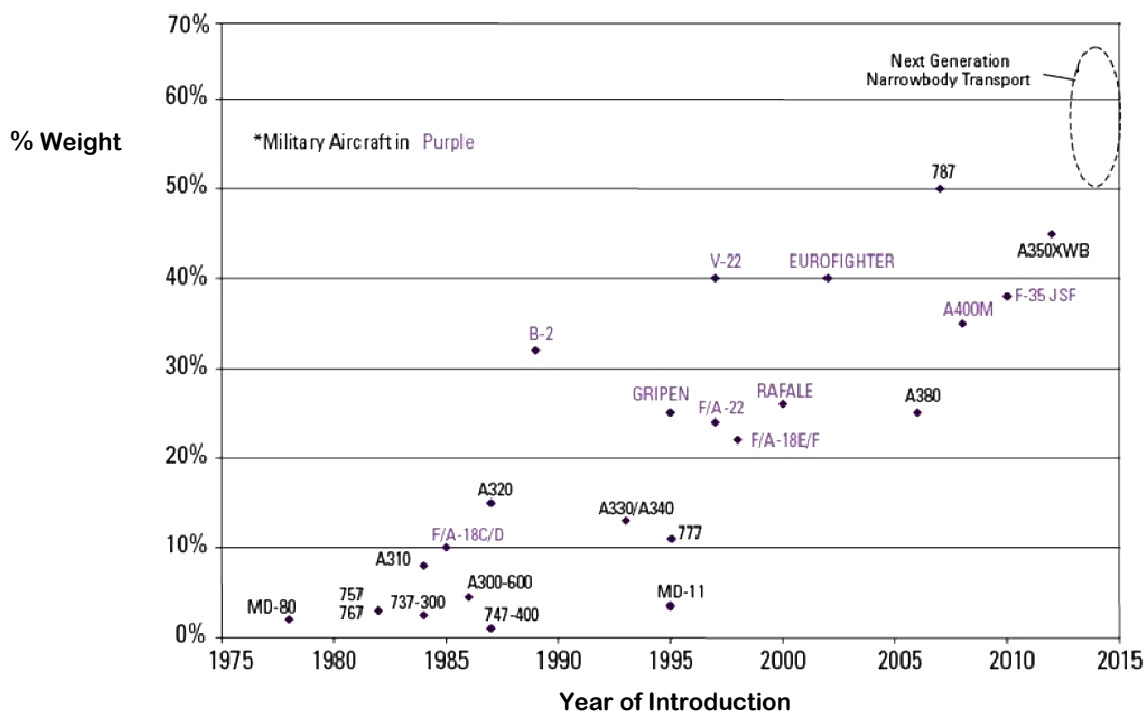


Figure 1.02 : Statistics on usage of composites by weight on major aircraft programs

## 1.2 Failure Modes of Composite Laminates

The inherent anisotropy and the brittle characteristics in fiber direction of composite laminates result in macroscopic failure mechanisms that are very different from those of homogeneous monolithic materials. Therefore, it is important to understand the factors influencing failure of fibrous composites (i.e., damage development) under various environmental and mechanical loading conditions. By gaining a proper understanding of the failure phenomena of the composite, one can then use it to further optimize the final design. The present study also focuses in achieving those objectives.

Failure of laminated composites is a complicated phenomenon and requires the understanding of its behavior at all scales. These heterogeneous, laminated materials typically exhibit many local failures prior to rupture into two or more distinct pieces. The local (initiation) failures are referred to as, 'damage', and the development of additional local failures with increasing load or time is called 'damage accumulation' [Herakovich C. T., 1997]. The accumulation of damage has a direct impact on material's stiffness. As the damage grows the stiffness of the structure also degrades gradually. The 'final failure' or 'final fracture' of the composite takes place when the structure is no more capable of withstanding the service load due to the reduction of its stiffness. Fiber composite materials fail in a variety of mechanisms at both micro and macro levels. Table 1.01 given below includes a comprehensive list of those failure mechanisms.

Mechanism	Description
Fiber failure	Fiber fracture, fiber pullout, fiber splitting, fiber buckling (kinking)
Matrix failure	Matrix cracking, degradation caused by radiation or moisture
Fiber/Matrix interface failure	Debonding at the fiber-matrix interface, radial interface cracking
Inter-laminar interface failure (Delamination)	Progressive debonding or separation of two adjacent laminae (plies/layers)

Table 1.01 : Failure mechanisms in fiber-reinforced composites

## 1.3 Delamination of Composite Laminates

Among all different failure mechanisms, 'Delamination' is considered to be the most prominent mode of failure in fiber-reinforced laminates due to their relatively weak inter-laminar properties (i.e., strength, energy release rate, etc). When laminated composites are subjected to static, dynamic or cyclic loadings, the inter-laminar adhesion strength between individual plies tends to deteriorate significantly. Ultimately, the laminate reaches a point where it can no longer sustain the loading, causing separation of the plies. However, recent researches have shown that delamination is a much more complex phenomena involving degradation of both the layers (brittle fracture of fibers, progressive transverse matrix cracking, debonding of fiber-matrix interface) and inter-laminar adhesion. At the microscopic level, the growth of an inter-laminar crack is preceded by the formation of a damage zone ahead of the crack tip. The size and shape of the damage zone will depend on matrix's toughness and more importantly, upon state of the stress (i.e., Mode-I, Mode-II, Mode-III or Mixed-Mode). The Figure 1.03 given below illustrates each of these crack propagation modes.

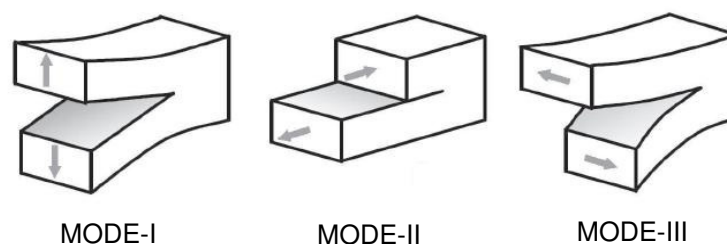


Figure 1.03 : Crack propagation modes

The size of the damage zone ahead of crack tip for Mode-II and Mode-III (shear) loadings is greater than for Mode-I (opening) loading. This is because; the stress field tends to decay much slowly for shear loadings. On the other hand, a brittle matrix material has a much smaller damage zone compared to a ductile matrix material.

Delamination may take place in the interior (inner delamination) as well on the exterior (near-surface delamination) of the composite laminates. Inner delamination, significantly reduce the load carrying capacity of the material. However, unlike in near-surface delamination, the delaminated laminae are within the composite. Hence, upon flexure all laminae deflect in a similar manner. On the other hand, when dealing with near-surface delamination one has to take into account not only the debonding of lamina but also its local stability. Delamination may arise in various circumstances. For example, they may originate from abrupt changes in the laminate (such as; ply drop-offs, free edges/ flanges, stiffener terminations, holes or bonded/ bolted joints). Curved sections (such as; tubular or spherical segments) may also promote delamination. In addition, temperature effects, moisture effects, impact events and fabrication defects (such as, incomplete wetting, air entrapment) would also act as possible sources.

The study of delamination is of great importance since it significantly affects the global stability of the structure. Delamination can cause local buckling and drastic reduction of the bending stiffness. It could also facilitate a direct way for the moisture or air to seep into the laminate. In addition, it may cause excessive vibrations, reduction of fatigue life, etc. As this failure mechanism forms inside the laminates one would find it extremely difficult to detect them during service conditions to take necessary actions. Therefore, an efficient and reliable design tool capable of predicting initiation of delamination would certainly improve the design based on strength criteria. Despite of many research studies and publications on this topic, delamination failure mechanisms in composites are still not well understood. It is this fact that has limited the use of fibrous laminated composites to its full potential.

## 1.4 Delamination Modelling with Fracture / Damage Mechanics

Delamination prediction is still considered to be a formidable challenge both from scientific and industrial point of view. This is because the analysts have to take into account a large number of parameters that are involved in design of composite laminates. In addition, the state of stress which is responsible for initiation and growth of delamination tends to be much more complicated as in the case of conventional monolithic materials. A notable effort has been devoted to the numerical and theoretical modelling of delamination in the last few decades but a number of issues still need to be further investigated [Allix O. and Ladevèze P., 1992/4; Corigliano A., 1993; Gornet L., 1996; Alfano G. and Crisfield M.A., 2001; Ijaz H., 2009].

The formation of a delamination in a flawless structure can be divided into two parts; delamination onset (initiation) and delamination growth (propagation). Traditionally, for initiation of delamination, the tolerance prediction was based on 'semi-empirical' criteria, such as point stress or average stress. Here, the inter-laminar stress state is computed and a strength criterion is utilized to predict onset of delamination. This criterion essentially requires a large number of experimental tests to certify the tolerance. Recently a novel method known as First Ply Failure criterion has also been introduced considering delamination as a special mode of ply failure.

For the simulation of delamination propagation, methods employing 'Fracture Mechanics', have been extensively used by many researchers throughout last several decades. Linear Elastic Fracture Mechanics (LEFM) has proven to be a suitable choice for predicting delamination growth when material non-linearity can be neglected. This is true in case of composite laminates since laminae are very stiff in the laminate plane and behave as linear elastic materials in their gross deformation. Therefore, LEFM serves as a very useful tool for the analysis of inter-laminar toughness. The core idea here is to express inter-laminar toughness in terms of 'Energy Release Rate'. Here, delamination is set to propagate when the energy release rate reaches its critical value. In addition, for the prediction of delamination growth, methods like, 'Virtual Crack Closure Technique' (VCCT), 'J-integral', and 'Virtual Crack Extension Technique' (VCET) have also been used successfully. The VCCT technique is based on the assumption that when a crack extends by a small amount, the energy released in the process is equal to the work required to close the crack to its original length. This approach is said to be computationally effective since the energy release rates can be obtained

from only one analysis. However, all these techniques can only be applied when a starting crack exists. In other words, they require the knowledge of initial delamination pattern. To make matters worse for certain geometries and load cases, the location of the delamination front might well be difficult to determine. In addition, even for 2D applications, difficulties may arise when more than one crack propagate simultaneously.

Another approach for numerical simulation of delamination is the 'Cohesive Zone Method' (CZM). Main advantage of CZM is the capability to predict both onset and growth of delamination without previous knowledge of the crack location and propagation direction. The technique is based on the framework of 'Damage Mechanics' and 'Strain Softening'. Here, a thin layer of matrix material is assumed to exist between plies. Delamination is interpreted as creation of a cohesive damage zone in front of the delamination-front, separating adjacent plies. Here, traction–separation laws for the interface can be defined within Finite Element (FE) Method. On the flip side, CZM is considered to be numerically expensive since it requires a fine mesh in order to represent damaged zone accurately. However, the growing power of computers and damage mechanics formulation, offer the possibility of avoiding many experimental tests and other prerequisites (i.e., initial delamination pattern). In recent times, XFEM methods along with levels sets has also been successfully used to predict initiation and propagation of cracks in metals/ concrete using cohesive zone approach and is a good candidate for prediction of delamination in composite laminates.

## 1.5 Objectives and Scope of Study

The most severe limitation to the application of composites is the lack of engineering knowledge to design with these materials. As the understanding of their behavior improves, the number and range of applications will grow rapidly. This requires a continuous development in the analysis and design procedures of composites. Present study is focused on taking a step forward in this respect.

The usual methodology for the determination of composites reliability involves large number of mechanical tests. This would ultimately lead to high design costs. Moreover, existing tests are not sufficient to reproduce all possible circumstances encountered during material's service life. These circumstances include environmental conditions (temperatures effects, moisture effects, etc), loading conditions (static, dynamic, fatigue, etc) or a combination of both. As a result, a reliable, robust and efficient design tool is required to predict the behaviour of composites, especially their failure mechanisms (i.e., delamination). When designing advanced composites one can resort to either to a strength verification approach or damage-tolerance verification approach. Here, 'damage-tolerance' verification uses a more realistic approach in appreciating the material's degradation of mechanical properties and reduction of its functionality. The methodology takes into account the formation of subcritical cracks that will continue to grow and cause the final rupture. Therefore, it may prove useful in analyzing composite structures loaded under fatigue load conditions, where the growth of subcritical cracks is the main cause of failure. In addition, this approach could provide a quantitative guidance to repair or replace a damage component of a structure before any catastrophic structural failure. Based on all these facts, a design tool based on damage-tolerance approach is considered to be the most logical option to study failure mechanisms. Therefore, the present study mainly focuses on predicting delamination failure using the Damage Mechanics frame work.

The main objectives of this thesis work is to study an existing design tool [Allix O. and Ladevèze P., 1994; Gornet L., 1996; Ijaz H., 2009] and then to enhance it further to predict initiation and propagation of delamination under both quasi-static (i.e., monotonic) and fatigue (i.e., cyclic) loading conditions. At first, an existing damage model was exhaustively studied. Then after, effort was put in to improving its functionality and versatility in simulating delamination initiation and growth for different loading conditions. In the first part of the work, model's effectiveness was tested in all dimensions with appropriate modifications for static loading condition. Next, a regularization scheme (i.e., nonlocal integral-type) was introduced to overcome the spurious localization problem associated to the existing local model. Special attention was devoted to improve the FE formulation of the nonlocal model to reduce computational cost. The second part of the work focuses on delamination due to fatigue loading condition. It includes an introduction to a new fatigue damage evolution law, its derivation and implementation. The existing local model was used as a platform to build the new fatigue model. Fracture Mechanics test results were used to validate and verify the effectiveness of both static (local/nonlocal) and fatigue models.

## Chapter 2

### Interface Damage Model

#### 2.1 Meso-Modelling Concept

The failure of composites has been investigated extensively from the micro-mechanical and macro-mechanical points of view. On the micro-mechanical scale, failure mechanisms and processes vary widely with type of loading and are intimately related to the properties of the constituent phases, i.e., matrix, reinforcement, and interface-interphase. Failure predictions based on micro-mechanics, even when they are accurate with regard to failure initiation at critical points, are only approximate with regard to global failure of a lamina and failure progression to ultimate failure of a multi-directional laminate [Issac M. Daniel, 2005]. Although, macro-mechanical approach could provide a better understanding of composite's behavior, it may not be sufficient enough to appreciate the progressive degradation effect or the influence of various parameters that contribute to ultimate failure of the material. As a result of these inherent limitations encountered at both micro and macro scales, researchers have turned their focus on predicting failure characteristics at a more intermediate scale known as the 'meso-scale' [Allix O. and Ladevèze P., 1992/4; Corigliano A., 1993; Gornet L., 1996].

Figure 2.01 given below illustrates a classical meso-model of a laminated composite. Typically a laminate is composed of plies having different fiber orientations. Plies were originally manufactured as unidirectional prepregs and were later fabricated together by a bonding process.

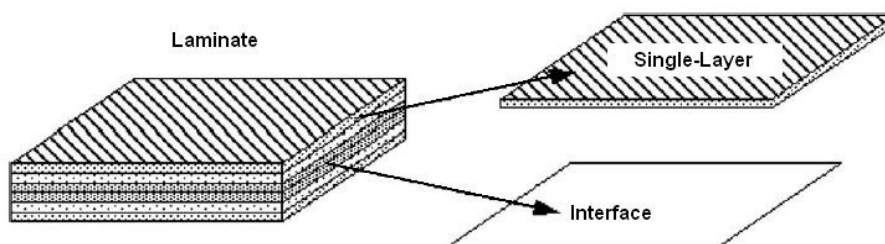


Figure 2.01 : Meso-model of a laminate

At the meso-scale, the laminate is described as a stacking sequence of inelastic and damageable homogeneous layers throughout the thickness and of damageable inter-laminar interfaces [Allix O. and Ladevèze P., 1992]. Here importantly, the inter-laminar interface is considered to be a mechanical constituent. The term 'interface' represents a physical yet two-dimensional medium that is responsible in transferring traction and displacements between two adjacent layers. The mechanical properties of the interface mainly depend on the relative orientation of their fibers. Since the thickness of a single layer is considerably small, the kinematics of the deterioration of the ply is assumed to be homogeneous. In other words, one can assume a uniform damage state throughout a given ply.

The Cohesive Zone Method (See Section - 1.4) can then be used to predict failure of the interface. The concept of 'decohesion zones' to simulate delamination in composites is usually implemented by means of 'interface (decohesion) elements' connecting the individual plies of a composite laminate. Physically, the cohesive zone represents the coalescence of crazes in the resin rich layer located at the delamination tip and reflects the way by which the material loses load-carrying capacity. Cohesive zone models are particularly attractive when interfacial strengths are relatively weak when compared with the adjoining material, as is the case in composite laminates. According to the model formulation, the process zone or cohesive zone is located ahead of the delamination tip. In this particular zone the damage variable ' $d$ ' varies from '0' to '1'. Note that,  $d=1$  in the zone where delamination has already taken place and  $d=0$  where there is no damage (See Figure 5.01). The damage variable will then be used to define the variation of rigidities of the interface elements and thereby the degradation effect of the interface.



## 2.2 Constitutive equation for the Interface

The need for an appropriate constitutive equation in the formulation of the interface modelling is fundamental for an accurate simulation of the inter-laminar debonding process. The constitutive equations for the interface are phenomenological mechanical relations between the tractions (stresses) and interfacial separations (displacement discontinuities). With increasing interfacial separation, the tractions across the interface reach a maximum (i.e., interfacial normal or shear tractions attain their respective inter-laminar tensile or shear strengths), decrease, and vanish when complete debonding occurs. The work of normal and tangential separation can be related to the critical values of energy release rates. It is this particular characteristic that allows one to link fracture mechanics with damage mechanics.

The displacement discontinuity or jump of one layer to the other layer can be written as,

$$[\vec{U}] = \vec{U}^+ - \vec{U}^- = [U_1]\vec{N}_1 + [U_2]\vec{N}_2 + [U_3]\vec{N}_3 \quad (2.01)$$

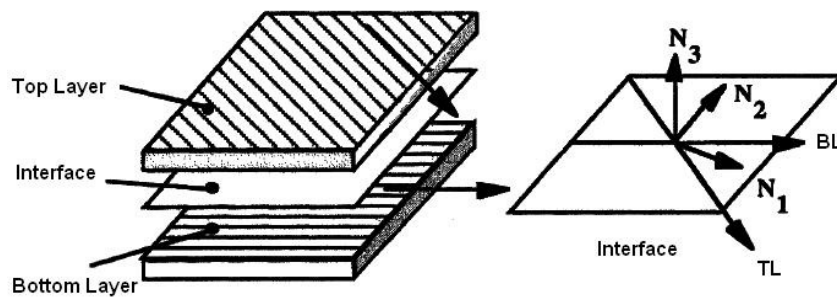


Figure 2.02 : Basic building blocks of the interface model

Let the bisectors of the fiber direction be  $\vec{N}_1$  and  $\vec{N}_2$ . The direction of  $\vec{N}_3$  is normal to the interface.

Essentially, all are 'orthotropic' directions of the interface. Here,  $\vec{U}^+$  and  $\vec{U}^-$  are displacement vectors of the top and bottom layers respectively. See Figure 2.02 given above.

Let  $k_1^0, k_2^0$  and  $k_3^0$  are initial interface rigidities (stiffness) associated to damage variables  $d_1, d_2$  and  $d_3$  along the orthotropic directions  $\vec{N}_1, \vec{N}_2$  and  $\vec{N}_3$  respectively. Here,  $d_3$  is associated to the opening mode (Mode-I) of the inter-laminar connections, where  $d_1$  and  $d_2$  are associated to in-plane and out-of-plane shearing modes (Mode-II and Mode-III) respectively.

The relation between the stresses and displacement jumps along the orthotropic axes can then be expressed as follows (for simplicity displacement jumps in each orthogonal direction is written without square brackets),

$$\begin{pmatrix} \sigma_{13} \\ \sigma_{23} \\ \sigma_{33} \end{pmatrix} = \begin{pmatrix} k_1^0(1-d_1) & 0 & 0 \\ 0 & k_2^0(1-d_2) & 0 \\ 0 & 0 & k_3^0(1-d_3) \end{pmatrix} \begin{pmatrix} U_1 \\ U_2 \\ U_3 \end{pmatrix} \quad (2.02)$$

### 2.3 FE Implementation of Damage Models

The constitutive behaviour laws presented in the present study are implemented in a finite element code called CAST3M via a user-subroutine known as UMAT.

CAST3M is a system designed and developed so as to overcome the hurdles of adaptability provided by conventional codes. It has been developed by the *Département Mécanique et Technologie (DMT) du Commissariat Français à l'Energie Atomique (ECA)*. CAST3M can be used as support for design, dimensioning and analysis of structures and components. It presents a complete system, integrating; tools for calculation, tools for model construction (pre-processor) and tools for processing results (post-processor). On the contrary of other systems, made to solve some defined problems, CAST3M is a program the user can adapt to his own needs. In practice, the program is made of a set of elementary operators (written in GIBIANE) and objects. Each operator is attributed to the execution of one unique operation. The user can manipulate the objects and operators to build a new application or customize an existing application. CAST3M enables the processing of linear and nonlinear problems in static, dynamic or cyclic fields. In addition, CAST3M is equipped with interface or joint elements that can be readily used for model construction (i.e., interface model of laminates). Therefore, all these facts make CAST3M an ideal research tool for the present study under consideration.

UMAT user-subroutine (written in FORTRAN) merely allows researchers to work on material response modelling. The subroutine can be used to introduce a new constitutive law (i.e., with damage effects) in to finite element code of CAST3M. In addition, the subroutine can also be used to define solution-dependent state variables. During a nonlinear calculation, the subroutine will be called at all material calculation points of elements for which the material definition includes a user-defined material behavior. It will also update the stresses and solution-dependent state variables to their values at the end of each increment for which it is called. The smaller the time increment the greater the accuracy of the result, but at the expense of high computational cost.

## **PART – II**

### **Delamination under Quasi-Static Loading**

Section comprises two chapters, Chapter 3 and Chapter 4.

Chapter 3 mainly contains formulations of the static damage model. In addition, to overcome spurious localizations of local models, existing theories on regularization methodologies were studied and reported.

Chapter 4 includes detailed descriptions on FE simulations performed using static damage models (i.e., local and proposed nonlocal). At first, the local model's effectiveness was tested in all dimensional spaces with appropriate modifications for static loading condition. Identification of model parameters, preliminary investigation results and main investigation results are detailed out comprehensively. Next, a nonlocal integral-type regularization scheme was introduced to overcome the spurious localization problem of the existing local model. Special attention was devoted to improve the FE formulation of the nonlocal model to reduce computational cost. Details of complete formulation, simulations procedures, investigations results and conclusions are also included.

## Chapter 3

### Static Damage Evolution

#### 3.1 Energy Formulation

In Chapter 2 under Section 2.2 a constitutive relation for the interface was introduced. Here, the effect of deterioration of the inter-laminar connection on its mechanical behaviour is taken into account by means of three internal damage variables which are associated to Mode-I, Mode-II and Mode-III failure modes. The total strain energy [Ladevèze P., 1986; Allix O. and Ladevèze P., 1992; Gornet L. et al 1997] in the system can now be expressed as,

$$E_D = \frac{1}{2} \left[ \frac{\langle \sigma_{33} \rangle_-^2}{k_3^0} + \frac{\langle \sigma_{33} \rangle_+^2}{k_3^0(1-d_3)} + \frac{\sigma_{23}^2}{k_2^0(1-d_2)} + \frac{\sigma_{13}^2}{k_1^0(1-d_1)} \right] \quad (3.01)$$

Note that, different types of damageable behaviour in ‘tension’ and in ‘compression’ are distinguished by splitting the strain energy into ‘tension energy’ and ‘compression energy’. For example,  $\langle X \rangle_+$  and  $\langle X \rangle_-$  represents the tension and compression parts of  $\langle X \rangle$ .

The thermodynamic model is built by taking into account the three possible modes of delaminations. The thermodynamic forces associated to each damage variable can be defined as,

$$Y_{d_1} = - \left[ \frac{\partial E_D}{\partial d_1} \right]_{|\sigma} \quad Y_{d_2} = - \left[ \frac{\partial E_D}{\partial d_2} \right]_{|\sigma} \quad Y_{d_3} = - \left[ \frac{\partial E_D}{\partial d_3} \right]_{|\sigma} \quad (3.02)$$

Now, using equations 3.01 and 3.02, the following relations can be derived,

$$\left. \begin{aligned} Y_{d_1} &= \frac{1}{2} \frac{\sigma_{13}^2}{k_1^0(1-d_1)^2} = \frac{1}{2} k_1^0 [U_1]^2 \\ Y_{d_2} &= \frac{1}{2} \frac{\sigma_{23}^2}{k_2^0(1-d_2)^2} = \frac{1}{2} k_2^0 [U_2]^2 \\ Y_{d_3} &= \frac{1}{2} \frac{\langle \sigma_{33} \rangle_+^2}{k_3^0(1-d_3)^2} = \frac{1}{2} k_3^0 \langle [U_3] \rangle_+^2 \end{aligned} \right\} \quad (3.03)$$

The total energy dissipated from the system can be expressed as,

$$\Phi = Y_{d_1} \dot{d}_1 + Y_{d_2} \dot{d}_2 + Y_{d_3} \dot{d}_3 \quad \text{Where, } (\Phi \geq 0) \quad (3.04)$$

One may also use classical Fracture Mechanics theory to determine the critical energy characteristics of the interface damage model. As explained in Section – 1.4, LEFM is a proven approach for dealing with propagation of delamination. In that respect, LEFM serves as a reference to compare and contrast any new modelling of delamination propagation. Since delamination is a dissipative phenomenon [Allix O. et al., 1995] a simple way to compare LEFM with the presented model is to compare their mechanical dissipations.

In LEFM, the total energy dissipated from a single crack whose area given by  $S$  is written as,

$$\Phi_{LEFM} = \int_a^{a+\Delta a} G \dot{S} \quad (3.05)$$

If one consider a specimen having a constant width equaling to  $b$ , then  $\dot{S} = b \dot{a}$ , where  $a$  is the length of the crack. Here,  $G$  is the ‘Energy Release Rate’ and it depends only on the material properties. In order to explicitly integrate equation 3.05, ‘steady-state’ propagation of the crack is assumed. In other words, crack length is assumed to increase by  $\Delta a$  over a time increment of  $\Delta t$ .

The Figure 3.01 given below illustrates the propagation of the crack tip along with the cohesive zone in a steady-state delamination process.

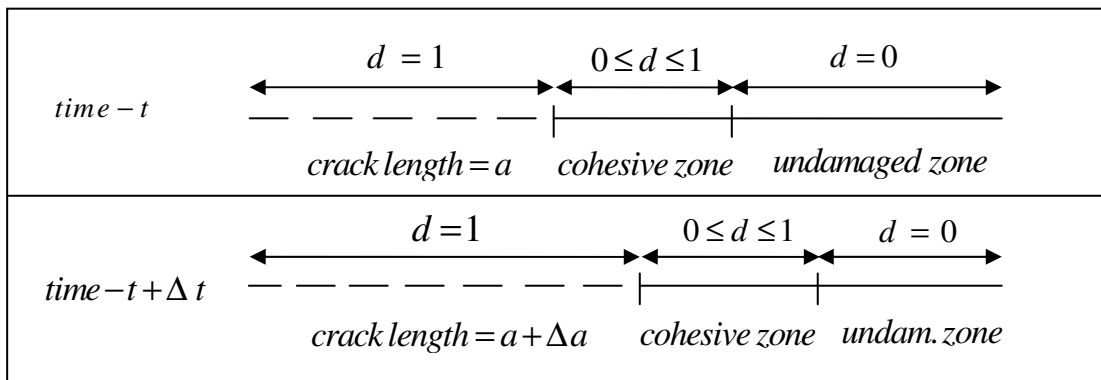


Figure 3.01 : Steady-state delamination process

Similarly, from equation 3.04, the total energy dissipated from the interface according to damage mechanics formulation can be written as,

$$\Phi_{IDM} = \int_{\Gamma} \left( \sum_{i=1}^3 Y_{d_i} \right) \dot{d}_i d\Gamma \quad (3.06)$$

Where,  $\Gamma$  represents the cohesive zone of the interface.

Since, the energy dissipated for propagation of delamination in LEFM should be equal to energy dissipated in interface damage model,  $\Phi_{LEFM} = \Phi_{IDM}$ .

Then integration of equations 3.05 and 3.06 for a given time increment, leads to the equations [Allix O. and Ladevèze P., 1996] given below,

$$\left. \begin{aligned} G_{IC} &= Y_C \\ G_{IIC} &= \frac{Y_C}{\gamma_1} \\ G_{IIIC} &= \frac{Y_C}{\gamma_2} \end{aligned} \right\} \quad (3.07)$$

$G_{IC}$ ,  $G_{IIC}$  and  $G_{IIIC}$  are critical energy release rates associated to Mode-I, Mode-II and Mode-III failure modes respectively. Here,  $Y_C$  is the critical ‘Damage Energy Release Rate’ and  $\gamma_1$  and  $\gamma_2$  are ‘Coupling Parameters’ for  $G_{IIC}$  and  $G_{IIIC}$  respectively.

In order to satisfy the energy balance principle, the area under the stress vs. displacement jump-curve for the 'Debonding Process' (DP) should be equal to critical energy release rate (i.e.,  $G_{IC}$ ,  $G_{IIC}$  &  $G_{IIIC}$ ) for each failure mode [Ladevèze P. et al., 1998; Corigliano A. and Allix O., 2000]. The resulting equations can be written as follows,

$$\left. \begin{aligned} G_{IC} &= \int_{DP} \langle \sigma_{33} \rangle^+ dU_3 \\ G_{IIC} &= \int_{DP} \sigma_{13} dU_1 \\ G_{IIIC} &= \int_{DP} \sigma_{23} dU_2 \end{aligned} \right\} \quad (3.08)$$

Notice that, delamination doesn't take place when the laminate is subjected to compressive stresses. See Figure 3.02 given below.

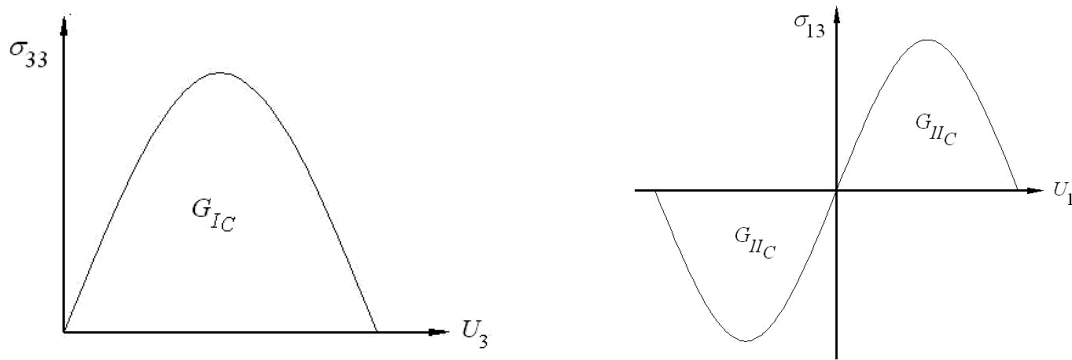


Figure 3.02 : Graphical representation of critical energy release rate for Mode-I and Mode-II

For the Mixed-Mode failure case, critical energy release rate for the system can be expressed as,

$$G_C = G_I + G_{II} + G_{III} \quad (3.09)$$

In addition, one can also write a relation for the Mixed-Mode failure as given below,

$$\left( \frac{G_I}{G_{IC}} \right)^\alpha + \left( \frac{G_{II}}{G_{IIC}} \right)^\alpha + \left( \frac{G_{III}}{G_{IIIC}} \right)^\alpha = 1 \quad (3.10)$$

Therefore, parameter  $\alpha$  governs the damage evolution in Mixed-Mode.

### 3.2 Local Damage Evolution

The damage evolution law explained hereunder is based on the assumption that the evolutions of the different damage variables are driven by a single variable known as 'Equivalent Damage Energy Release Rate' [Allix O. et al., 1995]. The expression for 'Equivalent Damage Energy Release Rate' is written as,

$$\bar{Y}(t) = \text{Max}_{\tau \leq t} \left( \left( Y_{d_3} \right)^\alpha + \left( \gamma_1 Y_{d_1} \right)^\alpha + \left( \gamma_2 Y_{d_2} \right)^\alpha \right)^{1/\alpha} \quad (3.11)$$

Note that, the first, second and third terms in the expression are associated to the Mode-I, Mode-II and Mode-III failure modes respectively.

Some important characteristics of this technique can be listed as follows,

- I. Evolution of the damage variables are assumed to strongly coupled (i.e., 'ISOTROPIC' damage evolution)
- II. Damage taken place within a given time period will depend on the maximum value of the equivalent damage energy release rate.

The damage or material function [Allix O. and Ladevèze P., 1996] is selected in the form,

$$w(\bar{Y}) = \left[ \frac{n \langle \bar{Y} - Y_0 \rangle_+}{n+1 Y_C - Y_0} \right]^n \quad (3.12)$$

Where,  $n$  is called the characteristic function of the material and  $Y_C$  and  $Y_0$  are critical and threshold damage energy release rates.  $Y_0$  is introduced to enlarge the possibility to describe both initiation and propagation of delamination (usually,  $Y_0 = 0$ ).

The damage evolution procedure can then be defined as follows,

*if*  $[d_3 < 1 \text{ and } \bar{Y} < Y_R]$

$$d_1 = d_2 = d_3 = w(\bar{Y})$$

*else*

$$d_1 = d_2 = d_3 = 1$$

*endif*

Where,

$$Y_R = Y_0 + \frac{n+1}{n} d_C (Y_C - Y_0) \quad (3.13)$$

Here,  $Y_R$  is the damage energy corresponding to rupture, and  $d_C$  is the critical damage value, which is usually taken as equal to one ( $d_C = 1$ ). In general,  $d_C \in [0, 1]$ .

### 3.3 Regularization Damage Evolution

The interface damage model presented in Section – 3.2 is called a local damage model as it does not account for ‘spurious localizations’. The classical damage models lead inevitably to strain softening response and the numerical solutions obtained with such models tend to exhibit severe mesh dependence in the presence of ‘strain-softening’. The problem is much more evident in three-dimensional (3D) problems rather than in two-dimensional (2D) problems [Gornet L. et al., 1997]. In addition, classical damage models tend to produce multiple solutions when the tangent operator becomes negative. This phenomenon is presented pictorially in Figure 3.03.

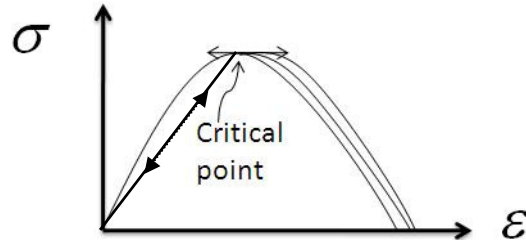


Figure 3.03 : Graphical representation of numerical problems of local damage models

Thus, to avoid spurious localization and other undesired effects one may need to regularize the damage evolution. Regularized evolution laws currently available fall into two major categories; namely, Rate-dependent damage evolution laws and Non-local (differential/ integral -type) damage evolution laws.

#### 3.3.1 Rate-Dependent Damage Evolution

The core idea here is to introduce a delay effect into the classical damage model. The variant damage model ensures that physical variation of driving force ( $\bar{Y}$ ) does not lead to an instantaneous variation of damage variable ( $d$ ). In other words, the ‘damage-rate’ is kept bounded. An example of a damage law with delay effect [Ladevèze P., 1995] is given below.

*if* [ $d_3 < 1$  and  $\bar{Y} < Y_R$ ]

$$\dot{d}_3 = k \left( \langle w(\bar{Y}) - d_3 \rangle^+ \right)^m, w(\bar{Y}) \leq 1$$

$$\dot{d}_1 = \dot{d}_2 = \dot{d}_3$$

*else*

$$d_1 = d_2 = d_3 = 1$$

*endif*

Note that, parameters  $m$  and  $k$  need to be determined experimentally. Here, an implicit Euler algorithm can be employed and the nonlinear discretized damage equation can be solved by Newton method. Damage models with delay effects have been successfully used to predict delamination process [Gornet L. et al., 1997, Marguet S. et al., 2006]. Simulated results were in well agreement with the experimental results. However, rate dependent models suffer from some limitations. For example, in case of fatigue loadings it is not logical to use rate dependent models as the load is inherently varying between the maximum and minimum values.



### 3.3.2 Non-Local Integral-Type Damage Evolution

The models are formulated by introducing a length scale into the governing equations to avoid the pathological mesh sensitivity. They have been successfully used to eliminate the effect of localization in civil engineering work by many researchers but domain is still open to implement them in interface damage modelling for delamination of composite laminates [Borino G. et al., 2007; Milan Jirásek, 2007].

In integral-type non-local damage models, the equivalent strain ( $\varepsilon_{eq}$ ) is made non-local by means of spatial averaging over the surrounding volume ( $V$ ) with respect to a 'weight function' ( $\alpha_0$ ) (i.e., a Gaussian distribution function) [Zdeněk P. Bažant et al., 1988]. Now, the resulting average equivalent strain can be defined as follows.

$$\bar{\varepsilon}_{eq}(x) = \frac{\int_V \alpha_0(\|x - \zeta\|) \varepsilon_{eq}(\zeta) d\zeta}{\int_V \alpha_0(\|x - \zeta\|) d\zeta} \quad (3.14)$$

$$\text{Where, } \alpha_0(r) = \exp\left(-\frac{r^2}{2l^2}\right) \quad (3.15)$$

Here,  $r = \|x - \zeta\|$  and,  $l$  is the internal length scale.

The same idea can be applied for interface damage modelling by the following ways.

Method-1 Averaging relative displacement :

$$\varepsilon_{eq}(\zeta) \rightarrow [U(\zeta)] \quad \Rightarrow \quad \bar{\varepsilon}_{eq}(x) \rightarrow [\bar{U}(x)] \quad \Rightarrow \quad d = d(\bar{Y}([\bar{U}(x)]))$$

Method-2 Averaging damage energy release rate :

$$\varepsilon_{eq}(\zeta) \rightarrow \bar{Y}(\zeta) \quad \Rightarrow \quad \bar{\varepsilon}_{eq}(x) \rightarrow \bar{\bar{Y}}(x) \quad \Rightarrow \quad d = d(\bar{\bar{Y}}(x))$$

Method-3 Averaging damage variable :

$$\varepsilon_{eq}(\zeta) \rightarrow d(\zeta) \quad \Rightarrow \quad \bar{\varepsilon}_{eq}(x) \rightarrow \bar{d}(x) \quad \Rightarrow \quad d = \bar{d}(x)$$

Here, Method-1 is considered to be the most effective approach compared to Method-2 and Method-3. This is mainly because,  $\bar{Y}$  is a function of  $[U]$  and  $d$  is a function of  $\bar{Y}$ .

The computed damage variable will be non-local in nature due to its dependence on averaged variables. Now, the relation between the stresses and displacement jumps along the orthotropic axes can be re-written in terms of the nonlocal damage variables as follows (for simplicity displacement jumps in each orthogonal direction is written without square brackets),

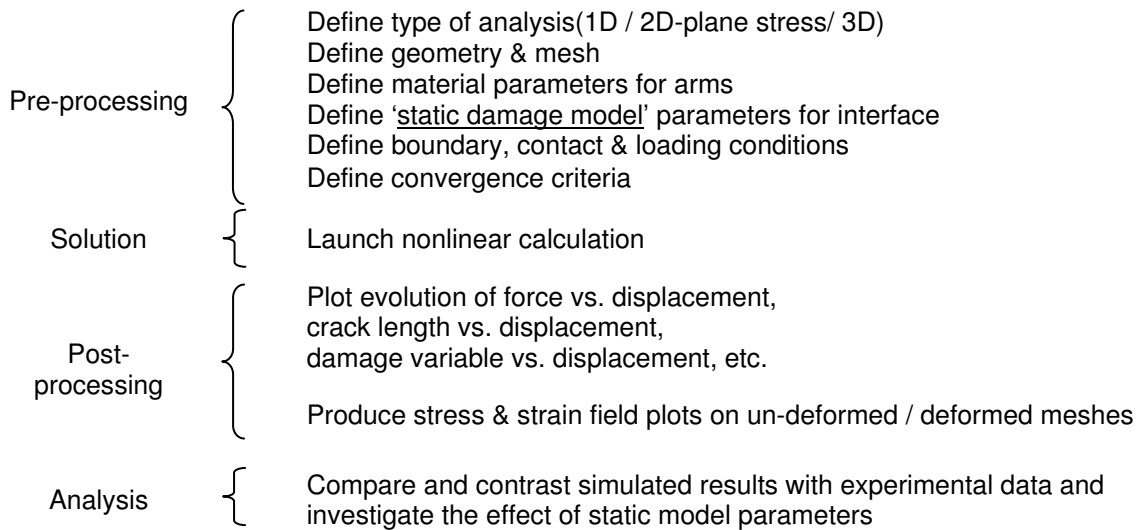
$$\begin{pmatrix} \sigma_{13} \\ \sigma_{23} \\ \sigma_{33} \end{pmatrix} = \begin{pmatrix} k_1^0(1-\bar{d}_1) & 0 & 0 \\ 0 & k_2^0(1-\bar{d}_2) & 0 \\ 0 & 0 & k_3^0(1-\bar{d}_3) \end{pmatrix} \begin{pmatrix} U_1 \\ U_2 \\ U_3 \end{pmatrix} \quad (3.16)$$

# Chapter 4

## Analysis of Static Simulations

### 4.1 Analysis Procedure

Given hereunder is the routine of a finite element simulation procedure and subsequent post treatments carried out for validation of the interface damage model.



The finite element implementation of the 'static damage model' for the interface in 3D is summarized in the Table 4.01 given below.

Step-1	Compute damage energy release rate, i.e., $\bar{Y}(t)$ using $Y_{d_1}, Y_{d_2}$ & $Y_{d_3}$
Step-2	Compute isotropic damage function, i.e., $w(\bar{Y})$ using $\bar{Y}, Y_C$ & $Y_0$
Step-3	Compute damage variables, i.e., $d_1 = d_2 = d_3 = w(\bar{Y})$
Step-4	Compute stresses for the interface, i.e., $\sigma_{13} = k_1^0(1-d_1)U_1, \sigma_{23} = k_2^0(1-d_2)U_2, \sigma_{33} = k_3^0(1-d_3)U_3$

Table 4.01 : FE implementation of static damage model

The constitutive law for the inter-laminar interface will be introduced in to the finite element formulation of CAST3M via UMAT.

One of the useful advantages of the sub-routine is its capability to print STATE VARIABLE values (i.e., stress, strain, damage variable, energy release rate, etc) at end of each loading increment during nonlinear calculation process. This option helps the user to properly calibrate the damage model with relative ease.

## 4.2 Identification of Static Damage Model Parameters

Prior to the commencement of simulations one needs to identify the model parameters of the damage evolution law. Parameters,  $Y_C$ ,  $Y_0$ ,  $\gamma_1$ ,  $\gamma_2$  and  $\alpha$  are computed based on values of the critical energy release rates of each failure mode. Thus they are strongly depended on the mechanical behaviour of the material and therefore called ‘material parameters’. In addition,  $k_1^0$ ,  $k_2^0$  (initial shear stiffness) and  $k_3^0$  (initial normal stiffness) are computed using maximum admissible stresses of the inter-laminar interface [Ladevèze P. et al., 1998; Ijaz H., 2009]. The parameter  $n$  has a significant influence on initiation of delamination and subsequent evolution of damage variable. In Section 3.1, the formulas for computing  $Y_C$ ,  $\gamma_1$  and  $\gamma_2$  are presented. In cases where inter-laminar fracture toughness for Mode-III is not available, one can take  $\gamma_2 = \gamma_1$  [Ladevèze P. et al., 1998] for 3D analyses.

### 4.2.1 Identification of Parameter - $\alpha$

Procedure for identification of  $\alpha$  is explained hereafter. Let us consider a state of stress, where Mode-III fracture toughness is considerably negligible. Now, the reduced equation can be written as,

$$\left( \frac{G_I}{G_{IC}} \right)^\alpha + \left( \frac{G_{II}}{G_{IIC}} \right)^\alpha = 1 \quad (4.01)$$

The value of  $\alpha$  can be obtained using the normalized Mode-I / Mode-II plane. Experimental results on critical energy release rate for pure Mode-I, pure Mode-II and Mixed-Mode (at least one data point) are used in the identification process. The example given hereunder considers the data reported [Rikard Borg et al., 2003] on a unidirectional composite laminate - HTA/6376C. The critical energy release rate (fracture toughness) for each failure mode is listed in the Table 4.02.

Failure Mode (Test Method)	$G_C$ [N / mm]
Mode-I (DCB)	0.259
Mixed-Mode (MMB) $G_{II} / (G_I + G_{II}) = 0.5$	0.447
Mode-II (3ENF)	1.008

Table 4.02 : Critical energy release rates of HTA/6376C for different failure modes

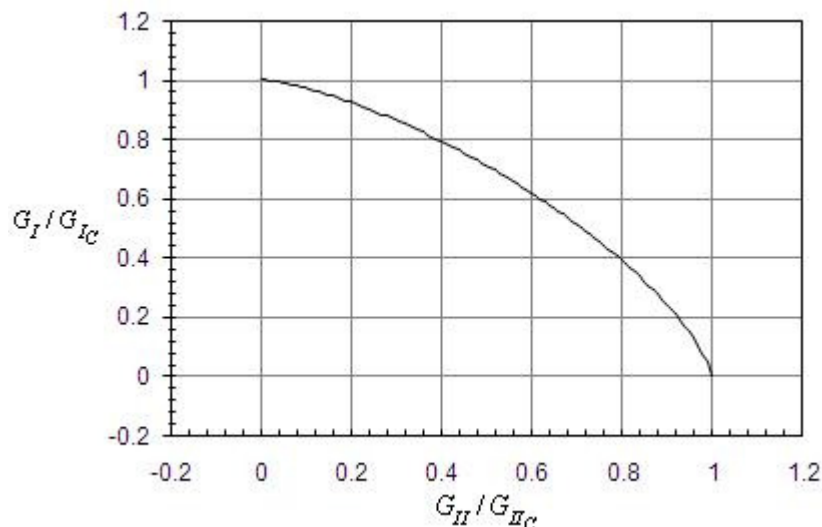


Figure 4.01 - Normalized Mode-I / Mode-II plane for HTA/6376C

Figure 4.01 given above shows the normalized Mode-I / Mode-II plane, developed using the available experimental results included in Table 4.01. The best fit of plot can be obtained when  $\alpha = 1.4$ . The good value of  $\alpha$  is usually identified by using several experimental results obtained for different mode mixtures. Note that,  $\alpha$  governs the shape of the failure locus in Mixed-Mode and has no influence on the evolution of the damage variable in either pure Mode-I or Mode-II cases. Typically, value of  $\alpha$  for most fibrous composites is in between 1 and 2 (i.e.,  $1 \leq \alpha \leq 2$ ) [Harper P. W., 2008].

#### 4.2.2 Identification of Initial Interface Rigidities

An ‘instability criteria’ [Gornet L., 1997; Ijaz H., 2009] has been derived to find the relationship between the maximum admissible stress (i.e.,  $\sigma_{3i}^{th}$   $i=1,2$  and 3) at the interface and the damage model parameters. Through this formulation one can determine the initial rigidities (i.e.,  $k_i^0$   $i=1,2$  and 3) of the interface.

For each failure mode the instability criteria can be expressed as follows,

$$\begin{aligned} \text{Mode-I} & : \dot{\sigma}_{33} = 0 \quad \text{for} \quad \dot{U}_3 \neq 0 \\ \text{Mode-II} & : \dot{\sigma}_{31} = 0 \quad \text{for} \quad \dot{U}_1 \neq 0 \\ \text{Mode-III} & : \dot{\sigma}_{32} = 0 \quad \text{for} \quad \dot{U}_2 \neq 0 \end{aligned}$$

For Mode-I, differentiating normal stress;  $\sigma_{33} = k_3^0 (1 - d_3) U_3$  with respect to  $U_3$  we have,

$$\frac{d\sigma_{33}}{dU_3} = k_3^0 \left[ (1 - d_3) + U_3 \frac{d}{dU_3} (1 - d_3) \right]$$

Next by considering equations 3.03 and 3.12

$$\begin{aligned} \frac{d\sigma_{33}}{dU_3} & = k_3^0 \left[ (1 - d_3) - U_3 \frac{d}{dY_{d_3}} W(Y_{d_3}) \cdot \frac{d}{dU_3} (Y_{d_3}) \right] \\ \frac{d\sigma_{33}}{dU_3} & = k_3^0 \left[ (1 - d_3) - 2Y_{d_3} \frac{d}{dY_{d_3}} W(Y_{d_3}) \right] \end{aligned} \quad (4.02)$$

According to instability criteria:  $\frac{d\sigma_{33}}{dU_3} = 0$ , which leads to the following expression,

$$(1 - d_3) - 2Y_{d_3} \frac{d}{dY_{d_3}} W(Y_{d_3}) = 0 \quad (4.03)$$

Now, one can re-write equation 4.03 in a more general form considering all failure modes.

$$(1 - d) - 2\bar{Y} \frac{d}{d\bar{Y}} W(\bar{Y}) = 0 \quad (4.04)$$

$$\begin{aligned}
\text{Mode-I} & : d = d_3 \quad \text{and} \quad \bar{Y} = Y_{d_3} \\
\text{Mode-II} & : d = d_1 \quad \text{and} \quad \bar{Y} = \gamma_1 Y_{d_1} \\
\text{Mode-III} & : d = d_2 \quad \text{and} \quad \bar{Y} = \gamma_2 Y_{d_2}
\end{aligned}$$

By considering equation 3.12 one can further expand equation 4.04 as follows,

$$\begin{aligned}
& \left( 1 - \left[ \frac{n}{n+1} \frac{\bar{Y} - Y_0}{Y_C - Y_0} \right]^n \right) - 2\bar{Y} \frac{d}{d\bar{Y}} \left( \left[ \frac{n}{n+1} \frac{\bar{Y} - Y_0}{Y_C - Y_0} \right]^n \right) = 0 \\
& 1 - \left( \frac{n}{n+1} \right)^n \hat{Y}^n - 2n \frac{\bar{Y}}{Y_C - Y_0} \left( \frac{n}{n+1} \right)^n \hat{Y}^{n-1} = 0 \quad \text{where, } \hat{Y} = \frac{\bar{Y} - Y_0}{Y_C - Y_0} \\
& (2n+1) \hat{Y}^n + 2n \frac{Y_0}{Y_C - Y_0} \hat{Y}^{n-1} - \left( \frac{n+1}{n} \right)^n = 0 \tag{4.05}
\end{aligned}$$

Analytical relation for initial interface rigidities in each direction can be derived from the above equation by taking,  $Y_0 = 0$ .

For Mode-I, considering equation 4.05 we have,

$$(2n+1) \left( \frac{Y_{d_3}}{Y_C} \right)^n - \left( \frac{n+1}{n} \right)^n = 0 \tag{4.06}$$

Now substituting equation 3.12 in  $\sigma_{33} = k_3^0 (1 - d_3) U_3$  with  $Y_0 = 0$ ,

$$\sigma_{33} = k_3^0 \left[ 1 - \left( \frac{n}{n+1} \frac{Y_{d_3}}{Y_C} \right)^n \right] U_3 \tag{4.07}$$

Considering equations 4.06 and 4.07 one can derive,

$$\sigma_{33} = \frac{2n}{2n+1} k_3^0 U_3 \tag{4.08}$$

Now by considering equations 3.03, 4.06 and 4.08 the initial interface rigidity in the out-of-plane direction (i.e., Mode-I) can be expressed as follows,

$$k_3^0 = \frac{(2n+1)^{2n+1/n}}{8n(n+1)Y_C} \sigma_{33}^2 \tag{4.09}$$

Similarly, the initial interface rigidities in the in-plane directions (i.e., Mode-II and Mode-III, respectively) can be expressed as follows,

$$k_1^0 = \frac{\gamma_1 (2n+1)^{2n+1/n}}{8n(n+1)Y_C} \sigma_{31}^2 \tag{4.10}, \quad k_2^0 = \frac{\gamma_2 (2n+1)^{2n+1/n}}{8n(n+1)Y_C} \sigma_{32}^2 \tag{4.11}$$

The initial interface rigidities can then be computed using above equations, if experimental data on maximum admissible stresses for the interface and critical energy release rate for Mode-I are available.. Note that, the value of 'n' will be assumed based on the brittleness of the interface. Usually,  $n < 1$  is used for brittle interfaces.

For a unidirectional composite laminate - HTA/6376C, maximum admissible stresses for the interface were reported [Robinson P. et al., 2005] to be approximately equal to 30 MPa (i.e.  $\sigma_C = \sigma_{33} = \sigma_{13} = 30 [MPa]$ ). By using the above equations and assuming  $n = 0.5$ , the normal and shear rigidities can be found  $k_3^0 = 9265 [MPa / mm]$  and  $k_1^0 = 2380 [MPa / mm]$ .

The results given hereunder illustrate the model response for both Mode-I and Mode-II failure modes for simple test-cases. The laminate material is considered to be HTA/6376C with  $[0/0]$  interface. Reported mechanical properties [Rikard Borg et al., 2003] of the lamina are included in Table 4.03.

$E_{11}$ - 146,000 [MPa]	$E_{22}$ - 10,500 [MPa]	$E_{33}$ - 10,500 [MPa]
$\nu_{12}$ - 0.3	$\nu_{23}$ - 0.51	$\nu_{13}$ - 0.3
$G_{12}$ - 5,250 [MPa]	$G_{23}$ - 3,480 [MPa]	$G_{13}$ - 5,250 [MPa]

Table 4.03 : Elastic properties of HTA/6376C unidirectional prepregs reported by Rikard B. et al.

Figure 4.02 given below illustrate the geometry of the test-piece, loading schemes i.e., imposed displacements;  $U_3$  for Mode-I and  $U_1$  for Mode-II, and boundary conditions.

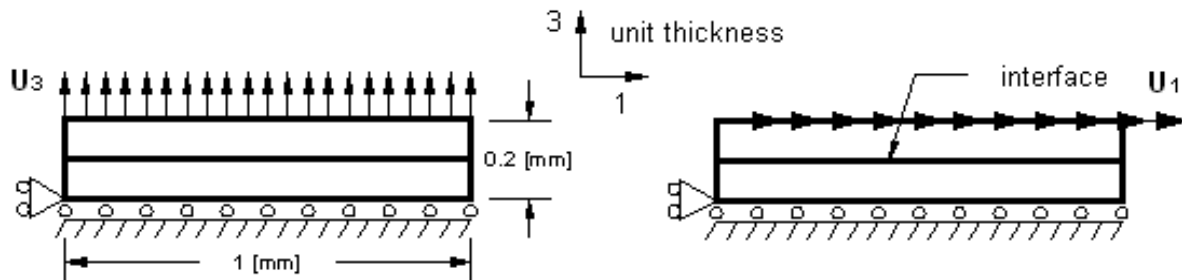


Figure 4.02 : Test-piece geometry and loading/boundary conditions

Simulations were performed considering a state of 'PLANE STRESS' in 2D. Values of the material parameters and FEs used are given in Table 4.04.

$Y_C [N / mm]$	0.259	Body (2D)	Geo. Support / Finite Element : QUA4 / QUA4
$\gamma_1$	0.2569		Deg. of Interpolation : 1
$\alpha$	1.4	Interface(1D)	Geo. Support / Finite Element : RAC2 / JOI2
$n$	0.5		Deg. of Interpolation : 1

Table 4.04 : Damage model parameters and 2D mesh details for simulations of test case

The mechanical behaviour of the 'Body' will be simulated with the classical constitutive law (i.e.,  $\sigma = k\varepsilon$ ) using solid elements. At the same time, the interface elements are used simulate the degradation of the interface using the modified constitutive law (i.e.,  $\sigma = (1 - d)k\varepsilon$ ) formulated according to damage mechanics. The damage model is simply responsible for computing the damage variable (i.e.,  $d$ ) at the end of each increment (i.e., force or imposed displacement). Note that, interface elements are zero thickness elements having same size of the solid elements (length and width directions) adjacent to them. Figure 4.03 illustrates the relative location of the interface elements for 1D and 2D cases.

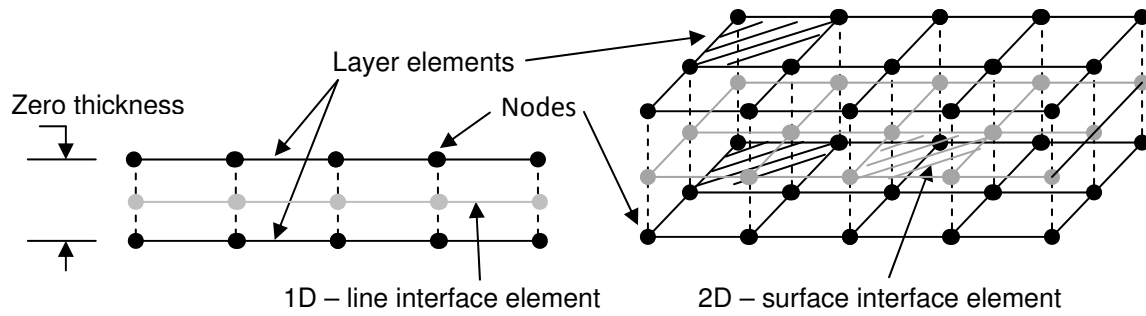


Figure 4.03 : Pictorial views of interface elements for 1D and 2D cases

The mechanical behaviour of the interface elements are mainly depended on the interface rigidities. Initially, there exist no crack, but with increasing displacement the damage starts to accumulate and as a consequence the normal interface rigidity degrades. As explained earlier the initial rigidities in normal and shear directions will be determined using the equations 4.09, 4.10 and 4.11.

Note that, the index given at the end of the element name (i.e., 4 for QUA4 and 2 for JOI2) denotes the number of nodes associated to each type of element. Interestingly, in CAST3M (2007, 2008 and 2009 versions) the gauss points of the interface elements are also located at the nodal points. Therefore, both stress and strain as well as all other user defined STATE VARIABLES (i.e., energy variables, damage variables, etc) are also calculated at the nodal points of the interface elements.

Table 4.05, includes plots representing ‘force vs. displacement’, ‘damage vs. displacement’ and ‘energy release rate ( $G$ ) vs. displacement’ for each failure mode. All simulations were performed up to the final failure (complete debonding).

	<i>Mode – I</i>	<i>Mode – II</i>
<i>Force [N] vs. Disp. [mm]</i>		
<i>Damage vs. Disp [mm]</i>		
<i>G [N/mm] vs. Disp [mm]</i>		

Table 4.05 : Evolution of force, damage and strain energy with displacement for the test case

Numerical simulations were also carried out in 3D with same material parameters. The types of FEs used are included in Table 4.06.

Body (3D)	Geo. Support / Finite Element : CUB8 / CUB8 Deg. of Interpolation : 1
Interface(2D)	Geo. Support / Finite Element : LIA4 / JOI4 Deg. of Interpolation : 1

Table 4.06 : 3D mesh details for simulation of test case

3D simulation results were identical to the ones obtained from 2D analysis. 3D model did not exhibit any spurious localization as a result of its simple geometry. However, the computational cost increased notably.

During the debonding process the damage variable increases from '0' to '1' and energy release rate from '0' to ' $G_C$ ' ( $=0.259$  for Mode-I and  $=1.008$  for Mode-II) simultaneously. Here, the evolution of energy release rate has been determined using global force and displacement variables. Note that, the area under the force vs. displacement curve is equal to the critical energy release rate for each failure mode. However, for analysis purposes it would be useful to determine the strain energy released in terms of local variables (i.e., stress and strain) during the calculation process. This is achieved by using the trapezoidal rule to compute the area under the 'stress vs. strain' curve at each increment and adding them progressively. Here, the computations are performed at each gauss points of the interface elements (in CAST3M the gauss points and nodal points are located at the same position).

Note that, classically energy release rate is defined in terms of failed surface area; therefore it can attributed as an area parameter. On the other hand, when strain energy release rate is calculated at the gauss points it is no more a physical quantity but a mere representation of this area parameter by numerical means. It is the responsibility of the user to properly appreciate the correlation between this numerical variable and the physical quantity. When the relation is established it serves as a useful numerical tool in determining the energy variation in the system. Theory associated to this methodology for determining the energy release rate was discussed earlier in Section 3.1.

Figure 4.04 given below illustrates the evolution of the energy release rate at the interface for Mode-I failure using both global and local variables. Note that, for the considered test-case, degradation of the entire interface takes place homogeneously and delamination happens all at once. In other words, crack front is not a point but a line. Hence, any selected gauss point of a given interface element can be indirectly used to determine the evolution of the energy release rate variable.

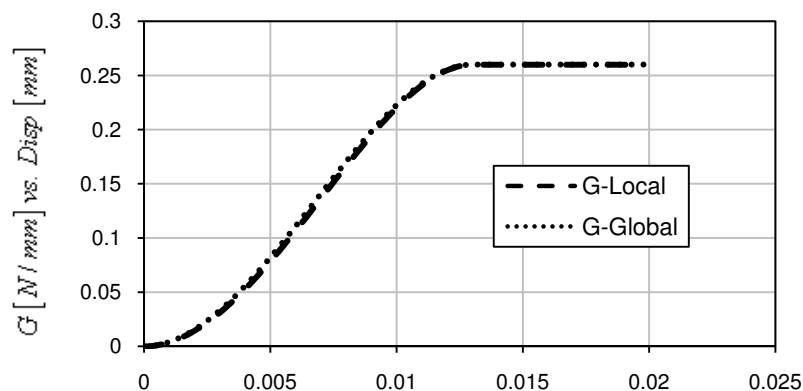


Figure 4.04 : Comparison of G-Global and G-Local

As it can be seen from the above figure, the variation of this numerical quantity can be directly mapped on to the evolution of the energy release rate obtained by the global variables. Therefore, this information justifies the applicability of the assumption for determination of energy release rate in terms of local variables.



Influences of parameter ' $n$ ' and admissible interfacial stresses on the model's response can be understood by investigating the force vs. displacement plots for Mode-I failure included in Table 4.07.

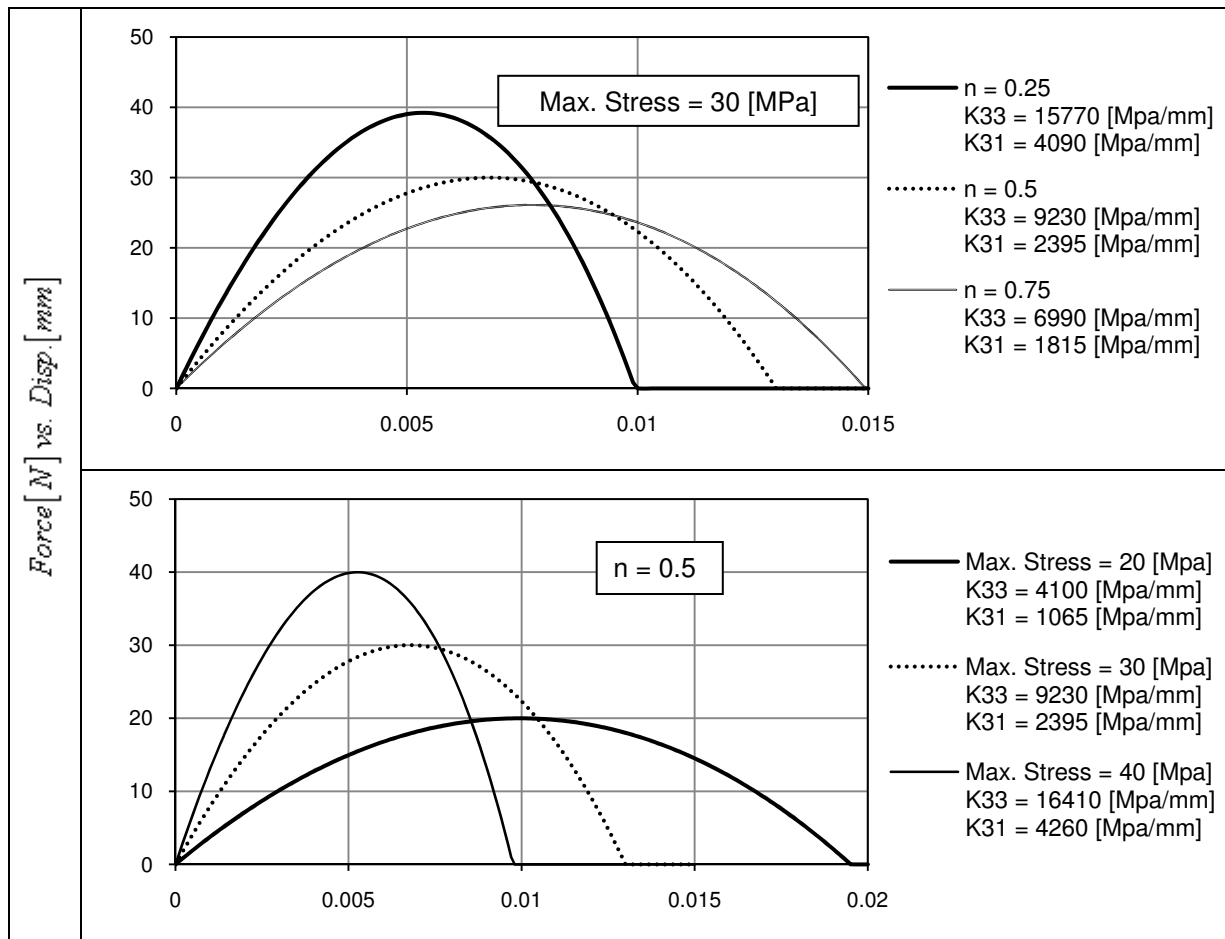


Table 4.07 : Evolution of force with displacement for different  $n$  and critical stress values

For smaller values of ' $n$ ' the computed interfacial rigidities are larger in magnitude and therefore the initial slope of the force-displacement curve will decrease as ' $n$ ' increases. On the other hand, as the maximum admissible stress is increased, the slopes of the curves tend to decrease. However, this phenomenon will not affect the computed maximum stress and critical energy release rate at the interface. In addition, both parameter ' $n$ ' and admissible stress have a significant influence on the maximum force computed and critical displacement for complete failure. Therefore, final results of the simulations are believed to be considerably sensitive to these parameters.

In cases where values of interfacial stresses are not available or not certain, the interfacial rigidities can be found by doing a comparison between numerical and experimental results under static loading conditions. Here, values for parameter ' $n$ ' could be determined through a trial-and-error approach. Usually domain of ' $n$ ' for brittle interfaces is considered to be  $0 < n < 1$ .

### 4.3 Overview of Fracture Mechanics Tests for Delamination

Fracture toughness of fiber reinforced composites is classically determined using test methods developed for unidirectional laminates. This is mainly due to intra ply damage, undesired edge effects and crack plane migration effects associated to multidirectional laminates. Although inter-laminar fracture toughness can be measured in any combination of three fracture modes, the most commonly used ones are Mode-I (opening), Mode-II (shearing) or combination of both [ASTM standards; Brunner A. J., et al, 2001a/2001b]. Note that, Mode-III (tearing) fracture toughness is relatively larger in magnitude and contribution to delamination is comparatively low.

Usually the fracture toughness is measured using a beam-type composite specimen with an initial crack between central plies. The initial delamination is introduced into the laminate by a thin non-stick film, usually made of polymer. Therefore, upon application of loading, delamination grows parallel to the plies in the mid-plane of the beam. Interestingly, during the propagation stage one can observe a non-straight crack front. This is mainly due to the variation of the energy release rate across the width of the specimen. While performing the test, both applied load and corresponding displacement are measured and are correlated to the delamination length. From correlated data, the inter-laminar fracture toughness will be obtained using simple beam theory principles or other empirical formulations.

Typically ‘Double Cantilever Beam (DCB)’ test method is used to obtain experimental inter-laminar fracture toughness for Mode-I. Note that, the initial delamination is introduced in the mid-plane of the beam at one end and the other end will be fixed. The initial delamination is then forced to grow by pulling the two beams at the free end in opposite directions (opening). A schematic illustration of the DCB test with loaded and unloaded conditions is given below in Figure 4.05.

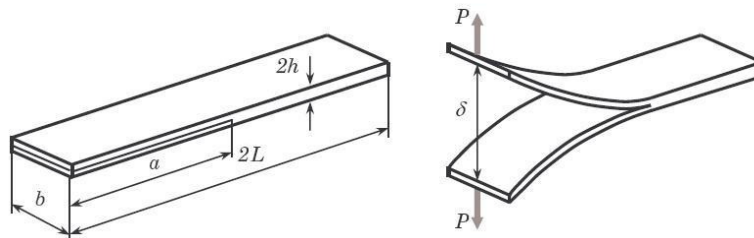


Figure 4.05 : DCB test with loaded and unloaded conditions

‘Three-point End Notched Flexure (3ENF)’ test is one of the most commonly used experimental methods for obtaining Mode-II inter-laminar fracture toughness. Here, initial delamination is forced to grow in the mid-plane by making crack faces slide (shearing) relative to each other. This is achieved by simply supporting the beam at the two ends and applying a compressive force at the mid span of the beam. Figure 4.06 given below shows a schematic of a 3ENF test with loaded and unloaded conditions.

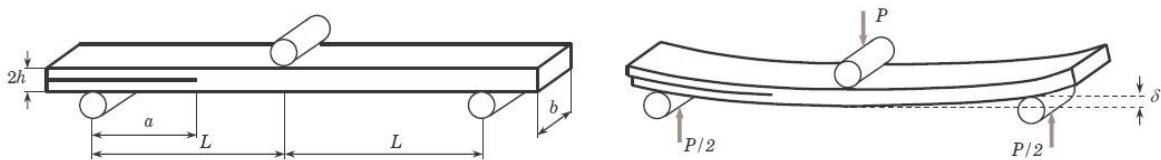


Figure 4.06 : 3ENF test with loaded and unloaded conditions

Mixed-Mode I/II inter-laminar fracture toughness is commonly evaluated using a ‘Mixed Mode Bending (MMB)’ test apparatus [Crews J. H. and Reeder J.R., 1988]. Here, initial delamination is forced to propagate by both opening and shearing actions. The test allows the determination of fracture toughness with a Mixed-Mode ratio ranging from pure Mode-I to pure Mode-II. Figure 4.07 given below illustrates a schematic of the test rig (source - Blanco Villaverde N., 2004). Usually different Mixed-Mode ratios are achieved by varying the position of the load point ( $c$ ) along the lever arm.

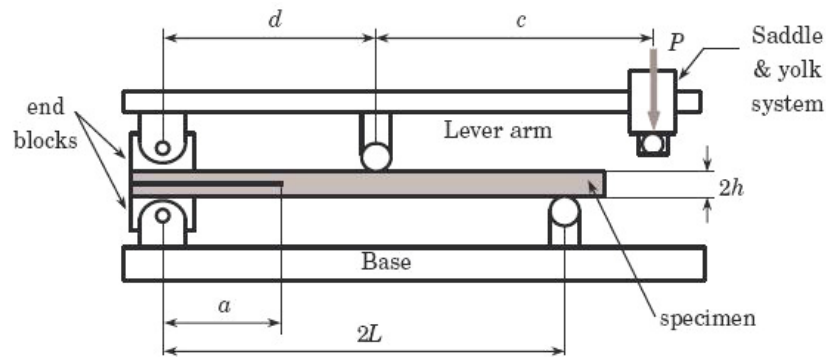


Figure 4.07 : Schematic of MMB test rig

#### 4.4 Validation of Static Damage Models

The following sections are dedicated for analyzing the applicability and effectiveness of the static damage models. Verifications will be done by closely examining the simulated results with experimental results. The two static damage models will be referred from hereafter as,

- I. Local static damage model
- II. Nonlocal static damage model (integral-type)

Previous studies [Allix O. and Ladevèze P., 1992; Corigliano A. and Allix O., 1996; Gornet L. et al 1997] shows that local damage model has a high potential for determining delamination in any given failure mode. The purpose of the ongoing investigation is to check its validity in all dimensional (1D, 2D or 3D) spaces and to provide a synopsis of its limitations and advantages.

Next, the FE implementation of the proposed nonlocal model will be discussed in details. In addition, the model will also be subjected to a thorough investigation to understand its behaviour. The analysis will include determination of influence of different parameters, such as element size, increment size, etc. However, the nonlocal model will only be tested for 2D applications.

Note that, the experimental results reported by Rikard Borg and coworkers [Rikard Borg et al, 2003] will be used to perform all verifications.

The investigations will be carried out considering pure Mode-I and Mode-II failure modes with DCB and 3ENF tests respectively. All specimens experimented were manufactured from HTA/6376C carbon/epoxy unidirectional prepreps and their elastic properties are given in Table 4.03. The dimensions of the rectangular shaped specimens are; width  $b = 20 [mm]$ , length  $l_s = 150 [mm]$ , and nominal thickness  $2h = 3.1 [mm]$ . In addition, the testing span of the specimen,  $2L = 100 [mm]$  and initial delamination length,  $a = 35 [mm]$ . Schematics of the specimens and their dimensions are illustrated in Figure 4.05 (DCB) and Figure 4.06 (3ENF). In addition, fracture toughness for each failure mode is tabulated in Table 4.02. Finally, values of the material parameters and initial interface rigidities are determined as explained in Section 4.2. Typically, for CFRP composites the critical stress value is believed to be in the range of 20 to 40 [MPa]. Note that, for the present study  $\sigma_c$  will be taken equal to 30 [MPa]. Previously, in Section 4.2.2 few computations were performed on a simple test-case considering Mode-I failure to understand model's behaviour for different values. However, to obtain a broader understanding, it is necessary to compare simulated results for different critical stress values with actual experimental results.

Since the investigation is twofold, the validations will be done in two different sections; one on the local model and another one on the nonlocal model.

## 4.5 Validation of Local Static Damage Model

### 4.5.1 FE Simulation Details

The local damage model was tested systematically in 1D, 2D and 3D spaces. The types of FEs used in each space are given in Table 4.08.

	Laminate arms	Inter-laminar interface
1D	Geo. Support / Finite Element : SEG2 / COQ2 Deg. of Interpolation : 1	Geo. Support / Finite Element : RAC2 / JOI2 Deg. of Interpolation : 1
2D	Geo. Support / Finite Element : QUA4 / QUA4 Deg. of Interpolation : 1	Geo. Support / Finite Element : RAC2 / JOI2 Deg. of Interpolation : 1
3D	Geo. Support / Finite Element : QUA4 / COQ4 Deg. of Interpolation : 1	Geo. Support / Finite Element : LIA4 / JOI4 Deg. of Interpolation : 1

Table 4.08 : Types of FE used for 1D, 2D and 3D models of DCB and 3ENF tests

Note that, in CAST3M the degree of the interpolation function associated to both 'RACCORD' (1D) and 'LIAISON' (2D) type interface elements are limited to '1'. As a consequence, laminate arms were also needed to be meshed with elements having interpolation functions of the same degree. In other words, compatibility of the elements used for the interface and laminate arms should be ascertained prior to the commencement of the simulations. Usually, for the case of simple beam bending, a coarse mesh with elements having degree-2 interpolation functions (i.e., QUA8 for 2D and CU20 for 3D) will be sufficient for simulating the bending behaviour accurately. However, to obtain a prediction using degree-1 elements (i.e., QUA4 for 2D and CUB8 for 3D) with similar accuracy, the mesh size needs to be reduced significantly.

In all models the laminate arms were formulated with 'ELASTIC ORTHOTROPIC' material behaviour. Note that, in previous studies some authors had introduced isotropic behaviour to obtain satisfactory results. However, the unidirectional composites are orthotropic by nature (see Table 4.03); therefore it is necessary to consider orthotropic behaviour for laminate arms to properly model the delamination characteristics. Although delamination is a phenomenon happening at the interface it is affected by the behaviour of the arms connected to the interface.

Here, both COQ2 and COQ4 belong to the 'SHELL' elements category and were found to be tailor-made for the problem in hand. For the 1D case, COQ2 elements allow the user to define elastic properties in both longitudinal and thickness directions, along with Poisson ratio and shear modulus. On the other hand, for the 3D case, COQ4 elements allow the user to define additional transverse shear components. The 2D simulations were performed considering a state of 'PLANE STRESS' ('PLANE STRAIN' also possible) along with a more comprehensive list of orthotropic material definitions. Table 4.09 given below includes a complete list of mechanical variables taken in to account by each type of elements.

Thin-Shell(1D) : COQ2	Solid(2D) : QUA4	Thin-Shell(3D) : COQ4
$E_{11}, E_{22}$	$E_{11}, E_{22}, E_{33}$	$E_{11}, E_{22}$
$\nu_{12}$	$\nu_{12}, \nu_{23}, \nu_{13}$	$\nu_{12}$
$G_{12}$	$G_{12}, G_{23}, G_{13}$	$G_{12}, G_{23}, G_{13}$

Table 4.09 : Mechanical variables associated to solid and shell elements

Note that, all computations were performed with imposed displacement loading condition as it was the case with the two experimental tests.

Since the two laminate arms are modeled separately it was necessary to take in to account the 'contact' action taking place between them. Note that, in reality the laminate layers will not penetrate each other upon compression. If the contact condition is not properly defined the meshes of the two arms will move freely and overlap each other as shown in Figure 4.08 (mesh with QUA4 elements). This issue has a significant effect on the 3ENF test compared to DCB test, because for 3ENF the loading direction is compressive by nature.

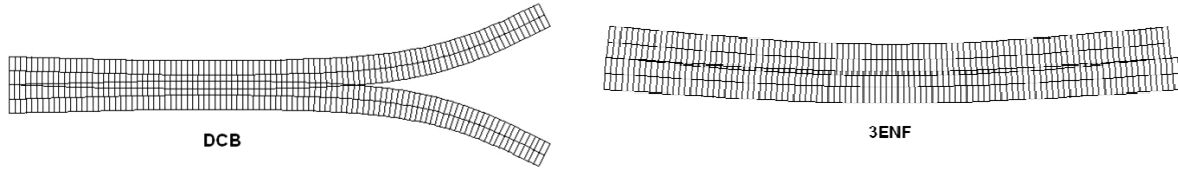


Figure 4.08 : Overlapping of the arms when contact condition not specified

In general, interface elements are designed to handle only tensile and shearing forces. Therefore, from a modelling point of view it is mandatory to restrain the motion of the bottom and top edges (or surfaces in 3D) of the top and bottom arms in the compression direction. In CAST3M, the contact relation can be defined with the operator named, 'RELA'. The command allows the user to define the minimum distance between two entities that is to be maintained at any given loading condition. In CAST3M, the 'PASAPAS' (meaning 'step-by-step') operator handles the nonlinear calculation process. The model definitions, material definitions, element rigidities, boundary conditions and loading conditions are given as input to this command. In addition, parameters relating to convergence criteria are also defined explicitly with PASAPAS.

The two main variables associated to the convergence criteria in CAST3M are 'Force Tolerance (FTOL)' and 'Moment Tolerance (MTOL)'. By default both of these values are set to  $1.0E-5$ . However for the present study, convergence of the numerical solutions can be achieved at a much larger tolerance level. As it will be revealed in the Fatigue study section, many number of simulations are required to calibrate the fatigue model and if the convergence tolerance is too severe then the computational cost (time) will become very expensive. Therefore, for a given type of test (i.e., DCB, 3ENF) with a fixed mesh the optimum convergence limits needs to be identified. The details of the convergence analysis performed on this regard are included in the later part of the upcoming sections.

### 4.5.2 Simulation Results for DCB Test

Application of the boundary conditions and loading conditions (i.e., imposed displacement) for 1D, 2D and 3D DCB test simulations are shown in Table 4.10.

1D	
2D	
3D	

Table 4.10 : Boundary and loading conditions for DCB static test

In order to determine the most optimum mesh size, a convergence analysis was also performed. In general, numerical solutions converged for element sizes smaller than  $0.2 [mm]$  (in the longitudinal direction). Details of the convergence analysis are presented at the end. Figure 4.09 given below shows a comparison of experimental and simulated results. Note that, all simulations were done with displacement increments equal to  $0.02 [mm]$ . Convergence limits were set to  $1.0E - 2$ .

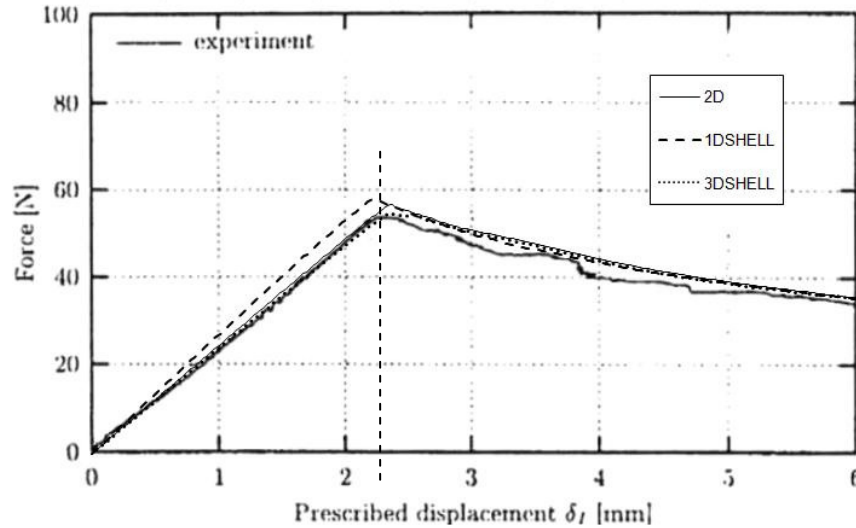


Figure 4.09 : Force vs. Displacement for DCB – simulated results for 1D, 2D and 3D models

Simulated results for 2D and 3D are in good agreement with the experimental results. However, for 1D, simulated result shows a small increase of the stiffness prior to crack propagation. All simulated curves attain their maximum values approximately at a similar displacement level (see vertical dotted line on figure drawn at  $2.3 [mm]$ ). Here, percentage errors for 1D, 2D and 3D are  $4.4%$ ,  $1.7%$  and  $1.7%$  respectively. Similarly, the maximum value of the force was also predicted well within an acceptable range. Percentage errors of prediction for 1D, 2D and 3D are  $7.5%$ ,  $4.9%$  and  $1.7%$  respectively. Although, 1D result shows some deviation at first, the propagation stage has been predicted with good accuracy compared to 2D and 3D results.

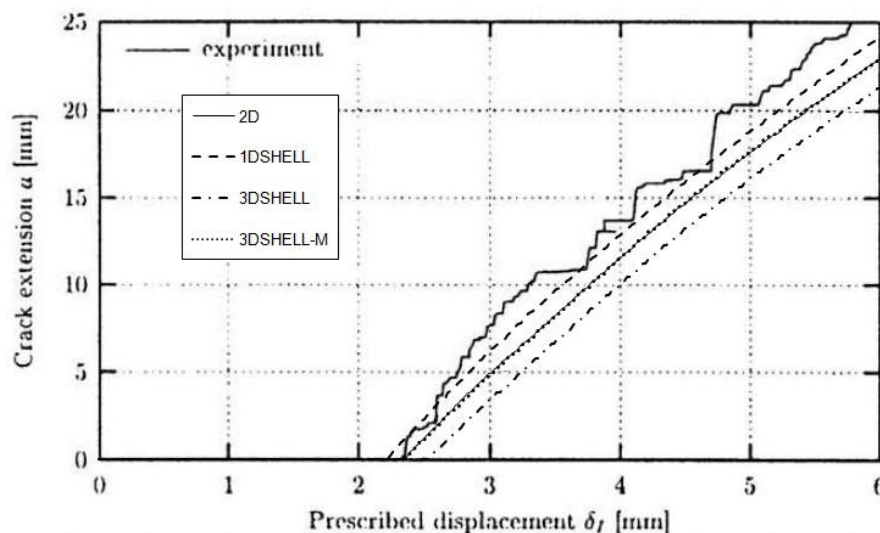


Figure 4.10 : Crack length vs. Displacement for DCB – simulated results for 1D, 2D and 3D models

Figure 4.10 given above illustrates the evolution of crack length against displacement for all simulations. The predictions seem to be satisfactorily when compared to experimental result. Interestingly, the experimental result shows some erratic variations which may probably be caused by vibration of the test piece during measurement. For all simulations, the prediction of initiation of crack

growth seems to be within an acceptable accuracy. In general, the initiation points directly correspond to the points at which the force vs. displacement curves starts to change direction. However, in 3D simulations the crack initiation seems to take place with a small delay. As it has been pointed out earlier, in 3D, crack front is a line. Therefore, crack growth in both center and outer edges of the model need to be determined. Two curves, named 3DSHELL and 3DSHELL-M given in the above figure refer to the crack growth at the outer edges and middle of the specimen. Figure 4.11 given below can be used to visualize the crack growth paths considered for plotting.

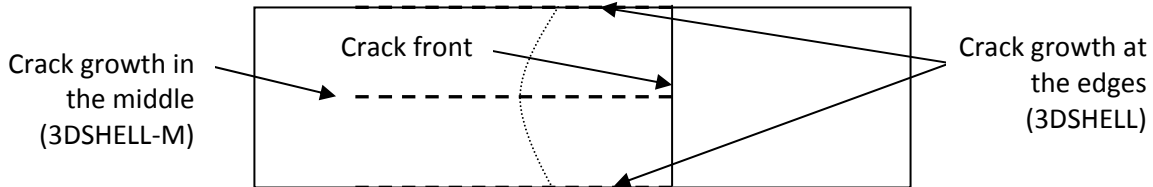


Figure 4.11 : Crack paths monitored for plotting

From these results it can be deduced that crack growth in the center of the specimen happens earlier than the outer edges. Notice that, the two evolution remains roughly parallel to each other. This suggests that the crack front moves forward with a curvature and the shape of the curvature remains roughly the same throughout the test. The stress field plots given in Figure 4.12 can be used to visualize this phenomenon. The stress fields given below have been produced using normal stress component (i.e., SMN) of the interface elements. Note that, the shear stress components are negligible because the tests were performed specifically to achieve Mode-I or Opening Mode failure.

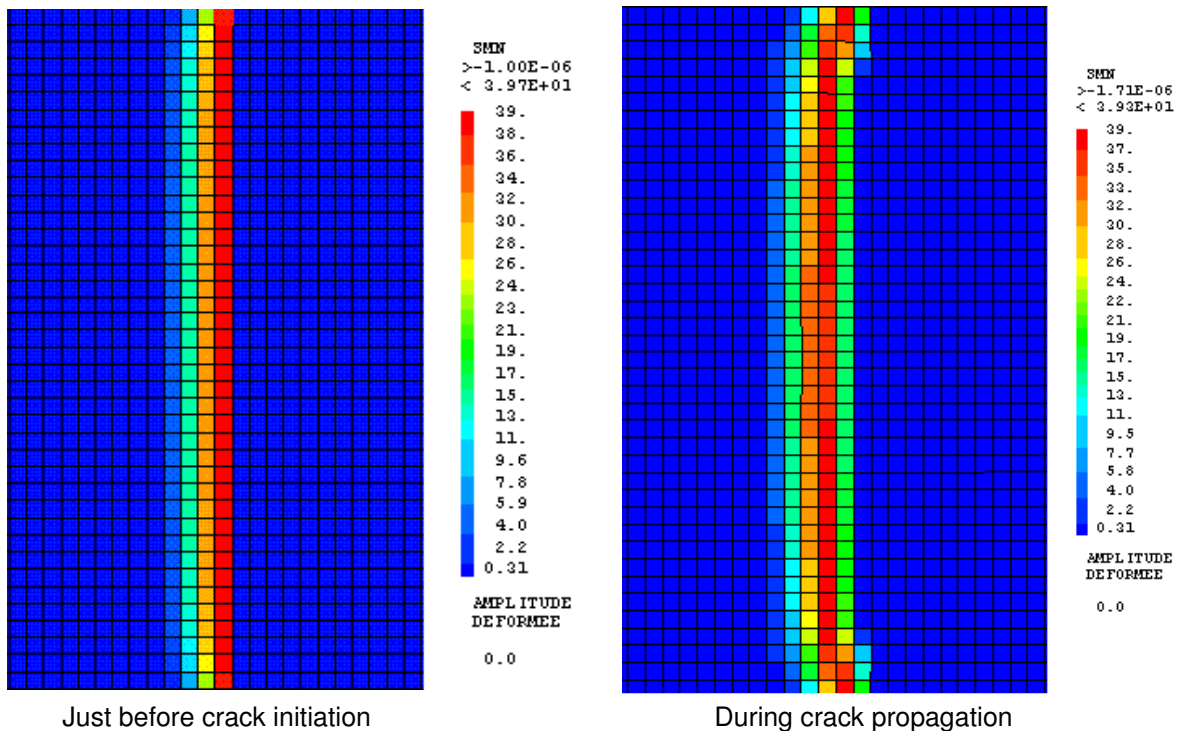


Figure 4.12 : Normal stress field at the crack front for DCB test

These stress fields also provide an insight to the shape of the process zone ahead of the crack front. Even before crack initiation, the intensity of stress fields close to the outer edges tends to be lesser than the central region. At the same time, evolution of the damage variable at the center happens at a faster rate compared to the outer edges. As the displacement of the arms increases the stress field also changes gradually. When the propagation stage is reached the stress field remains roughly the same until the end of the test. Note that, intensity of the stress field is a good representation of the shape of the crack front. Here, experimental result used for the comparison has been obtained by examining the delamination growth at the outer edges of the specimen. The predicted initiation point at the outer edge of the model has a percentage error of 8%.

The principal stress fields of the laminate arms of the 2D model are shown in Figure 4.13. Note that, plots have been made at a prescribed displacement of  $2.6 [mm]$ .

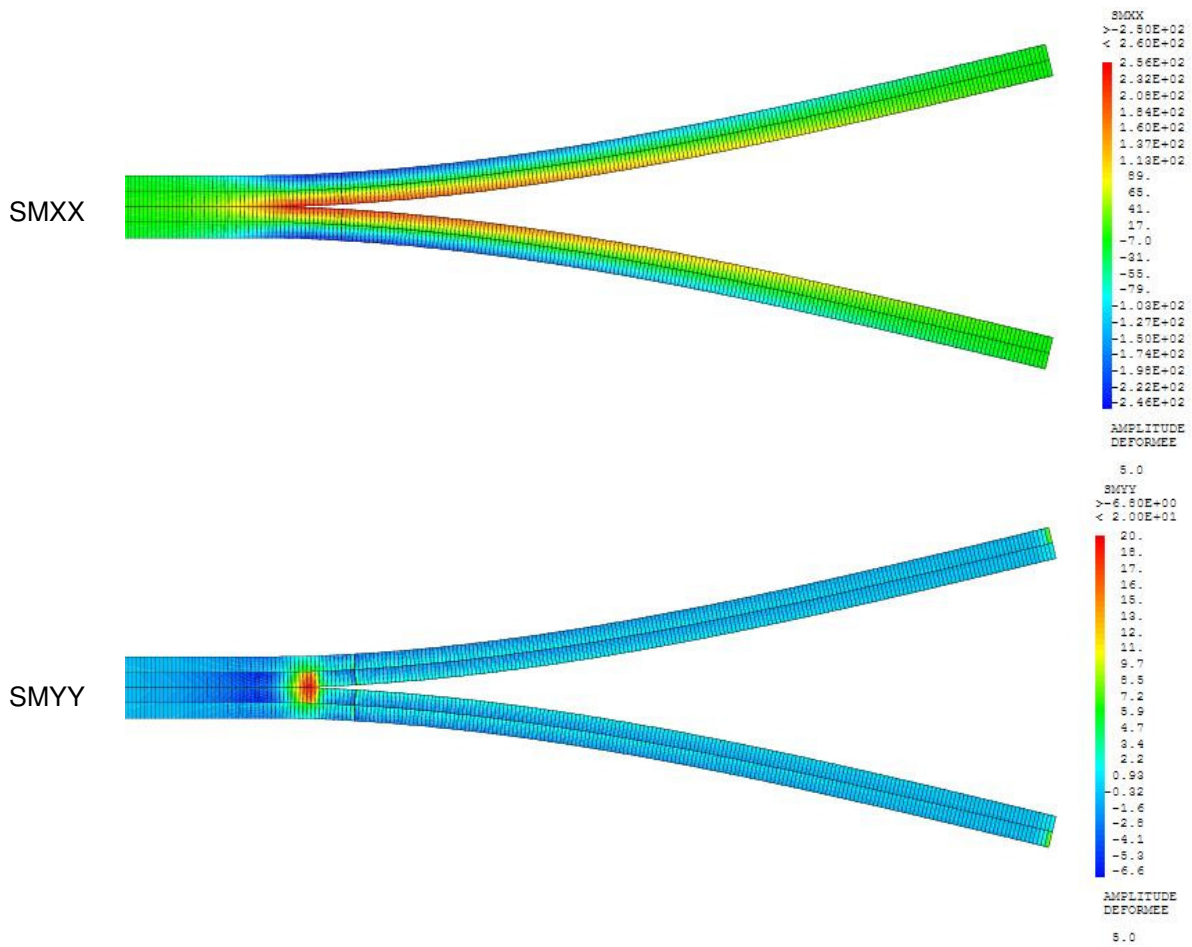


Figure 4.13 : Principal stress fields of the laminate arms for DCB test

As expected, the regions close to the inner edges of the laminate arms are in tension and the regions close to the outer edges are in compression (See SMXX stress field). The principal stress in Y-direction shows high stress concentration surrounding the crack tip. Note that, weaker objects in the FE model are interface elements. The deformation and stress distribution of the laminate arms are affected by the behaviour of these interface elements. Stress fields plots shown above justify the fact that continuity between solid elements and interface elements are maintained at all time.

Details of the convergence analysis for determination of the optimum mesh size are discussed hereunder. The investigation was performed on both 1D and 2D models.

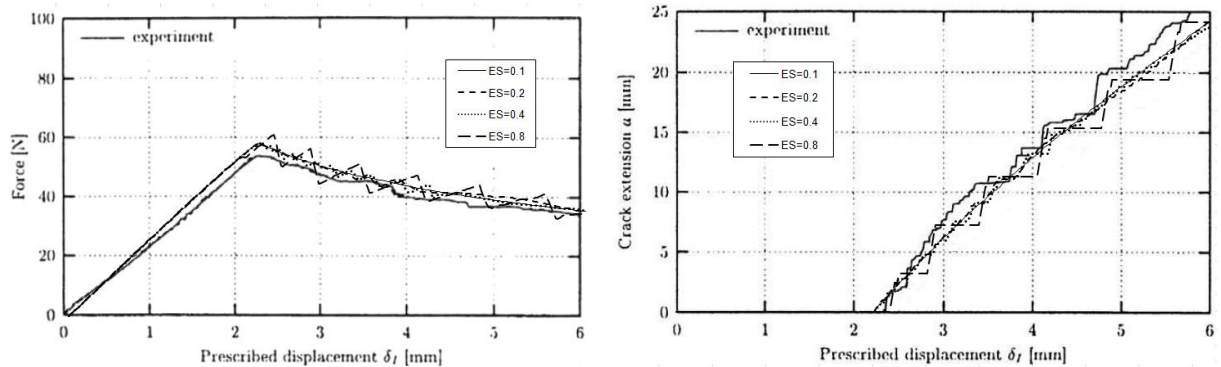


Figure 4.14 : Force & Crack length vs. Displacement for DCB – simulated results in 1D for different element sizes



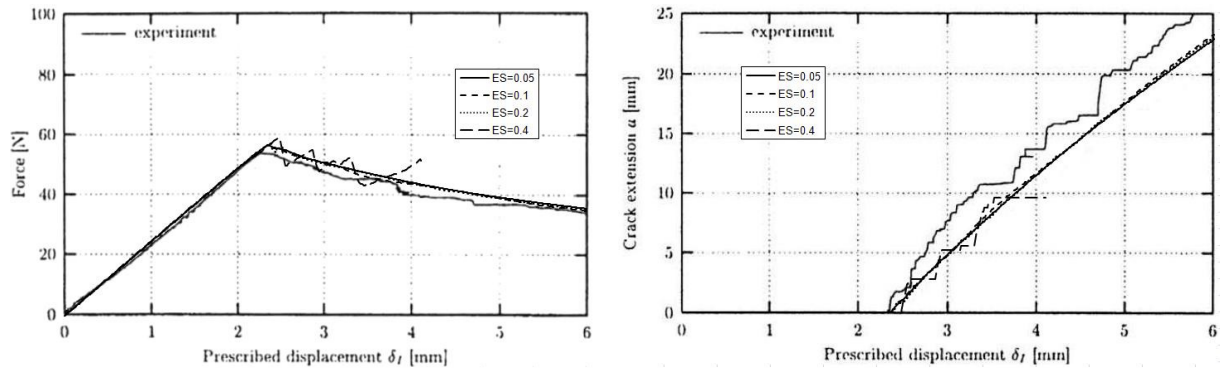


Figure 4.15 : Force & Crack length vs. Displacement for DCB – simulated results in 2D for different element sizes

Investigation results for 1D and 2D models are illustrated in Figure 4.14 and Figure 4.15 respectively. Note that, all simulations were done with a constant displacement increment, equal to  $0.04 [mm]$ . As it can be seen from each plot for both models, the convergence of the results were achieved for element sizes less than  $0.2 [mm]$ .

In case of 1D model, when the element size increases over the optimum value, undulation of the curve becomes more severe. Similar observations were also made for the 2D model. Interestingly in 2D, when the element size increases over  $0.2 [mm]$  the simulation tends to stop abruptly as a consequence of non-convergence. This may probably be associated to the severity of the erratic evolution of the variables. For example, the force-displacement curve for element size equals to  $0.4 [mm]$  shows a dramatic increase of the evolution and sudden collapse of the calculation process.

However, even at larger element sizes the behaviour of the model was observed to be intact. In other words, the predicted evolutions still follow the experimental results. Therefore, the damage model seems to be well reliable.

The results of the convergence analysis for determination of the optimum limits of ‘Force Tolerance’ and ‘Moment Tolerance’ are given in Figure 4.16.

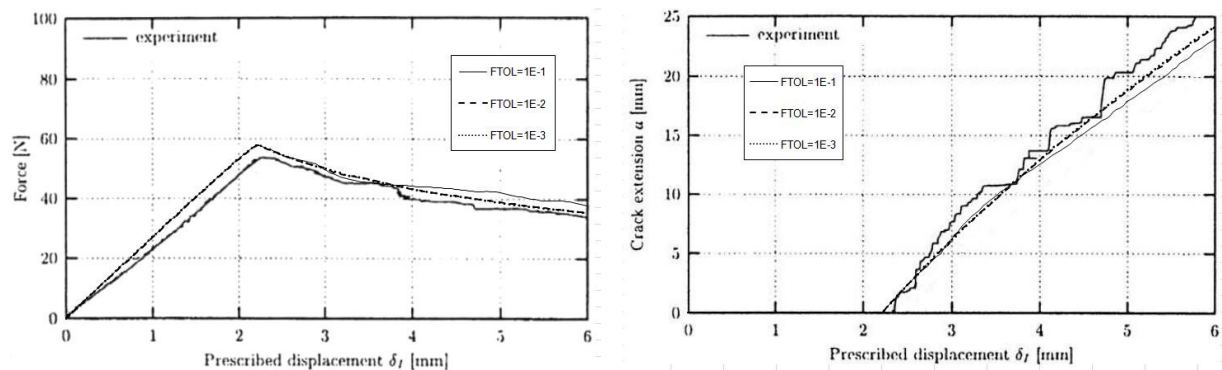


Figure 4.16 : Force & Crack length vs. Displacement for DCB – simulated results in 1D for different FTOL and MTOL values

The analysis was carried out considering the 1D model. Element size was selected equal to  $0.2 [mm]$ . Note that, for simplicity both FTOL and MTOL values were taken equal to each other.

According to plots given in above figure, FTOL (or MTOL) values smaller than  $1.0E-2$  produce converged results. Another convergence analysis was also performed considering the 2D model and similar conclusions can be made based on its results.

### 4.5.3 Simulation Results for 3ENF Test

Application of the boundary conditions and loading conditions (i.e., imposed displacement) for the 1D, 2D and 3D 3ENF test simulations are shown in Table 4.11.

1D	
2D	
3D	

Table 4.11 : Boundary and loading conditions for 3ENF static test

Note that, both 1D and 3D models were meshed with SHELL elements. Mid-planes of the shell elements for the two arms were located on either side at a distance equal to half the thickness of an arm. However, it's purely a virtual definition. Therefore, to define a compressive load on the shell mesh it then necessary to include an addition block (a line in 1D and a surface in 3D) at the location where the load it to be applied. The blocks were defined with same material properties used for the arms. Importantly, these entities (i.e., new mesh blocks) are only connected to the top laminate arm. Finally, the compressive loads were applied on each block as shown in above figures.

The optimum mesh size was once more identified by performing a convergence analysis. Interestingly, the optimum mesh size for 3ENF is larger than for DCB. For example, convergence of the numerical results was even achieved at element size equal to  $0.4 [mm]$ . However, to be consistent with DCB test conditions, element size equal to  $0.2 [mm]$  was selected for simulations.

Figure 4.17 given below shows a comparison of experimental and simulated results. Simulations were done with displacement increment equal to  $0.01 [mm]$ . Convergence limits were set to  $1.0E - 2$ .

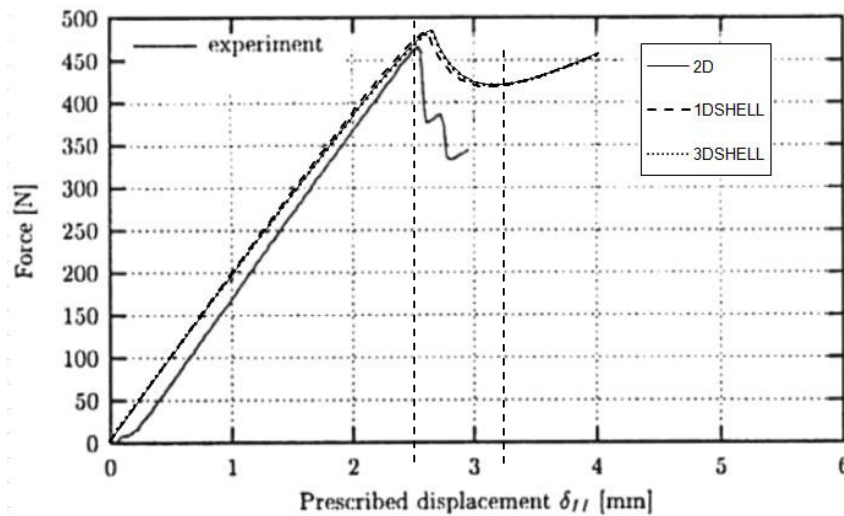


Figure 4.17 : Force vs. Displacement for 3ENF – simulated results for 1D, 2D and 3D models

Simulated results obtained for all 1D, 2D and 3D models seem to be acceptable in comparison to the experimental results. Interestingly, all 3ENF models have produced similar force-displacement evolutions. Note that, in DCB tests some difference was noted for the results obtained with 1D model.

In figure above, experimental curve shows a sudden dip in the beginning of the test. This uncharacteristic behaviour may probably be associated to some problematic issues encountered during testing. One possible reason is crushing of the laminate top surface during initial application of the loading. The static damage model does not consider this phenomenon and the laminate surface is considered to be a stiff body. The maximum value of the force was predicted within an acceptable range. Percentage errors of prediction for 1D, 2D and 3D are 2.8%, 2.8% and 3.1% respectively. In addition, a sudden drop in the experimental curve can be seen just after maximum force value. At this point, the laminates have been deformed roughly up to 2.5 [mm]. Since, the laminate arms are manufactured by bonding several unidirectional layers; it is more or less susceptible for compression failure at some point. It is probable that, when the interface starts to fail by shear, laminate arms also start failing by compression. In this particular case, the force value seems to reduce dramatically and the drop is almost vertical. Once more, the static damage model is not designed to take in to account the failure of the arms. Coupled failures, such as the one discussed here may pose many numerical difficulties (i.e., snap back phenomenon), and require much more rigorous investigation.

As expected, the simulation results shows a drop of the force with increasing displacement up to 3.0 [mm]. Note that, experimental results are not available beyond this value. However, simulation was continued further and an interesting phenomenon was observed. As the prescribed displacement goes beyond approximately 3.25 [mm], the global force starts to increase gradually. This phenomenon may probably be associated to the position of the crack tip relative to the load point position. This argument will be verified by investigating the evolution of the crack length.

Figure 4.18 given above illustrate the evolution of crack length obtained for each model. The initiation of the crack has been predicted satisfactorily by all models. As explained above, the propagation of the crack may not be accurately captured by the static model as a result of the probable coupled failure mechanism that has taken place during the test.

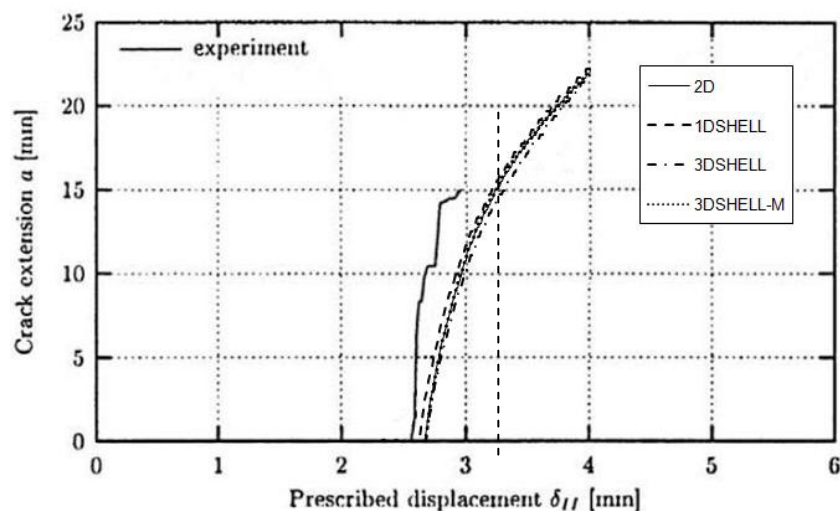


Figure 4.18 : Crack length vs. Displacement for 3ENF – simulated results for 1D, 2D and 3D models

As previously mentioned, the simulations were performed beyond the experimental range. When the prescribed displacement reaches 3.25 [mm] the crack extension increases up to 15 [mm]. Note that, the initial crack length is 35 [mm], therefore, total length of the crack is equal to 50 [mm]. This means that, at this instant the crack tip is directly underneath the loading point. As a result, the global force tends to increase with further increase of the prescribed displacement. At the same time shearing action also continues and the crack length will keep on growing gradually without showing any abrupt changes.

3D simulations performed on DCB tests revealed that crack front is not actually a straight line but a curved line. In other words, crack propagation at the center happens much earlier than at edges. It is interesting to note whether the same phenomenon is applicable for the Mode-II failures. Once more, the evolution of the crack length both at the center and the edges of the 3D model was determined. The corresponding curves are also included in Figure 4.20. Note that, curves named; 3DSHELL and 3DSHELL-M refers to the evolution of the crack length at the edges and center of the specimen respectively. Interestingly, evolutions of the two curves show only a small difference between them. This suggests that, in 3ENF test, crack propagation of the specimen tends to happen more uniformly throughout its width.

Figure 4.19 given below includes shear stress field plots close to the crack front and they correspond to two different loading stages of the test. The understanding of the state of stress at the interface will be useful for determining the shape of the crack front as described earlier with DCB tests.

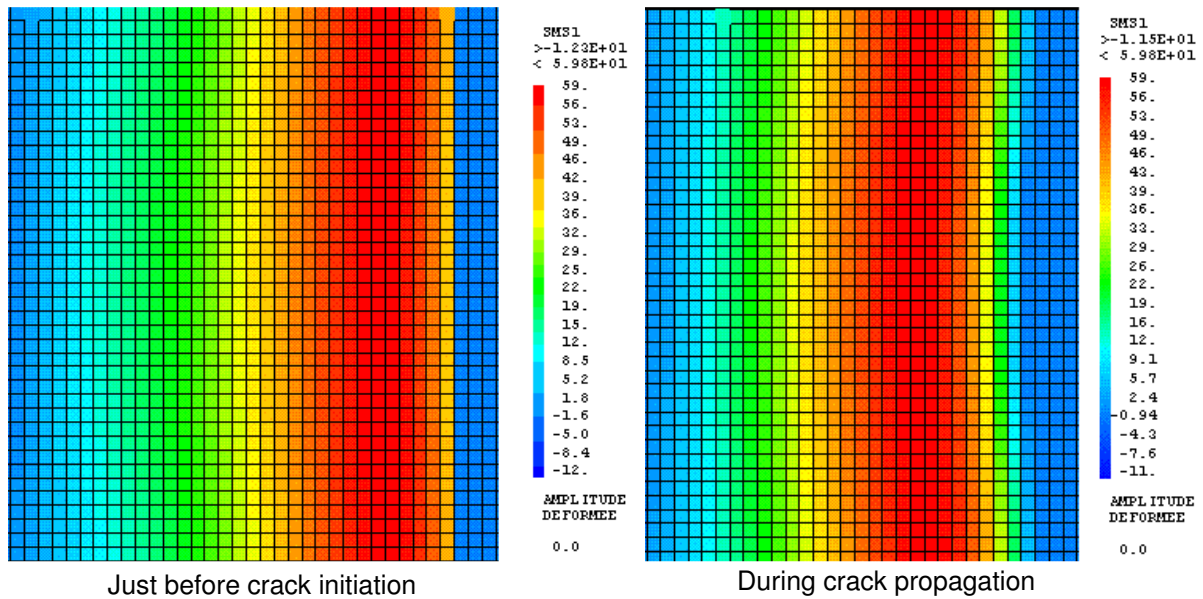


Figure 4.19 : Shear stress field at the crack front for 3ENF test

The stress fields shown above have been produced using shear stress component (i.e., SMS1) in the X-direction. Since, 3ENF test is specifically designed to obtain Mode-II or Shear Mode failure, the other stress components are found to be negligible.

Prior to crack initiation, the shear stress distribution seems to be more or less uniform throughout the specimen's width. However, stress intensity of the elements close to the edges is slightly smaller than elements at the center. After crack initiation, stress distribution changes slightly and remains roughly unchanged during the propagation stage. Similar observations were also made with DCB tests. Once again, a curved shape stress distribution can be seen ahead of the crack front. However, only small variations of the stress intensity can be seen for elements that are located on the same vertical line (Note that, for DCB tests, the variation of the stress intensity is found to be much more significant). In addition, evolutions of the damage variables for those elements are found to be similar to each other. As a result, evolution of the crack length across the specimen's width is found to be roughly uniform. However, it would be interesting to see if this phenomenon is true for other Mode-II tests, including 4ENF (four-point End Notch Flexure), ELS (End Load Split), etc.

The stress distribution plots also reveal the size of the process zone ahead of the crack front. By comparing the two stress fields given Figure 4.12 and Figure 4.19, one can appreciate the extent of the damage zone for each failure mode. Comparatively, the damaged zone in Mode-II is seen to be larger than the damage zone in Mode-I. This difference could probably be associated to the nature of the loading condition.

Similarly, as in the case of DCB tests, a convergence analysis was performed to determine optimum mesh size for 3ENF tests. The investigation was performed on both 1D and 2D models.

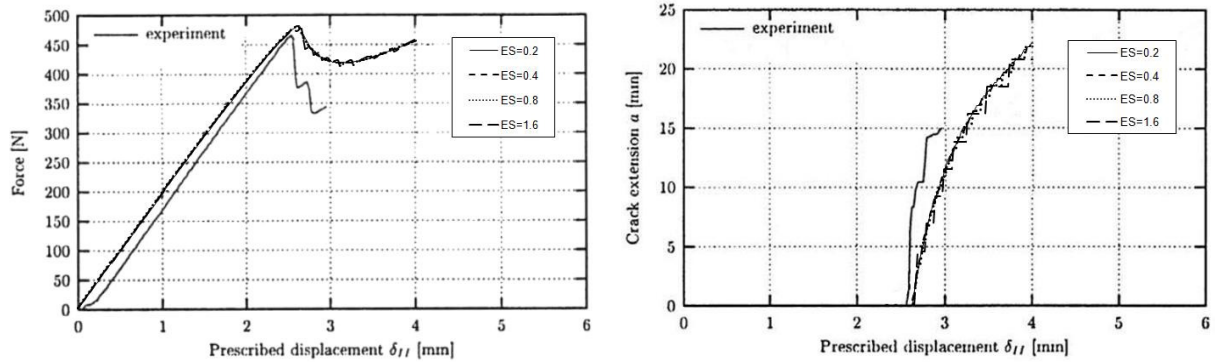


Figure 4.20 : Force & Crack length vs. Displacement for 3ENF – simulated results in 1D for different element sizes

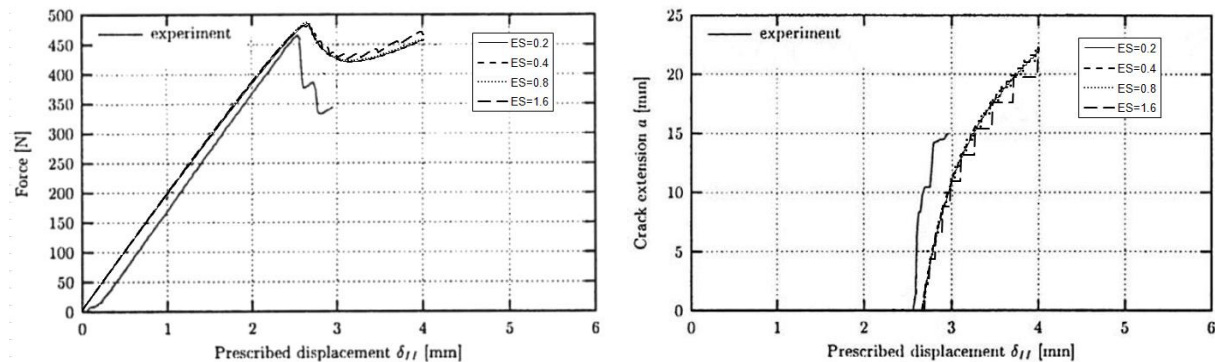


Figure 4.21 : Force & Crack length vs. Displacement for 3ENF – simulated results in 2D for different element sizes

Investigation results for 1D and 2D models are illustrated in Figure 4.20 and Figure 4.21 respectively. Note that, all simulations were done with a constant displacement increment, equal to  $0.02 [mm]$ . Interestingly, the convergence of the results were even achieved with element sizes equal to  $0.4 [mm]$ . Note that for DCB, the optimum mesh size was found to be  $0.2 [mm]$ . For element sizes over the optimum limit the undulations of the curve becomes more and more prominent, but the prediction of the general behaviour was found to be satisfactorily.

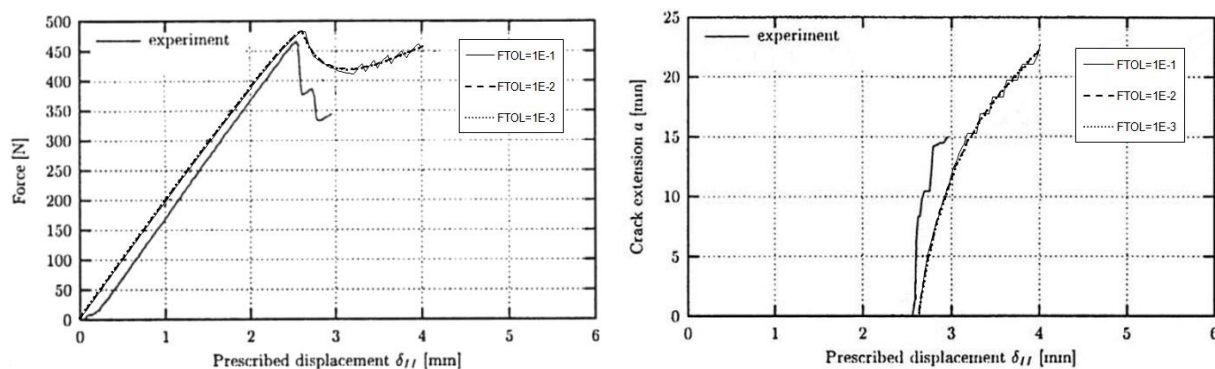


Figure 4.22 : Force & Crack length vs. Displacement for 3ENF – simulated results in 2D for different FTOL and MTOL values

Figure 4.22 given above illustrate the results of the convergence analysis performed for determination of optimum FTOL and MTOL limits. The analysis was performed considering 1D model with an element size equal to  $0.2 [mm]$ . Similarly, to DCB tests, the optimum FTOL and MTOL limits were found to be equal to  $1.0E-2$ . Convergence analysis performed on the 2D model also produced similar results.

## 4.6 Validation of Nonlocal Static Damage Model

### 4.6.1 FE Implementation of Nonlocal Model

The nonlocal damage model is basically formulated by introducing some intermediate steps in the finite element implementation of the local model. Note that, in the present study an integral-type nonlocal damage evolution law has been considered for implementation. Table 4.12 given below includes the two different approaches that were considered for simulations. As it was discussed earlier in Section 3.3.2, the regularization of the computed solution can easily be achieved by either directly averaging the damage variables or averaging the damage energy release rate variables.

	Averaging damage energy release rate	Averaging damage variable
Step-1	Compute damage energy release rate variables, i.e., $\bar{Y}(t)$ using $Y_{d_1}, Y_{d_2}$ & $Y_{d_3}$	Compute damage energy release rate variables, i.e., $\bar{Y}(t)$ using $Y_{d_1}, Y_{d_2}$ & $Y_{d_3}$
Step-2	Average damage energy release rate variables, i.e., $\bar{Y} \rightarrow \bar{\bar{Y}}$ using Eq. 3.14 & 3.15	Compute isotropic damage function, i.e., $w(\bar{Y})$ using $\bar{Y}, Y_C$ & $Y_0$
Step-3	Compute isotropic damage function, i.e., $w(\bar{\bar{Y}})$ using $\bar{\bar{Y}}, Y_C$ & $Y_0$	Compute damage variables, i.e., $d_1 = d_2 = d_3 = w(\bar{Y})$
Step-4	Compute damage variables, i.e., $\bar{d}_1 = \bar{d}_2 = \bar{d}_3 = w(\bar{\bar{Y}})$	Average damage variables i.e., $d \rightarrow \bar{d}$ using Eq. 3.14 & 3.15
Step-5	Compute stresses for the interface, $\sigma_{13} = k_1^0(1 - \bar{d}_1)U_1,$ i.e., $\sigma_{23} = k_2^0(1 - \bar{d}_2)U_2,$ $\sigma_{33} = k_3^0(1 - \bar{d}_3)U_3$	Compute stresses for the interface, $\sigma_{13} = k_1^0(1 - \bar{d}_1)U_1,$ i.e., $\sigma_{23} = k_2^0(1 - \bar{d}_2)U_2,$ $\sigma_{33} = k_3^0(1 - \bar{d}_3)U_3$

Table 4.12 : FE implementations of nonlocal static damage models

Note that, all simulations were performed in 2D considering the same DCB and 3ENF test cases. Therefore, all material properties and values of damage model parameters are same as given before. In each approach, averaging of the variables needs to be performed at each gauss point of the all the interface elements. Usually, averaging of the variables is performed starting from the second increment. If the increment size is sufficiently small, the effect of this delay is believed to have a negligible influence on the final result.

In CAST3M, averaging of the variables was achieved using a personal procedure; named PERSO1, available in the PASAPAS operator. As explained before, the modified damage model is once again introduced in to CAST3M using the UMAT sub-routine. Initially, for the first increment, values of all the STATE VARIABLES (damage variables, damage energy release rate variables, stress, strain, etc) will be computed at every gauss point in each element. When the convergence is achieved, the PASAPAS PROCEDURE will then launch the PERSO1 sub-procedure. At first, the procedure will call for the STATE VARIABLE data saved in the 'VARI' (i.e., initial internal variables) table. Each variable will then be saved in different tables for subsequent processing. Next, X and Y coordinates of all the gauss points (or nodal points) are also obtained and saved in two different tables. As explained in Section 3.3.2, a Gaussian distribution is used for the averaging of the selected variable (damage variable or damage energy release rate variable). The coordinates along with an internal length scale (i.e.,  $l$ ) is then used to compute the weight of each variable value for every node in the interface.

Figure 4.23 given hereunder illustrates the effect of the internal length scale on the Gaussian distribution. Note that, the selected Gaussian function has the form given in Equation 3.15.

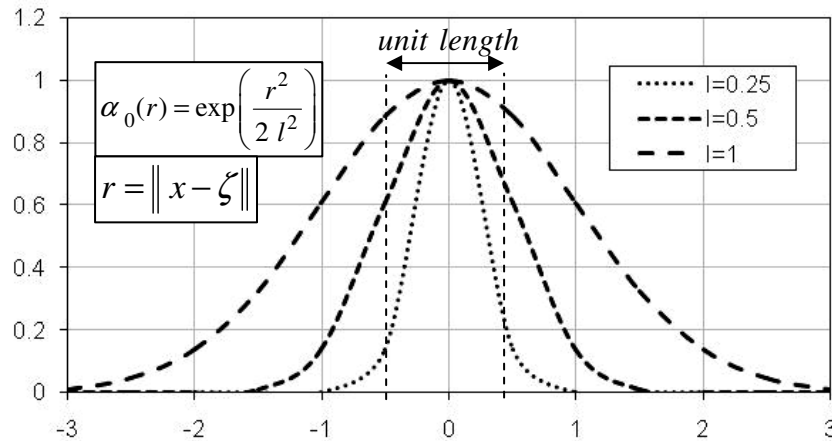


Figure 4.23 : Effect of internal length scale on weighing function

The Gaussian function is also referred as the weighing function, because it is used to determine contributions of variable values of every gauss point. For example, let us consider the variable value located at the origin. In order to find the nonlocal value at this point, the variable values located at the other points will be taken in to account. However, the level of contribution of those variable values will depend on the distance between the point of interest and the point from where the contribution is coming from. Contributions are simply computed by multiplying the weight at a given point with its variable value. In general, as the distance increase, the contribution tends to decrease further and further. In the next step, all the contributions and weights considered will be added together separately. The nonlocal value at the point of interest will then be computed by taking the ratio of total contributions with their weights.

As it can be seen from the above figure, the internal length scale is typically used to modify the shape of the weighing function. For higher values of  $l$ , bell shape tends to expand further. As a result, the contribution coming from each variable value will also increase. In addition, the domain of the points which are responsible for making significant contributions would also increase. If we consider the bell plots given above, the plots made with the higher length scales cover more points compared to the smaller ones.

In order to completely understand the behaviour of the nonlocal model, it is necessary to investigate the relation between element size and the internal length scale. Corresponding to the figure given above, let us consider a mesh with elements of unit length. When internal length scale is taken equal the element length (i.e.,  $l = 1$ ), we can clear observe some major contributions coming from many points in the neighborhood of the interested point (i.e., origin). When the internal length scale is taken half the size of the element length (i.e.,  $l = 0.5$ ), a considerable drop of both level of the contribution and the number of significant contributions can be seen. However, some notable contributions are still coming from some of the points that are located outside the zone (see figure) of interest. When we reduce the internal length scale further (i.e.,  $l = 0.25$ ), we clearly see no significant contribution coming from any points that are located outside. At this point, the nonlocal model starts to behave as a local model. Therefore, the nonlocal model is now incapable of regularizing the spurious localizations. In conclusion, the selection of the internal length scale is mainly connected to the element size of the FE model. In the present study, a good value for the internal length scale is determined by fitting the simulated data with experimental result.

One of the major limitations of the presented model is its high computational cost. Note that, averaging of the variables is performed at each gauss point of all the elements. As the number of elements increase, the number of variables that is to be averaged would also increase. In addition, averaging of variables at each point is normally performed by considering all the points in the interface. When the number points increase, the number of points that is to be considered for the averaging process will automatically increase. In fact, this is the main reason which is responsible for the notable increase of the computational time. In order to overcome this problem, an innovative but simple strategy was introduced.

As it was discussed earlier, only some points are providing major contributions for the averaging process. Figure 4.24 given below can be used to describe the magnitude of contributions coming from both left and right side of the point in interest.

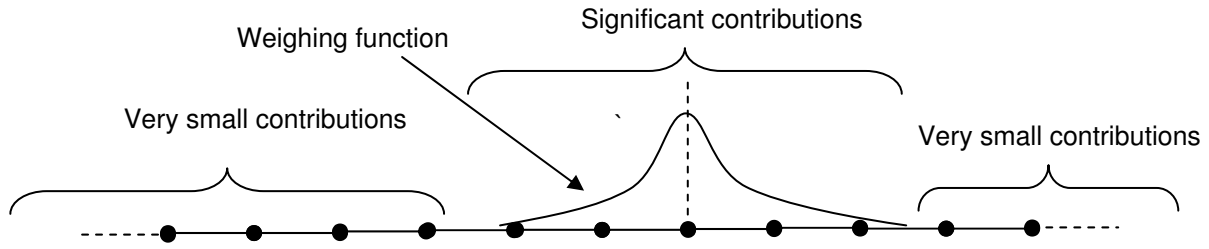


Figure 4.24 : Level of contributions received from left and right

As we move along in either direction, the magnitude of the weights determined by the weighing function will decrease gradually. With increasing distance, the values of the weights will decrease towards zero. As a result, contributions corresponding to those weights would automatically become very small. Therefore, averaging of the variables needs only be done with points that are in the neighbourhood. In fact, when we do averaging considering all the points in the whole domain, we perform unnecessary amount of computations without any useful gain. The idea was to eliminate redundant computations as much as possible. At the same care was taken to ensure the integrity of the nonlocal model.

Next, a criterion was introduced to estimate the number of points that should be considered for the averaging process. In accordance to Figure 4.25, the criterion should also necessarily consider the effect of the internal length scale. Thus, to reduce complexity a simple approach was adopted to identify the size of the neighbourhood points. Figure 4.25 given below illustrate the selected criterion in a graphical form.

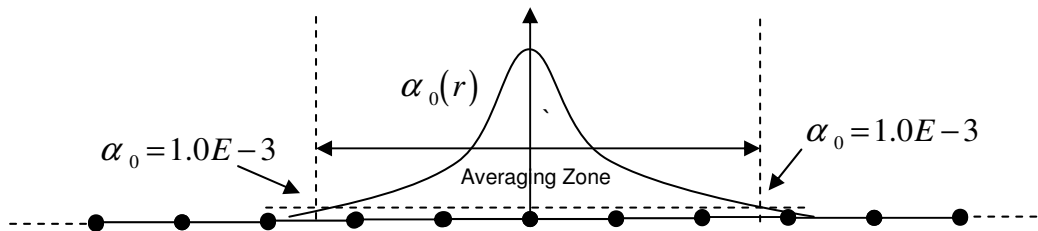


Figure 4.25 : Criterion for selecting neighbourhood points for averaging

As shown above, the points that are lying in between the two limits corresponding to  $\alpha_0 = 1.0E - 3$  are selected for the averaging process. The neighbourhood of these points is referred as the 'Averaging Zone'. These limits have been identified by performing a simple investigation between the original model and the modified model. Note that, the limits are inherently related to the problem in hand. In other words geometry of the model, mesh size, evolution of the variables, etc needs to be closely analyzed before making a decision. For test-cases where influence of parameters is not well understood, a limit corresponding to  $\alpha_0 = 1.0E - 6$  should at least be considered.

In addition, few other attempts were made to further optimize the computation process. Generally (except at the two ends of the domain), at each nodal point there are two gauss points belonging to the two elements connected to it. The evolution of the STATE VARIABLES at these two gauss points will essentially be equal to each other at every stage of the solution process. This is mainly because of the continuity relation that exists between elements connected to each other. However, when it comes to averaging of the variables, it is not necessary to perform the same computation twice for the two gauss points located at the same node. We only need to perform a single computation and then store the averaged values for the two gauss points separately. By adopting this method we can effectively reduce the total number of computations required for averaging by half. Therefore, it was possible to reduce the computational time significantly. Interestingly, a notable reduction of the computation cost was observed with all proposed modifications. The ratio of the computational time between the original and the modified model is in the range between 4 and 6.



Afterwards, the averaged variables need to be defined at each gauss point systematically. Note that, during the averaging process averaged values were written in separate tables. Now, all the STATE VARIABLES need to be redefined in the 'VARF' (i.e., final internal variables) table in PASAPAS PROCEDURE. In CAST3M, the user is only limited to define a common value over an element. The following choices were considered for determining a common or a representative value for each element using the values of the averaged variables at the two gauss points.

- I. Taking the average of the variable values at the two gauss points
- II. Taking the maximum value of the variable values at the two gauss points

Both approaches were tested methodically and comparisons were made for subsequent analysis. Results of the investigation are included in the next section. Interestingly, the two approaches present two different conceptual arguments. In CASE-I, when you do averaging on the already averaged variables, this new value will make the model more nonlocal in comparison. On the other hand, in CASE-II, when you assign the maximum value of the two averaged values, comparatively you are making the model more local.

However, when we perform further averaging on the already averaged values, we ultimately delay the evolution of the variables. As a consequence, the computational time tend to increase significantly. On the other hand, excessive averaging would also hinder accuracy of the computed value. It adds more viscosity in to the models behaviour. As a result, the STATE VARIABLES take more time to converge on to the realistic values. In comparison, taking the maximum value would help to reduce the computational time effectively. However, the lost information would obviously have a negative impact on the model's outcome. The details of the investigation comparing the two approaches are included in a later section. In general, the second approach seems to produce better results compared to the first one.

Efforts were also made to further expand the nonlocal model from 2D to 3D environment. In fact, spurious localization issues are more associated to 3D models in comparison to 2D models. As it was observed during the study, 2D FE models with uniform mesh do not give rise to any numerical localization. In fact, models with varying mesh sizes also produced localization free results. In 3D, the Gaussian function is no longer line but a three-dimensional surface having a 'bell shape'. As a result, in three-dimensions the computational time increases heavily and the investigation of the model's behaviour becomes more inflexible. Therefore, as mentioned earlier, all investigations of the present study are performed on 2D models.

#### 4.6.2 Simulation Results on DCB and 3ENF Tests

According to preliminary analysis, the approach based on 'averaging of damage energy release rate' was found to be more favourable in predicting the experimental results. Therefore, much attention was devoted to appreciate the behaviour of that model. Investigations was carried out on both DCB and 3ENF test models. Classically, for any given nonlocal model, the convergence of model's response should be verified. Experimental results used for verification of the local model will once again be used to validate the proposed nonlocal model. Note that, a choice should be made for selecting a representative value for each element using the averaged values. Simulation results given hereunder are based on the approach of taking the maximum value of the two gauss points belonging to each element.

At first, an investigation was carried out to identify a suitable value for the internal length scale. The results of this investigation is included in later section. The value for the internal length scale selected for the simulations equals to 0.25. Efforts were also made to identify correlations between the internal length scale and model parameters (i.e., dimensions of the geometry, element size, etc). The size of the averaging zone selected for the following simulations corresponds to the range of points enveloped by  $\alpha_0 = 5.0E - 3$ . For simplicity, a linear correlation was developed between the internal length scale and number nodes that should be considered for averaging (i.e, NAN). In addition, this relation is inherently depended on the selected element size. For example, for an element size of  $ES = 0.1$  the corresponding relation reads as,  $NAN = 32l + 1$ . Note that, in the developed program code the user needs to enter the value of NAN along with the value of internal length scale.

Given hereunder are the results and conclusions made on the convergence analysis of the proposed nonlocal model. Figure 4.26 given below, illustrate the evolution of the force against imposed displacement for different element sizes. Here, for both DCB and 3ENF tests, increment size and FTOL value were taken equal to  $0.01 [mm]$  and  $1.0E - 2$  respectively.

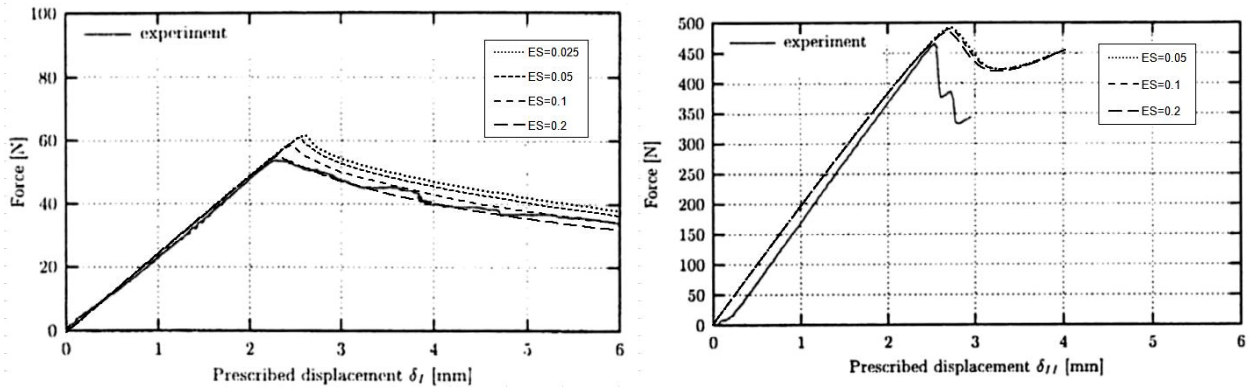


Figure 4.26 : Influence of element size on nonlocal model's behaviour for DCB and 3ENF tests

In addition, same simulations were performed with different increment sizes and similar observations were made on the behaviour of the evolutions. As it can be seen from each plot, with decreasing element size the force-displacement evolutions seems to converge towards each other. This is clearly visible with the DCB tests simulations, where as for 3ENF the convergence seems to have been already achieved with a larger element size (i.e.,  $ES = 0.2$ ). In case of DCB, the convergence is achieved roughly at an element size of  $ES = 0.05$ .

Interestingly, we can note that evolutions are moving away from the experimental result (especially in the case of DCB, in fact nonlocal results on 3ENF are very close to the local results). This observation automatically prompts a question on the influence of the increment size on the nonlocal model's behaviour. Therefore, a separate analysis was performed to understand the effect of increment size and the results of this analysis is shown in Figure 4.27 given below.

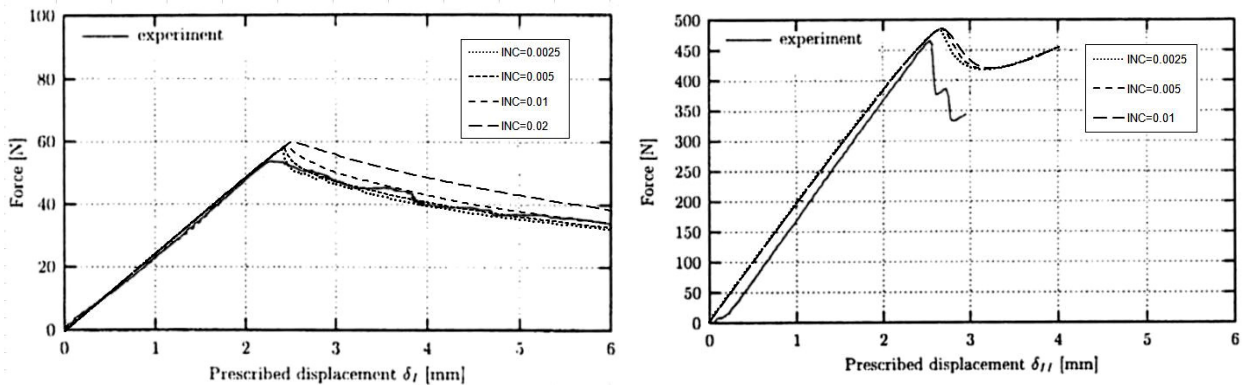


Figure 4.27 : Influence of increment size on nonlocal model's behaviour for DCB and 3ENF tests

Here, the simulations were performed with an element size equal to  $ES = 0.1$  for both tests. FTOL value was kept unchanged. Similarly, as with the case of element size, convergence is normally achieved with decreasing increment size. As it can be seen from the above plots, this is true for both DCB and 3ENF test results. Interestingly, force-displacement evolutions are moving towards the experimental results. Once more, as expected, this phenomenon is more visible for the DCB test case than for 3ENF test case. In case of DCB, the convergence is achieved roughly at an increment size of  $INC = 0.01$ . By considering the facts of the two analysis discussed earlier, it is now possible to perform the final simulations to obtain the converged solutions. Therefore, for the DCB test, the selected element and increment sizes are  $ES = 0.05$  and  $INC = 0.005$ . Similar values were also selected to simulate the 3ENF test.

Plots included in the Figure 4.28 given hereunder, shows the converged results of both local and nonlocal models for DCB test.

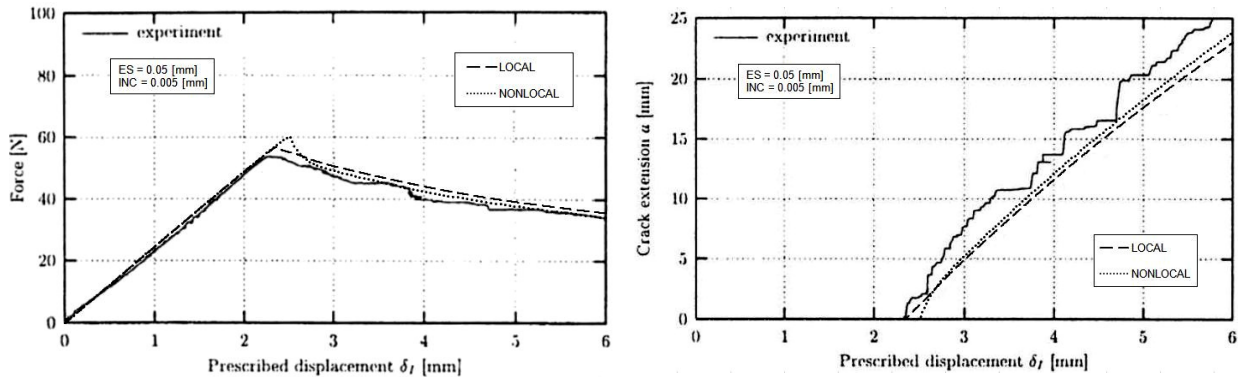


Figure 4.28 : Comparison of local and nonlocal results for DCB test

As it can be seen from force- and crack extension-displacement plots, the results produced by both local and nonlocal models are in good agreement with the experimental results. However, the nonlocal model seems to over predict the maximum value of the force. At the same time, prediction of the initiation of the crack happens with a little delay. After the force has reached its maximum value, with increasing displacement the curve starts drop steeply. This is reflected on the crack extension plot with a sudden increase of the crack length. When displacement is further increased, force evolution tend to settle down gradually and seem to follow the experimental result with good accuracy. Same is true for evolution of the crack length. The reason for the nonphysical increase of the force is related to the numerical viscosity induced by the nonlocal model. This is considered to be an inherent problem associated to integral-type regularization schemes.

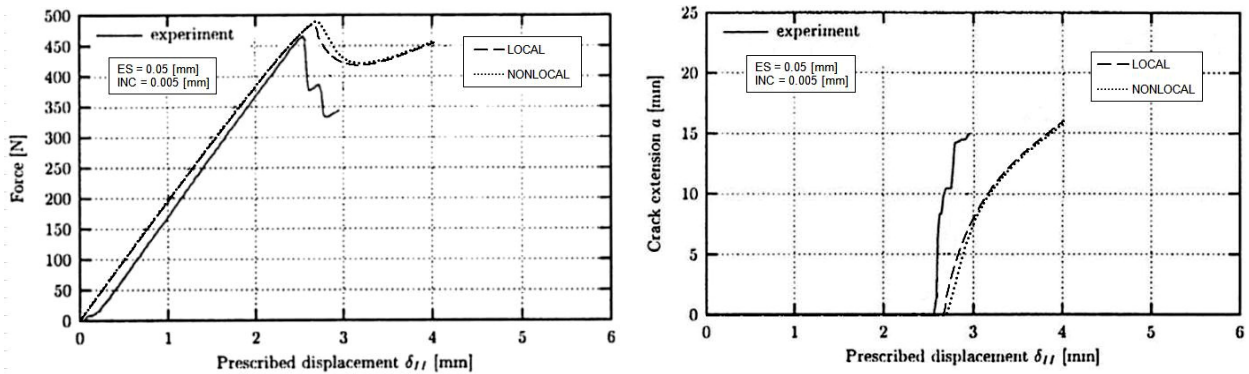


Figure 4.29 : Comparison of local and nonlocal results for 3ENF test

Figure 4.29 given above shows a similar comparison made between 3ENF test results obtained from local and nonlocal models. As explained before, nonlocal model's behaviour is closer to local model. Note that, convergence of the nonlocal results were even achieved with larger element/increment values. Interestingly, the local model also behaved in a similar manner and in comparison to DCB test simulations convergence was also achieved with larger element sizes.

### 4.6.3 Influence of the Internal Length Scale

In order to identify a suitable candidate for the internal length scale, a simple analysis was performed by comparing model's response for few arbitrary values. However, initial selection of these arbitrary values were done in relation to the element size of the FE model. As it was discussed earlier, the internal length scale should not be too small compared to the element size. If it goes below the critical value, then the nonlocal model will no longer be capable of regularizing the evolutions of the variables. If the value is too large, then we will be introducing more viscosity in to the model's behaviour. As a consequence, the maximum force value will increase unrealistically. Therefore, it is necessary to consider the element size to make a good prediction for the internal length scale.

Figure 4.30, give below can be used to visualize the effect of internal length scale (i.e., ILS) on both DCB and 3ENF test cases.

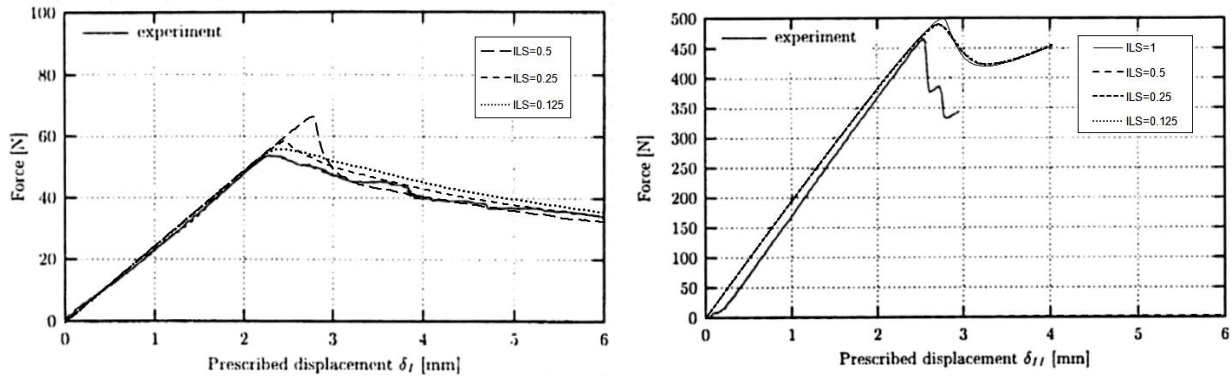


Figure 4.30 : Influence of internal length scale on nonlocal model's behaviour for DCB and 3ENF test

Note that, these simulations were performed with meshes having element sizes equal to  $ES = 0.1$ . In addition, for both tests increment size was taken equal to  $INC = 0.01$ . The internal length scale values selected range from  $l = 0.125$  to  $1$ . As it can be seen with the DCB plots, when the internal length scale is close to element size the model starts to behave inherently as a local model. As a consequence, evolution of force obtained with the nonlocal model is very similar to the result of the local model. On the other hand, when the internal length scale is five times larger to the element size, we can see a considerable increase of the maximum force value. In order to ascertain the effect of increment size on these observations, similar analysis was performed with larger increment size (i.e.,  $INC = 0.02$ ). As we have seen before, resulting evolutions shifts upward almost in parallel to this existing ones. However, for 3ENF test simulations these variation are very small. Only at an internal length scale of  $l = 1$  we can see a considerable deviation. The reason for this behaviour may probably associated to the relatively larger process zone in front of the crack tip.

#### 4.6.4 Comparison of different Modelling

Figure 4.31 given below shows a comparison made between the two approaches of selecting either a maximum value or an averaged value as the representative value for each element.

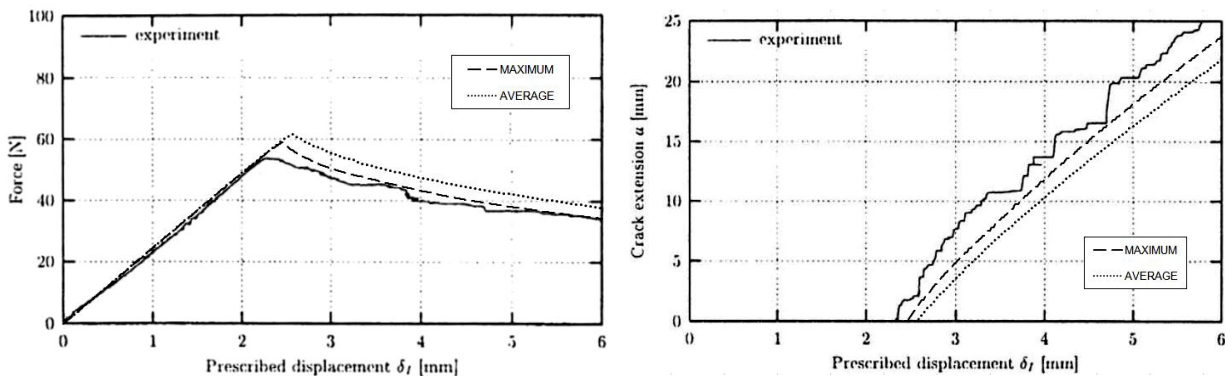


Figure 4.31 : Influence of taking average or maximum as a common value over an element

Here, simulation results corresponds to element size and increment size equal to  $ES = 0.1$  and  $INC = 0.01$  respectively. As it is can be seen from both plots, the results obtained with the averaging approach seems to deviate further away from the experimental results. Therefore, we see an increase of the maximum force value and also an increase of the delay of the crack initiation. This observation is inline with the comment that was made earlier regarding the use of larger internal length scale values for averaging. The averaging process induce numerical viscosity causing a resistance to the natural evolution of the variables. So when we perform averaging over already averaged variables the level of resistance also increase further. However, for the case of internal length scale the added

resistance seems to settle down gradually with increasing crack length and the phenomenon tends to balance out internally. On the other hand, averaging over the element is an external process. As a result, the corresponding evolution remains parallel and over the evolution of the second approach (which was obtained by taking the maximum value).

Figure 4.32 given hereunder are the results of an investigation performed to justify that the proposed modification to the nonlocal model does not effect the outcome of the original model's response. The main objective of the proposed modifications was to reduce computational time effectively.

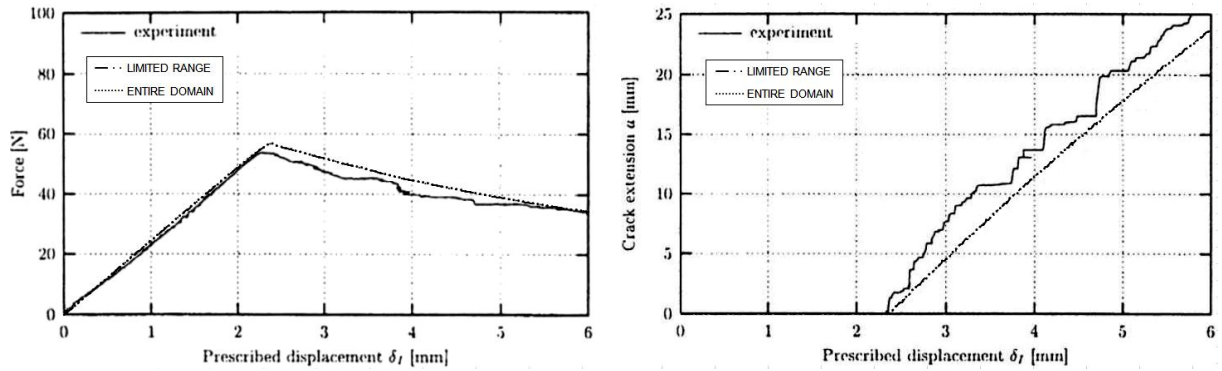


Figure 4.32 : Model responses on averaging over entire domain and averaging over a limited range

Selected element size and increment size for analysis are  $ES = 0.2$  and  $INC = 0.02$  respectively. As it is evident from each plot, both force and crack length evolutions are almost identical. In other words, averaging over the entire domain or over a limited range (i.e., averaging-zone as seen on Figure 4.25) yield same results. Computation times were noted for both simulations. Here, the tests were done on the same computer and care was taken to impose same testing conditions. Finally, the ratio of the simulation time between averaging over the whole domain and averaging over a limited range was found to approximately equal to 6. Therefore, the time saved by the proposed modifications is fairly significant. In case of 3D modelling, these modifications may prove to be very useful.

### 4.6.5 Comparison of Evolution of Damage Variable

In order to appreciate the behaviour of each model it would be useful to compare the evolution of damage variable at different gauss points along the interface starting from the crack tip.

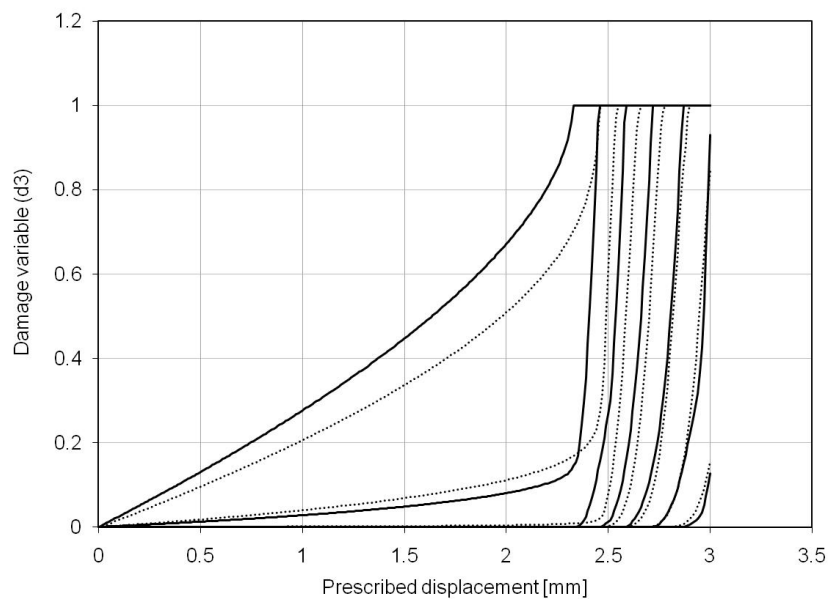


Figure 4.33 : Evolutions of damage variables for local and nonlocal models

Figure 4.33 given above shows the variations of damage variables associated to the gauss points in the vicinity of the crack tip for the DCB test. Note that, element size and increment size used for simulations are  $ES = 0.1$  and  $INC = 0.01$  respectively. The first continuous line and the dotted line starting from the left correspond to the the evolutions of the damage variables at the crack tip for local and nonlocal models respectively. The other curves correspond to the evolution of damage variable at gauss points located at  $1[mm]$  away from each other one after the other.

Interestingly, for both models the evolution of damage variable at the crack tip starts to take place gradually and with increasing displacement the evolution rate increase rapidly. This behaviour becomes more severe as we move along the interface from the crack tip. For the case of local model, when the first element gets debonded, undamaged gauss points located futher inside the interface also begins to get damaged. Interestingly, we can see a sudden increase of the evolution of damage variables at these gauss points. This behaviour then seems to continue in a similar fashion with increasing crack length. Note that, the nonlocal results correpond to an intenal length scale of  $0.25$  . The main difference between the two evolutions are the starting point of the damage for gauss points that are located internally. From the plots we can clearly see that the starting of the damage for internal gauss points also happens at the begining of the test for the nonlocal model.

#### 4.6.6 Size Effect Prediction using Nonlocal Model

In solid mechanics, the scaling problem of main interest is the effect of size of structure on its strength [Zdeněk P. Bažant, 2005]. It is well known that there is a tendency for the strength of laminated composite materials to decrease with increasing specimen size. This is the so called ‘size effect’ phenomenon. Note that, at laboratory we determine the design allowables of a given composite using small test coupons or specimens. Therefore, when it comes to design of large composite structures one needs to take account the size effect correctly. For a long time, size effect was explained by randomness of material properties. However, the probabilistic aspect in statistical approaches does not introduce an internal length scale that is necessary to reproduce experimental measurements [Zdeněk P. Bažant, 2005; Frédéric D., 2007]. Last two decades, much attention was given on deveoping nonlinear fracture models and nonlocal damage models [Zdeněk P. Bažant et al., 1996] to predict effect of structure size on the nominal strength. Given hereunder are the results of an investigation performed to predict scaling effect using the the proposed nonlocal model.

The DCB test case was considered for the analysis. Simulations were performed on three similar geometries with different sizes. The scaling of the geometries were achieved by multiplying each dimension with the coefficients;  $0.5$ ,  $1$  and  $2$ . The height of the laminate arm will be taken as the representative dimension for each scale. Once more, the internal length scale was selected equal to  $0.25$  . Element size and increment size were taken equal to  $ES = 0.1$  and  $INC = 0.01$  respectively.

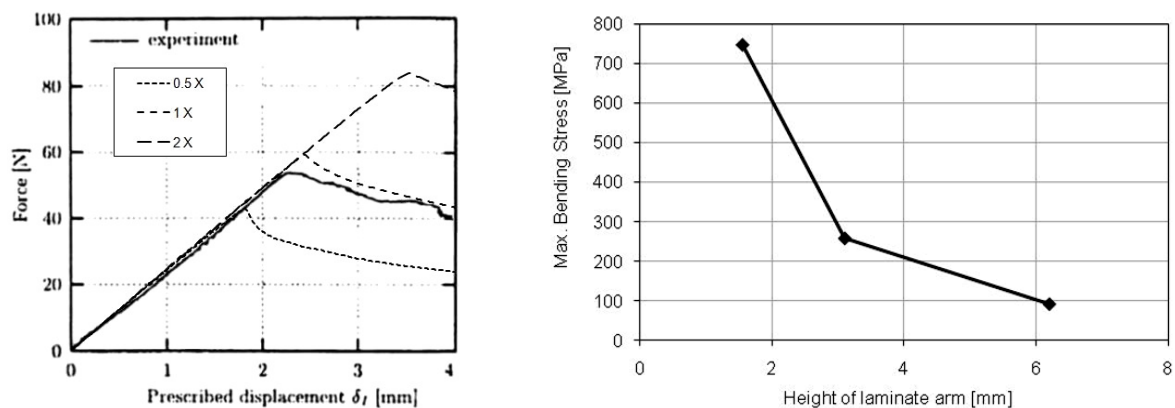


Figure 4.34 : Predictions made on size effect using nonlocal model

Figure 4.34 given above shows an increase of the maximum force value with increasing scale. On the other hand, the maximum bending stress (obatined by simulations) tend to decrease with increasing size. Therefore, the nonlocal model seems to capture the size effect as expected.

## **PART – III**

### **Delamination under Fatigue Loading**

Section comprises two chapters, Chapter 5 and Chapter 6.

Chapter 5 starts with an overview of fatigue and related theories associated to metal and composite laminates. Then after, includes an introduction to a new fatigue damage evolution law, its derivation and implementation. The existing local model was used as a platform to build the new fatigue model.

Chapter 6 includes details of FE simulations performed to validate the proposed fatigue damage model. Versatility of the fatigue model was also checked for different mode-ratios. Procedure of identification of model parameters, simulation results and accompanying conclusions are also detailed out completely.

## Chapter 5

### Fatigue Damage Evolution

#### 5.1 Overview of Fatigue and related Theories

The understanding of the phenomenon of 'fatigue' is crucial for designing structures made of any given material (i.e., single-phase or composite). The 'fatigue life' is basically responsible for the durability of the structure and hence considered as the most important design criterion. Fatigue is a state in which a material fails under repetitive loading generating stresses lower than the initial static strength or rather yielding limit of the material. In a scientific point of view, fatigue can be defined as a permanent, localized, and progressive structural change (i.e., damage) that occurs in a material subjected to cyclic or fluctuating strains. The fatigue life of a material or a component is defined as the total number of stress cycles required to cause the critical failure, where failure being understood as a state in which the structure no longer functions for the purpose it is designed.

The ability of predicting the life of laminates is important for designing, operation, and safety analysis of a composite structure under specific conditions. In laminated composite materials, the fatigue process also involves several damage mechanisms that result in the degradation of the structure. Once more, the most important fatigue damage mechanism is considered to be inter-laminar damage or delamination. Unlike in the case of delamination under static loadings, the number of research work devoted in appreciating damage evolution (i.e., for delamination) under fatigue loading seems to be significantly scarce [Leif E. Asp et al., 2001; Robinson P. et al., 2005; Ijaz H, 2009].

The determination of fatigue characteristics for metals is based on a well known theory called 'Paris Law' [Paris P., 1964]. Simply, the theory made it possible to make a quantitative prediction of the residual life for a crack of a certain size. In more details, Paris theory suggests that, for metals, there is a threshold level in the stress intensity amplitude below which no crack growth will take place. Above this threshold value, fatigue crack propagation is governed by the expression given below.

$$\frac{dA}{dN} = E(\Delta K)^r \quad (5.01)$$

Here,  $A$  is the surface area of crack and  $N$  is the number of loading cycles. Therefore,  $\frac{dA}{dN}$  is the fatigue crack growth rate and  $\Delta K$  is the magnitude of the stress intensity factor variation (i.e.,  $\Delta K = K_{\max} - K_{\min}$ ). Note that, the constants  $E$  and  $r$  are evaluated using the Paris Plot. However, evaluation of stress intensity factors is problematic for composite materials. Therefore, a relationship similar to that in Equation 5.01, based on strain energy release rate was proposed [Wilkins D. J. et al., 1982]. The equation has the form,

$$\frac{dA}{dN} = D(\Delta G)^s \quad (5.02)$$

Here,  $\Delta G = G_{\max} - G_{\min}$ , is the cyclic variation of energy release rate and depends on the loading conditions. Here,  $G_{\max}$  and  $G_{\min}$  are directly associated with  $F_{\max}$  (or  $U_{\max}$ ) and  $F_{\min}$  (or  $U_{\min}$ ) respectively. For a composite laminate specimen having a constant width,  $b$  and a crack length,  $a$ ; then  $A = ab$ . Now, if one consider the minimum energy release rate,  $G_{\min}$  to be zero, then  $\Delta G = G_{\max}$ .



Therefore, Paris Law for composite material can be rewritten in the form,

$$\frac{da}{dN} = B(G_{\max})^m \quad (5.03)$$

Parameters  $B$  and  $m$  depends on the material and fracture mode, which is typically quantified with a parameter known as ‘mode-ratio’ ( $\phi_1 = G_I / (G_I + G_{II} + G_{III})$  for MODE-I or  $\phi_2 = G_{II} / (G_I + G_{II} + G_{III})$  for MODE-II or  $\phi_3 = G_{III} / (G_I + G_{II} + G_{III})$  for MODE-III in 3-dimensions). Note that, In the Fracture Mechanics approach for predicting delamination, the fatigue crack growth rate is assumed to be intimately related to the amplitude of the critical energy release rate (i.e.,  $G_C$ ) and therefore one could write the above formulation in a normalized form as expressed below.

$$\frac{da}{dN} = P \left( \frac{G_{\max}}{G_C} \right)^q \quad (5.04)$$

This correlation can then be plotted on a log-log diagram as shown in Figure 5.01.

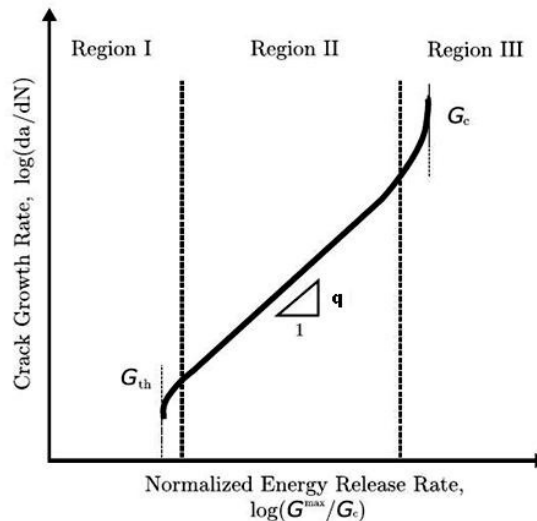


Figure 5.01 : Fatigue crack growth characteristic curve

As depicted in the figure, the diagram can be divided into three main regions depending upon the differences of the slopes of the curve. In Region I; there exists a threshold value of the energy release rate ( $G_{th}$ ) below which no crack growth will take place. Beyond the threshold limit the crack starts to propagate non-linearly with increasing  $G_{\max}$  (actually  $\Delta G$ ). Next, in Region II; the crack growth rate is linearly related to the normalized energy release rate and, slope of the curve is equal to exponent,  $q$  of Paris law. Finally, in Region III; the crack growth rate increases non-linearly to a point that corresponds to the fracture toughness ( $G_C$ ) of the material.

In literature, Damage Mechanics approaches are available for predicting fatigue damage evolution in both metals and laminated composites [Daudeville L. et al., 2002; Robinson P. et al., 2005; Turon A. et al., 2007]. There are several models that extend cohesive laws for monotonic loading into forms suitable for cyclic loading. Here, the total damage in the interface is considered to be sum of the damage caused by quasi-static load and the damage that results from the cyclic loads. Most of these extended cohesive models determine the accumulated damage on a cycle-by-cycle basis and as the number of cycles grows the computation becomes much more intractable. Therefore, for high-cycle fatigue, the damage evolution that results from cyclic loads needs to be formulated as a function of the number of cycles and displacement jumps.

## 5.2 Proposed Fatigue Damage Evolution Law

The approach is based on the cohesive zone model concept. A constitutive damage model previously developed for quasi-static loads is enhanced to incorporate a damage evolution law for high-cycle fatigue. For different problems, the damage law needs to be calibrated by adjusting several parameters through trial-and-error analysis. The key idea here is to use results of Fracture Mechanics tests to determine parameters of the Damage Mechanics model. Therefore, the proposed model basically links Fracture Mechanics to Damage Mechanics. The model relates damage accumulation to number of load cycles while taking into account different loading conditions (i.e., Pure-/ Mixed-Mode).

In the proposed model, the delamination growth under high-cycle fatigue is considered to be a combination of delamination due to quasi-static loading and cyclic variation of the loading [Robinson P. et al., 2005, Ijaz H., 2009]. Therefore, total damage evolution in rate format is expressed as,

$$\dot{d}_{iT} = \dot{d}_{iS} + \dot{d}_{iF} \quad (5.05)$$

Here, subscript  $i=1,2,3$ , corresponds to failure modes; Mode-I, Mode-II and Mode-III respectively.

The terms,  $\dot{d}_{iS}$  and  $\dot{d}_{iF}$  are associated to static and fatigue delamination respectively. Therefore, the formulation of the fatigue damage model is performed in two steps. Note that hereafter, subscript  $i$  is eliminated for the sake of simplicity.

### 5.2.1. Static Part of Cyclic Damage Evolution

If the 'material function' describing the static damage evolution is written as,

$$\dot{d}_S = \left[ \frac{n}{n+1} \frac{\langle \bar{Y} - Y_0 \rangle_+}{Y_C - Y_0} \right]^n \quad (5.06)$$

Then by integrating Equation 5.08 with time,

$$\int_t^{t+\Delta t} \dot{d}_S dt = \int_t^{t+\Delta t} \left[ \frac{n}{n+1} \frac{\langle \bar{Y} - Y_0 \rangle_+}{Y_C - Y_0} \right]^n dt$$

$$\int_t^{t+\Delta t} \dot{d}_S dt = n \left[ \frac{n}{n+1} \frac{1}{Y_C - Y_0} \right]^{t+\Delta t} \int_t^{t+\Delta t} \left[ \langle \bar{Y} - Y_0 \rangle_+ \right]^{n-1} \dot{\bar{Y}} dt \quad (5.07)$$

Let  $t$  and  $t+\Delta t$  are times corresponding to end of load cycles  $N$  and  $N+\Delta N$  respectively,

$$\int_{d_S(N)}^{d_S(N+\Delta N)} d(d_S) = n \left[ \frac{n}{n+1} \frac{1}{Y_C - Y_0} \right]^n \int_{\bar{Y}(N)}^{\bar{Y}(N+\Delta N)} \left[ \langle \bar{Y} - Y_0 \rangle_+ \right]^{n-1} d(\bar{Y}) \quad (5.08)$$

Here,  $d_S(N+\Delta N)$  &  $\bar{Y}(N+\Delta N)$  correspond to end of cycle  $N+\Delta N$  and,  $d_S(N)$  &  $\bar{Y}(N)$  correspond to end of cycle  $N$ .

Hence, expression for the static damage variable can be written as,

$$d_S(N + \Delta N) = d_S(N) + \left[ \frac{n}{n+1} \frac{1}{Y_C - Y_0} \right]^n \left[ \left( \langle \bar{Y}(N + \Delta N) - Y_0 \rangle_+ \right)^n - \left( \langle \bar{Y}(N) - Y_0 \rangle_+ \right)^n \right] \quad (5.09)$$

Here,

$$\bar{Y}(N + \Delta N) = \left( \left( Y_{d_3}(N + \Delta N) \right)^\alpha + \left( \gamma_1 Y_{d_1}(N + \Delta N) \right)^\alpha + \left( \gamma_2 Y_{d_2}(N + \Delta N) \right)^\alpha \right)^{1/\alpha} \quad (5.10)$$

$$\bar{Y}(N + \Delta N) = \left( \left( Y_{d_3}(N) \right)^\alpha + \left( \gamma_1 Y_{d_1}(N) \right)^\alpha + \left( \gamma_2 Y_{d_2}(N) \right)^\alpha \right)^{1/\alpha} \quad (5.11)$$

$$\left. \begin{aligned} Y_{d_3}(N + \Delta N) &= \frac{1}{2} k_3^0 \left[ \langle U_3(N + \Delta N) \rangle_+ \right]^2 \\ Y_{d_1}(N + \Delta N) &= \frac{1}{2} k_1^0 \left[ U_1(N + \Delta N) \right]^2 \\ Y_{d_2}(N + \Delta N) &= \frac{1}{2} k_2^0 \left[ U_2(N + \Delta N) \right]^2 \end{aligned} \right\} \quad (5.12)$$

$$\left. \begin{aligned} Y_{d_3}(N) &= \frac{1}{2} k_3^0 \left[ \langle U_3(N) \rangle_+ \right]^2 \\ Y_{d_1}(N) &= \frac{1}{2} k_1^0 \left[ U_1(N) \right]^2 \\ Y_{d_2}(N) &= \frac{1}{2} k_2^0 \left[ U_2(N) \right]^2 \end{aligned} \right\} \quad (5.13)$$

Note that, the effect of static delamination in fatigue will be considered if and only if,

$\bar{Y}(N + \Delta N) \geq \bar{Y}(N)$  (i.e.,  $d_S(N + \Delta N) \geq d_S(N)$ ) is true.

### 5.2.2. Fatigue Part of Cyclic Damage Evolution

The proposed fatigue damage evolution is a function of the equivalent damage energy release rate ( $\bar{Y}$ ) and critical damage energy release rate ( $Y_C$ ). The definition of the proposed damage model [Gornet L. and Ijaz H., 2010] is as follows,

$$\dot{d}_F = \begin{cases} g \left( d, \frac{\bar{Y}}{Y_C} \right) \frac{\dot{\bar{Y}}}{Y_C} & \text{if } \bar{Y} \geq 0 \text{ and } f \geq 0 \\ 0 & \text{if } \bar{Y} < 0 \text{ and } f < 0 \end{cases} \quad (5.14)$$

Here,  $f = \bar{Y} - Y_{th}$  is called the damage loading function and defines the threshold of the fatigue delamination growth. Note that, damage will grow if and only if,  $f \geq 0$ .

The dimensionless function,  $g$  is expressed as,

$$g\left(d, \frac{\bar{Y}}{Y_C}\right) = C e^{\lambda d} \left(\frac{\bar{Y}}{Y_C}\right)^{\beta} \quad (5.15)$$

Here;  $C$ ,  $\lambda$  and  $\beta$  are 'material parameters' that need to be determined experimentally. Note that,  $\lambda$  is a constant; where  $C$  and  $\beta$  are functions of 'mode-ratio ( $\phi$ )'.

The formulation given here does not take into account the effect of 'R-ratio' (i.e.,  $R=0$ , where  $R = \frac{\bar{Y}_{\min}}{\bar{Y}_{\max}}$ ) In other words, the load is said to be varying between maximum  $F_{\max}$  (or  $U_{\max}$ ) and zero values.

The fatigue damage variable at end of each time increment ( $\Delta t$ ) can be written as,

$$d_F(t + \Delta t) = d_F(t) + \int_t^{t+\Delta t} \dot{d}_F dt \quad (5.16)$$

Let  $t$  and  $t + \Delta t$  are the times corresponding to end of cycles  $N$  and  $N + \Delta N$  respectively, and if  $t_n$  and  $t_n + \Delta t_n$  are the times corresponding to start and end of cycle number  $n$  (i.e.,  $n=1, 2, 3, \dots, \Delta N$ ).

Then equation can be rewritten as,

$$d_F(t + \Delta t) = d_F(t) + \int_t^{t+\Delta t} \left[ \int_{t_n}^{t_n+\Delta t_n} \dot{d}_F dt \right] dt \quad (5.17)$$

For  $\bar{Y} \geq 0$  and  $f \geq 0$ , the evolution of damage within one cycle due to fatigue,

$$\int_{t_n}^{t_n+\Delta t_n} \dot{d}_F dt = \int_{t_n}^{t_n+\Delta t_n} C e^{\lambda d} \left(\frac{\bar{Y}}{Y_C}\right)^{\beta} \frac{\dot{\bar{Y}}}{Y_C} dt = \int_{Y(N+n)}^{Y(N+n+1)} C e^{\lambda d} \left(\frac{1}{Y_C}\right)^{1+\beta} \bar{Y}^{-\beta} d\bar{Y} \quad (5.18)$$

Here,  $Y(N+n+1)$  and  $Y(N+n)$  correspond to cycle number  $N+n$  and  $N+n+1$  respectively.

Therefore, the fatigue damage evolution for a single cycle can be written as,

$$G_F(d_F, \bar{Y}) = \frac{C e^{\lambda d}}{1 + \beta} \left(\frac{\bar{Y}}{Y_C}\right)^{1+\beta} \quad (5.19)$$

Note that, the value of  $\bar{Y}$  for a given cycle corresponds to the peak value of the cycle. Next, the integral in equation can be approximated by a numerical integration scheme. Possible choices include Trapezoidal rule and Simpson's rule.

The formulation given hereunder is based on the trapezoidal rule. The estimate is made by multiplying the average of  $G_F(d_F, \bar{Y})$  (evaluated by taking values at the beginning (i.e.,  $t \rightarrow N$ ) and end (i.e.,  $t \rightarrow N + \Delta N$ ) of the increment) with number of cycles in the increment (i.e.,  $\Delta N$ ).

Now, expression for the fatigue damage variable can be written as,

$$d_F(N + \Delta N) = d_F(N) + \frac{1}{2} \left[ G_F(d_F(N + \Delta N), \bar{Y}(N + \Delta N)) + G_F(d_F(N), \bar{Y}(N)) \right] \Delta N \quad (5.20)$$

Note that, equation is an implicit equation. For example, the fatigue damage variable  $d_F(N + \Delta N)$ , appears on both side of the relation. To solve the equation, one can replace  $d_F(N + \Delta N)$  on the right-hand side with a 'predictor' (i.e.,  $d_F^P$ ) based on Forward Euler Technique [Ijaz H., 2009; and other]. The expression used for the predictor is as follows,

$$d_F^P = d_F(N) + G_F(d_F(N), \bar{Y}(N)) \Delta N \quad (5.21)$$

Therefore, the modified expression for the fatigue damage variable reads as,

$$d_F(N + \Delta N) = d_F(N) + \frac{1}{2} \left[ G_F(d_F^P, \bar{Y}(N + \Delta N)) + G_F(d_F(N), \bar{Y}(N)) \right] \Delta N \quad (5.22)$$

### 5.2.3. Complete Fatigue Damage Evolution Law

Finally, the expression for the complete fatigue damage evolution law is as follows,

$$d(N + \Delta N) = d(N) + \left[ \frac{n}{n+1} \frac{1}{Y_C - Y_0} \right]^n \left[ \left( \langle \bar{Y}(N + \Delta N) - Y_0 \rangle_+ \right)^n - \left( \langle \bar{Y}(N) - Y_0 \rangle_+ \right)^n \right] + \frac{1}{2} \left[ G_F(d_F^P, \bar{Y}(N + \Delta N)) + G_F(d_F(N), \bar{Y}(N)) \right] \Delta N \quad (5.23)$$

Here,

$$d(N + \Delta N) = d_S(N + \Delta N) + d_F(N + \Delta N) \text{ and } d(N) = d_S(N) + d_F(N) .$$

# Chapter 6

## Analysis of Fatigue Simulations

### 6.1 Analysis Procedure

The analysis procedure is similar to the description given for static case with the following exceptions.

Pre-processing	}	Define type of analysis (2D-plane stress) Define geometry & mesh Define material parameters for arms Define ' <u>fatigue damage model</u> ' parameters for interface Define boundary, contact & loading conditions Define convergence criteria
Solution	}	Launch nonlinear calculation
Post-processing	}	Plot evolution of crack length vs. number of loading cycles, damage variable vs. number of loading cycles, energy release rate vs. crack length, etc.
Analysis	}	Compare and contrast simulated results with experimental data and investigate the effect of fatigue model parameters

Note that, for simulation purposes a static load increasing from 'zero' to 'peak of the cyclic load' will precede the start of the envelope of the cyclic (fatigue) load. A pictorial representation of the actual applied load and the numerically applied load are shown in Figure 6.01 given below.

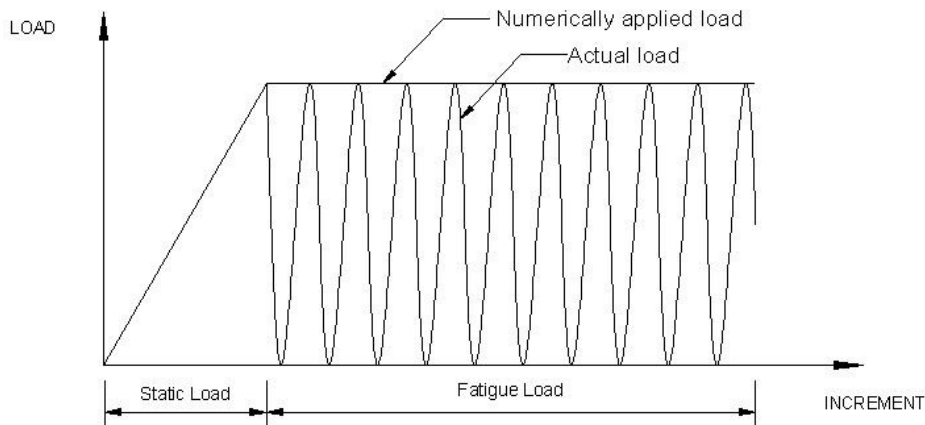


Figure 6.01 : Numerically applied fatigue load and actual fatigue load

In addition, an assumption is made with regard to the numerically applied loading to simplify the nonlinear calculation process [Robinson P. et al., 2005; Ijaz H., 2009]. Note that, in this particular study the actual load is taken to be oscillating between zero and a maximum value as illustrated in the above figure. For high cyclic fatigue the numerically applied load will be taken as a constant and will be equal to the maximum value of the actual load. The cyclic nature of the loading scheme will be taken in to account by the formulas governing the evolution of the fatigue damage variable. Interestingly this means that fatigue damage evolution should be independent of the number of loading cycles specified for a given increment. This aspect will be validated in a later section. The finite element implementation of the 'fatigue damage model' for the interface in 3D is summarized in the two tables given hereafter.

First, Table 6.01 includes the damage evolution criteria for the ‘Static Load’ of the loading scheme. Secondly, Table 6.02 includes the damage evolution criteria for the ‘Fatigue Load’ of the loading scheme.

Static Load Scheme	
Step-1	Compute damage energy release rate, i.e., $\bar{Y}(t)$ using $Y_{d_1}, Y_{d_2}$ & $Y_{d_3}$
Step-2	Compute isotropic damage function, i.e., $w(\bar{Y})$ using $\bar{Y}, Y_C$ & $Y_0$
Step-3	Compute damage variables, i.e., $d_1 = d_2 = d_3 = w(\bar{Y})$
Step-4	Compute stresses for the interface, i.e., $\sigma_{13} = k_1^0(1-d_1)U_1, \sigma_{23} = k_2^0(1-d_2)U_2, \sigma_{33} = k_3^0(1-d_3)U_3$

Table 6.01 : FE implementation of static load scheme of fatigue damage model

Fatigue Load Scheme	
Step-1	Compute damage energy release rate, i.e., $\bar{Y}(N + \Delta N)$ using $Y_{d_1}(N + \Delta N), Y_{d_2}(N + \Delta N)$ & $Y_{d_3}(N + \Delta N)$
Step-2	Compute static damage function, i.e., $w(\bar{Y}(N + \Delta N))$ using $\bar{Y}(N + \Delta N), Y_C$ & $Y_0$
Step-3	Compute static damage variable, i.e., $d_S(N + \Delta N)$ using $w(\bar{Y}(N + \Delta N))$ & $w(\bar{Y})_{ending\ value\ of\ static\ load\ scheme}$
Step-4	Compute predictor fatigue damage variable, i.e., $d_F^P$ using $d_F(N), \bar{Y}(N)$ & $\Delta N$
Step-5	Compute fatigue damage function, i.e., $G_F(N + \Delta N)$ using $C, \lambda, \beta, d_F^P, \bar{Y}(N + \Delta N)$ & $Y_C$
Step-6	Compute fatigue damage variable, i.e., $d_F(N + \Delta N)$ using $G_F(N + \Delta N), G_F(N)$ & $\Delta N$
Step-7	Compute complete damage variable, i.e., $d(N + \Delta N)$ using $d_S(N + \Delta N)$ & $d_F(N + \Delta N)$
Step-8	Compute stresses for the interface, i.e., $\sigma_{13} = k_1^0(1-d_1)U_1, \sigma_{23} = k_2^0(1-d_2)U_2, \sigma_{33} = k_3^0(1-d_3)U_3$

Table 6.02 : FE implementation of fatigue load scheme of fatigue damage model

Initially, the nonlinear calculations will be performed using the Static part (i.e., Table 6.01) of the model. After a given number of increments the Fatigue part (i.e., Table 6.02) of the model will be used. Note that, with this formulation the user can easily select the maximum value of the numerically applied load (or  $\bar{Y}_{max}$ ) by specifying the number of increments to switch from Static to Fatigue loading scheme.

## 6.2 Identification of Fatigue Damage Model Parameters

Similarly as explained in Chapter 4, the damage model parameters of the fatigue damage evolution law needs to be determined beforehand. There are number of parameters to be determined for both static part and also fatigue part of the damage evolution. Identification of parameters such as,  $Y_C$ ,  $Y_0$ ,  $\gamma_1$ ,  $\gamma_2$ ,  $\alpha$  and interface rigidities ( $k_1^0$ ,  $k_2^0$  and  $k_3^0$ ) have already been discussed in Section 4.2.

For the fatigue part, one needs to identify four key parameters, namely  $C$ ,  $\lambda$ ,  $\beta$  and  $\Delta N$ . The choice of the number of increment cycles does not affect the final result of the computation. Therefore, the user is free to select a value for  $\Delta N$ . Note that, if  $\Delta N$  is too large for a given time increment several elements may fail together and the delamination process will be rapid. On the other hand, if  $\Delta N$  is too small, the computational time would increase significantly. Generally, a good value for  $\Delta N$  is selected depending on the problem in hand.

The identification of parameter values for  $C$ ,  $\lambda$  and  $\beta$  are based on a trial-and-error approach. In other words, the user has to calibrate the fatigue damage model for a given problem using the available experimental data. However, FE inverse-optimization procedure may also be considered for future studies. Identification of these parameters without the aid of any experimental data for any given Mixed-Mode delamination will also be discussed in a later section. Here, a non-monotonic relation between each parameter is established using results of three different Fracture Mechanics tests.

## 6.3 Validation of Proposed Fatigue Damage Model

### 6.3.1 Experimental Methods

To validate and verify the effectiveness of the proposed fatigue damage model, simulation results need to be compared with experimental results. In the present study, experimental results reported by Leif E. Asp. et al., 2001 will be used and details of test methods are explained hereunder. As reported, they studied the effects on the inter-laminar toughness in fatigue with three different test methods. For pure Mode-I and Mode-II, the DCB and 3ENF tests were utilized, respectively. For Mixed-Mode load case (i.e.,  $G_{II} / G_{Total} = 0.5$ ) the MMB test was employed. All tests were performed on a modified version of the Mixed-Mode bending test rig [Crews J. H. and Reeder J.R., 1988].

All specimens had been manufactured from HTA/6376C carbon/epoxy unidirectional prepregs and the cured plies have the elastic properties given in Table 6.03.

$E_{11}$ - 120,000 [MPa]	$E_{22}$ - 10,500 [MPa]	$E_{33}$ - 10,500 [MPa]
$\nu_{12}$ - 0.3	$\nu_{23}$ - 0.51	$\nu_{13}$ - 0.3
$G_{12}$ - 5,250 [MPa]	$G_{23}$ - 3,480 [MPa]	$G_{13}$ - 5,250 [MPa]

Table 6.03 : Elastic properties of HTA/6376C unidirectional prepregs reported by Leif Asp. et al.

The ply thickness was  $130[\mu m]$  and the specimen lay-up was  $[0_{12} // (\pm 5 / 0_4)_S]$ , where sign '//' refers to the plane of the artificial delamination. The dimensions of the rectangular shaped specimens were; width  $b = 20[mm]$ , length  $l_s = 150[mm]$ , and nominal thickness  $2h = 3.1[mm]$ .



A schematic of the specimen geometry is shown in Figure 6.02.

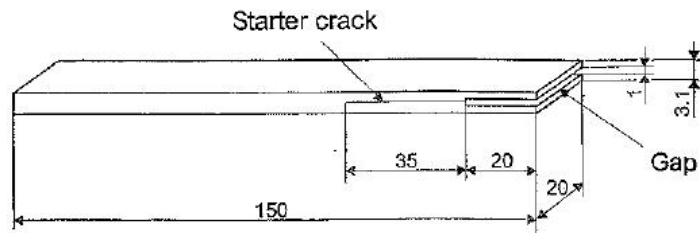


Figure 6.02 : Schematic of the specimen geometry used for fatigue test

A schematic of the test rig and the applied load is (i.e.,  $P$ ) is presented in Figure 6.03. Here, span of the testing length  $2L = 100 [mm]$ .

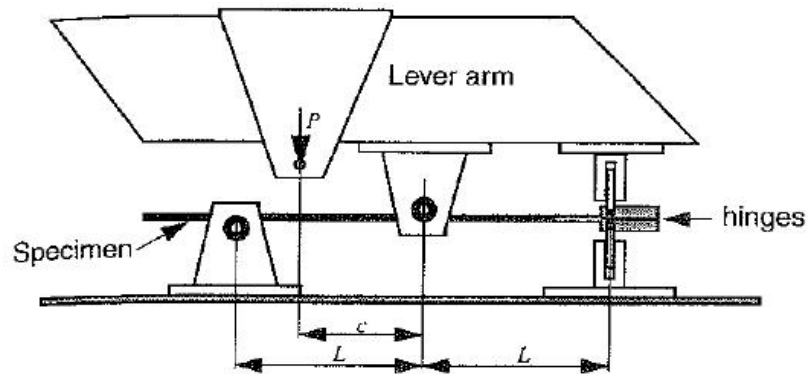


Figure 6.03 : Schematic of the fatigue test rig

All tests were performed at a frequency of  $5 [Hz]$ , at ambient conditions. The specimens were tested in fatigue at a R-value of 0.1, at a constant displacement amplitude. For each test method, at least seven specimens were tested. The experimental results were then presented in the form of 'normalized Paris plots' as shown in Figure 6.04 given below.

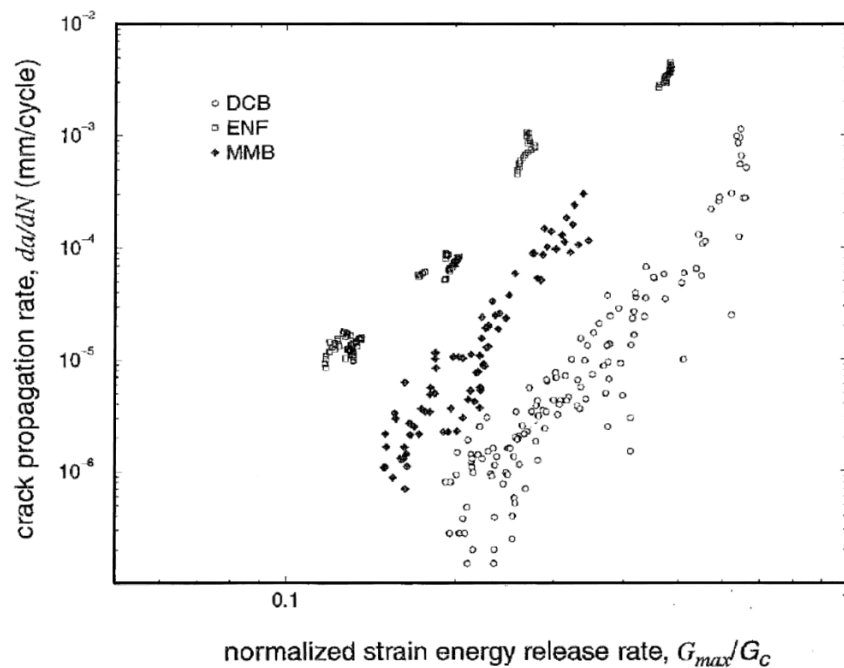


Figure 6.04 : Normalized Paris Plots for DCB, ENF and MMB tests

### 6.3.2 Finite Element Model

The geometry of the finite element model used for all three tests, namely DCB (pure MODE-I), 3ENF (pure MODE-II) and MMB (Mixed-Mode;  $G_{II} / G_{Total} = 0.5$ ) is given below in Figure 6.05.

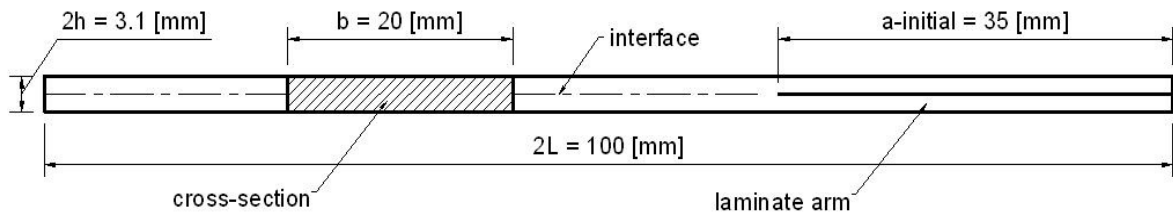


Figure 6.05 : Dimensions of the model geometry used for fatigue simulations

Simulations were performed considering a state of 'PLANE STRESS' condition in 2D. Values of some model parameters and FEs used are given in Table 6.04.

Laminate arms (2D)	Geo. Support / Finite Element : QUA4 / QUA4 Deg. of Interpolation : 1
Interface (1D)	Geo. Support / Finite Element : RAC2 / JOI2 Deg. of Interpolation : 1

Table 6.04 : Finite elements for arms and interface for fatigue simulations

In addition, a FE mesh having a constant element size, i.e.,  $ES = 0.1$  was used to carry out all simulations for each test campaign. An example of a mesh for a DCB test is given in Figure 6.06.

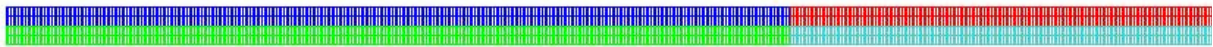


Figure 6.06 : Mesh used for fatigue simulations

The model parameters that were commonly used in all simulation are listed in Table 6.05 given below. Note that, the identification of these parameters was carried out earlier under the static part of the study.

$Y_C [N/mm]$	-	0.26	$\alpha$	-	1.4	$k_1^0$	-	2395 [MPa/mm]
$\gamma_1$	-	0.2595	$n$	-	0.5	$k_3^0$	-	9230 [MPa/mm]

Table 6.05 : General fatigue damage model parameters

The simulations were performed at different values of  $G_{max}/G_C$  for each type of mode of failure. The additional data plots describing the evolution of ' $G_{max}$ ' with crack length ' $a$ ' were also used to determine the potential number of loading cycles elapsed for the extension of the cracks. Note that, the selected ratios of  $G_{max}/G_C$  correspond to the crack extension data given in those plots (also included in the upcoming Sections).

Here, for tests at  $R = 0.1$ ,  $G_{min}$  is considerably small compared to  $G_{max}$ , hence the proposed fatigue damage model can be used to simulate the above mentioned tests. In all simulations evolution of energy release rate at the crack tip (i.e., using gauss point) was monitored by local means. It was then used to determine the  $G_{max}/G_C$  ratio corresponding to each increment of the loading condition.

Now, depending on the test data simulations were performed for different  $G_{max}/G_C$  ratios by simply changing the prescribed displacement limit for the static loading scheme accordingly. At the end of the static loading scheme the fatigue model will take over the computation process.

### 6.3.3 Simulation Results for DCB Fatigue Test

Application of the boundary conditions and loading conditions (i.e., imposed displacement) for the DCB test is as shown in Figure 6.07.

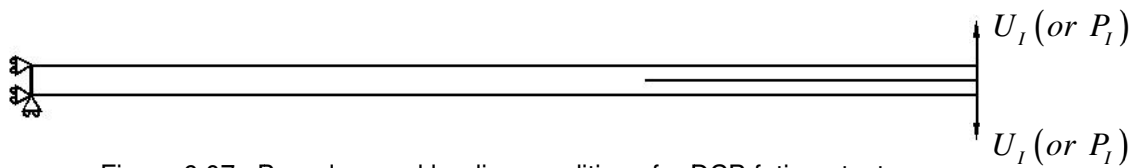


Figure 6.07 : Boundary and loading conditions for DCB fatigue test

The parameters  $C$ ,  $\beta$  and  $\lambda$  were calibrated carefully and were found to be equal to values given below. Note that, Mode-I fracture toughness of the specimen was reported equal to  $0.26 [N/mm]$ .

$$C = 2.0E-2 \quad \beta = 3.5 \quad \lambda = 1$$

Figure 6.08 given below illustrate the simulated results plotted over the experimental results.

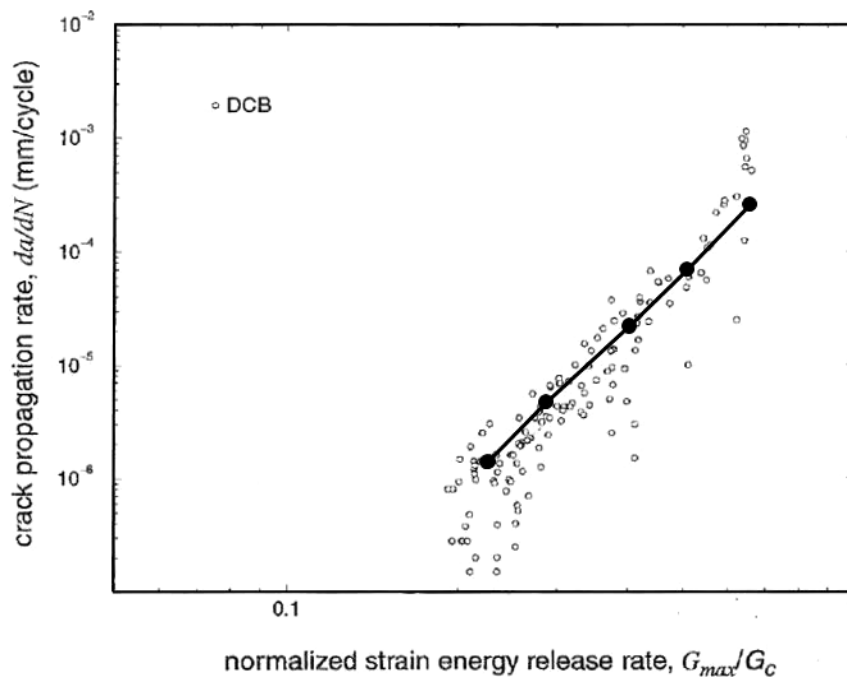


Figure 6.08 : Normalized Paris Plot for simulated and experimental results of DCB fatigue tests

The normalized Paris Plot obtained from the simulated results is found to be in good agreement with the experimental data. Interestingly, the simulated results for  $G_{max}/G_{IC}$  ratio equal to 0.23 and 0.66 have retained their linear relationship with rest of the points although they close to the nonlinear regions of the Paris plot.

Figure 6.09 given below includes evolution of crack extension against number of loading cycles. Here, a comparison has been made between simulated results for each  $G_{max}/G_{IC}$  ratio and provides a better understand of the delamination growth for Mode-I.

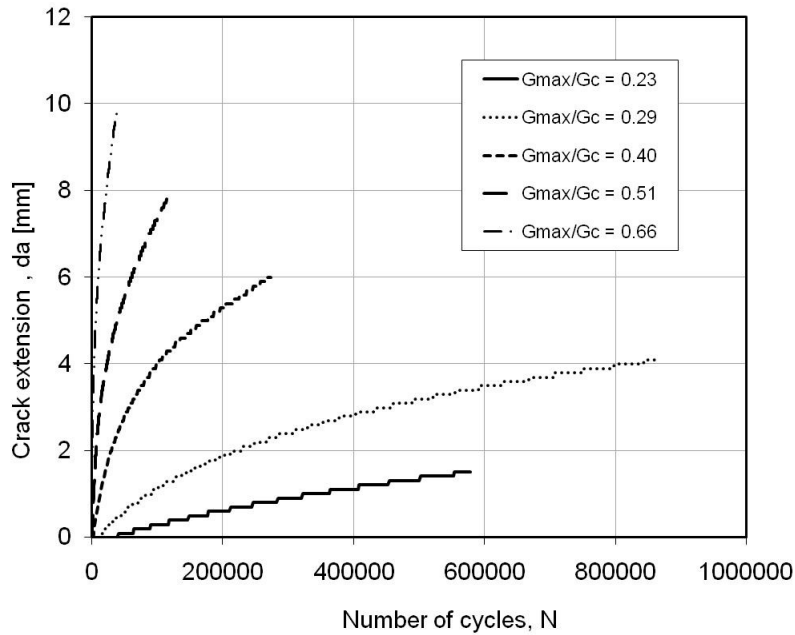


Figure 6.09 : Evolution of crack extension for different  $G_{max}/G_{IC}$  ratios in DCB fatigue test

Note that, simulations were only performed in the range as specified in experiments. From figure it can be seen that, crack growth becomes non-linear with increasing number of cycles. This phenomenon seems to become gradually severe with increasing  $G_{max}/G_{IC}$  ratio. As a result the crack growth rate seems to vary with number of loading cycles. Here, a simple assumption is made and crack growth rate is considered to be constant. In practice, it is determined by taking the ratio of total crack extension with number of cycles elapsed. However, the final results as illustrated in Figure 6.08 are satisfactory.

A comparison between experimental and simulated results was made to investigate the variation of strain energy release rate with crack growth. See Figure 6.10.

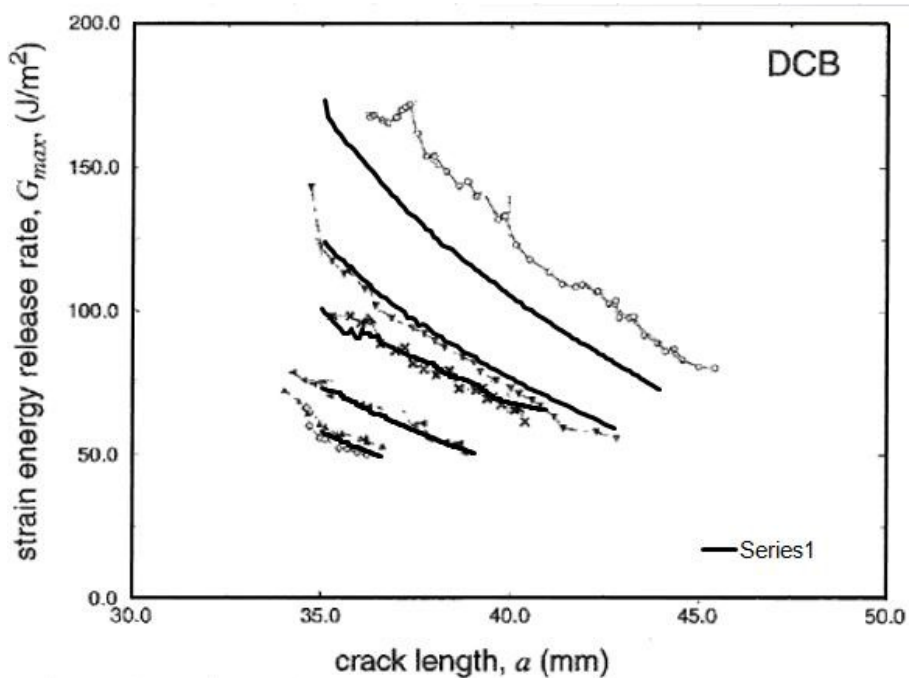


Figure 6.10 : Evolution of  $G_{max}$  for different  $G_{max}/G_{IC}$  ratios in DCB fatigue test

Variations of strain energy release rate with crack length are found to be in good agreement with the experimental data. Note that, numerical result obtained for the highest  $G_{max}/G_{IC}$  ratio does not correspond to the experimental result (here, lack of information on actual start point of the experimental result made it difficult to do a proper analysis). The additional simulation was performed to appreciate the evolution of the crack length and energy release rate for  $G_{max}/G_{IC} = 0.66$ . In general, the decrease of the energy release rate with increasing crack length seems to be satisfactorily captured by the proposed model.

Identification of  $B$  and  $m$  of the Paris Law was carried out by plotting  $G_{max}$  against  $da/dN$  and obtaining the equation of the best fit for the data points as given below in Figure 6.11.

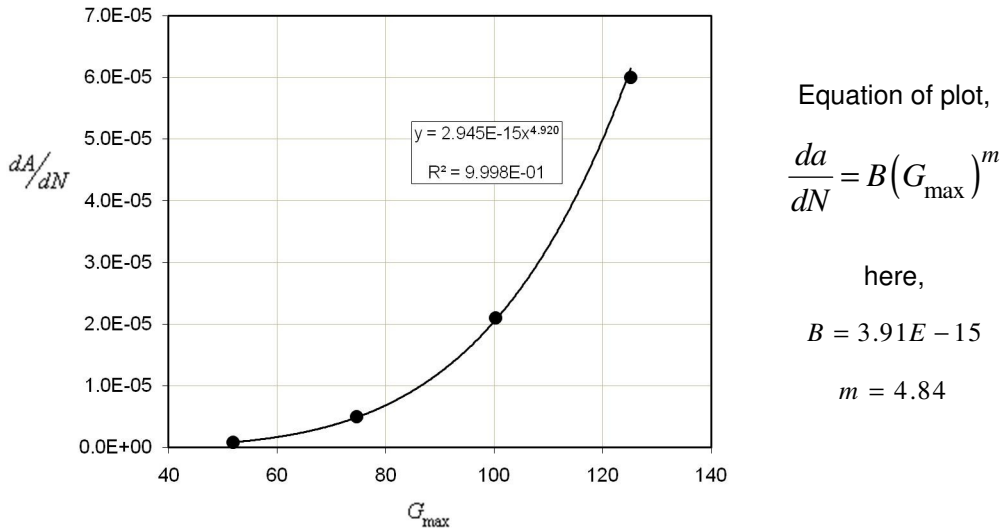


Figure 6.11 : Evolution of crack growth rate with  $G_{max}$  in DCB fatigue test

### 6.3.4 Simulation Results for 3ENF Fatigue Test

Application of the boundary conditions and loading conditions (i.e., imposed displacement) for the 3ENF test is illustrated in Figure 6.12.

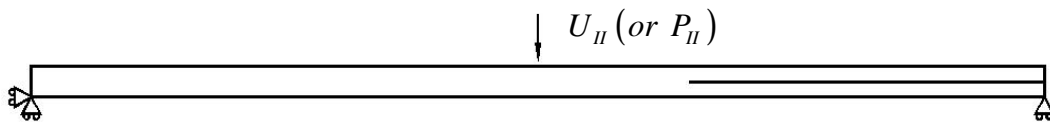


Figure 6.12 : Boundary and loading conditions for 3ENF fatigue test

The parameters  $C$ ,  $\beta$  and  $\lambda$  were calibrated once more and were found to be equal to values given below. Note that, Mode-II fracture toughness of the specimen was reported equal to  $1.002 [N / mm]$ .

$$C = 7.0E-3 \quad \beta = 2.2 \quad \lambda = 1$$

Figure 6.13 given below illustrate the simulated results plotted over the experimental results.

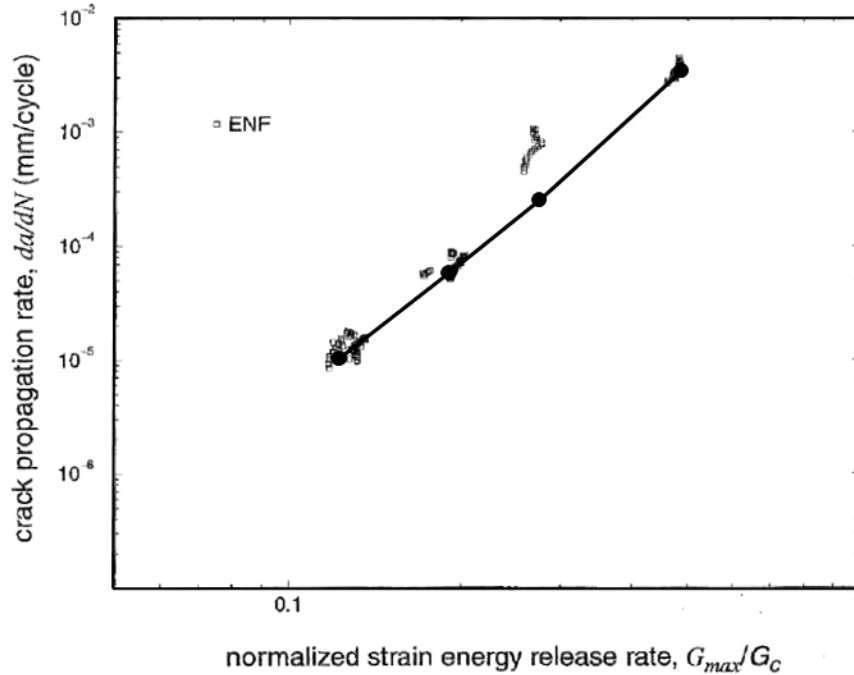


Figure 6.13 : Normalized Paris Plot for simulated and experimental results of 3ENF fatigue tests

The simulated results are in good agreement with the experimental results. Once more, the fatigue model successfully reproduces the expected linear behaviour. Note that, this test case only considers the fatigue behaviour of the linear region in the Paris Plot. The cluster of experimental data points corresponding to  $G_{max}/G_{IC} = 0.27$  seem to be lying slightly outside of the linear trend of the rest of the points. However, the corresponding numerical result has predicted the linear behaviour much more accurately.

In addition, an analysis on the evolution of crack extension with number of loading cycles was carried out and the results are given in Figure 6.14.

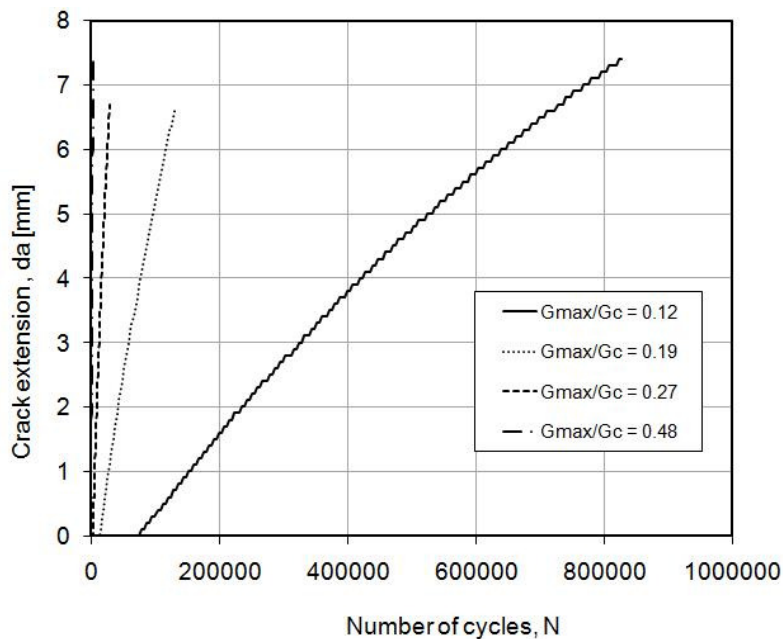


Figure 6.14 : Evolution of crack extension for different  $G_{max}/G_{IC}$  ratios in 3ENF fatigue test

Unlike in the case of DCB test, crack growth seems to take place in a linear manner and the phenomenon is unaffected by for different ratio values. Therefore, computation of crack growth rate becomes trivial.

Evolution of strain energy release rate with crack growth was monitored for each  $G_{max}/G_{IIC}$  ratio and comparisons of simulated against experimental results are given in Figure 6.15.

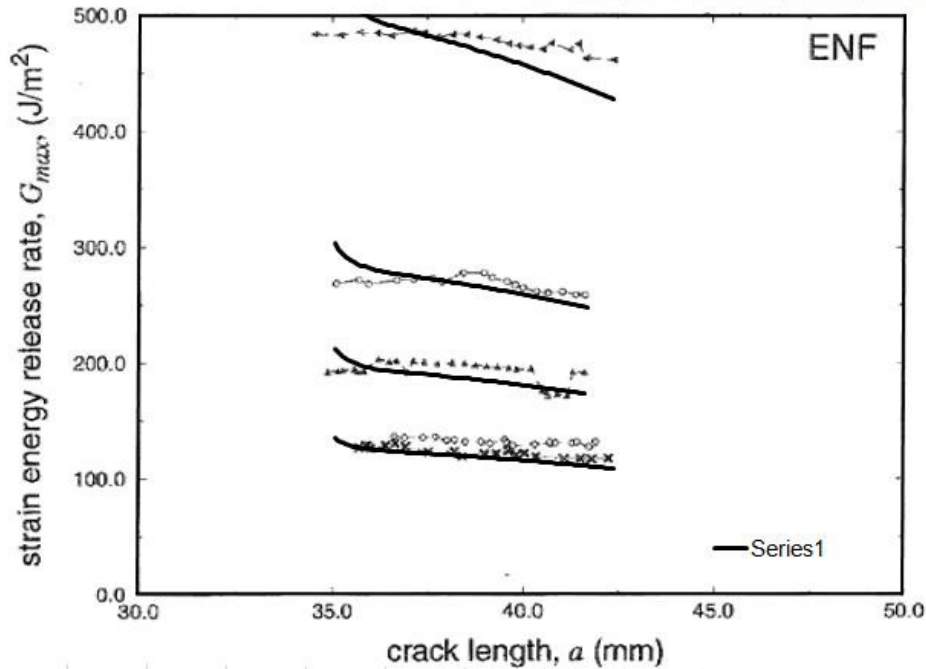
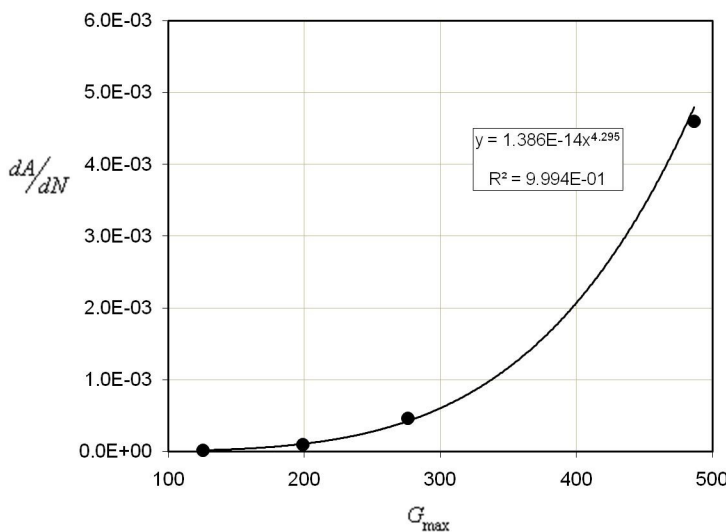


Figure 6.15 : Evolution of  $G_{max}$  for different  $G_{max}/G_{IIC}$  ratios in 3ENF fatigue test

Evolutions of strain energy release rate predicted by the simulations are in very good agreement with experimental results, except for the highest ratio where the result can only be considered satisfactorily. It is interesting to note that, for 3ENF test, reduction of strain energy release rate is considerably smaller than for DCB test.

Once more, the identification of  $B$  and  $m$  of the Paris law was performed by plotting  $G_{max}$  against  $da/dN$  and obtaining the equation of its best fit as given in Figure 6.16.



Equation of plot,

$$\frac{da}{dN} = B(G_{max})^m$$

here,

$$B = 1.36 E - 14$$

$$m = 4.23$$

Figure 6.16 : Evolution of crack growth rate with  $G_{max}$  in 3ENF fatigue test

### 6.3.5 Simulation Results for MMB Fatigue Test

Application of the boundary conditions and loading conditions (i.e., imposed displacement) for the MMB test is illustrated in Figure 6.17.

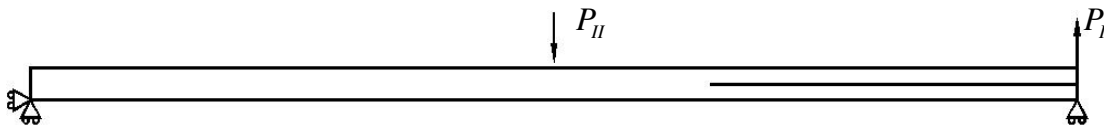


Figure 6.17 : Boundary and loading conditions for MMB fatigue test

Although evaluation of loads in the DCB and 3ENF tests were trivial, for the MMB test, however, the Mode-I and Mode-II load components (i.e.,  $P_I$  and  $P_{II}$  respectively) must be obtained from equilibrium considerations [Crews J. H. and Reeder J.R., 1988] and depend on the applied load (i.e.,  $P$ ) as shown in Figure 6.18 given below.

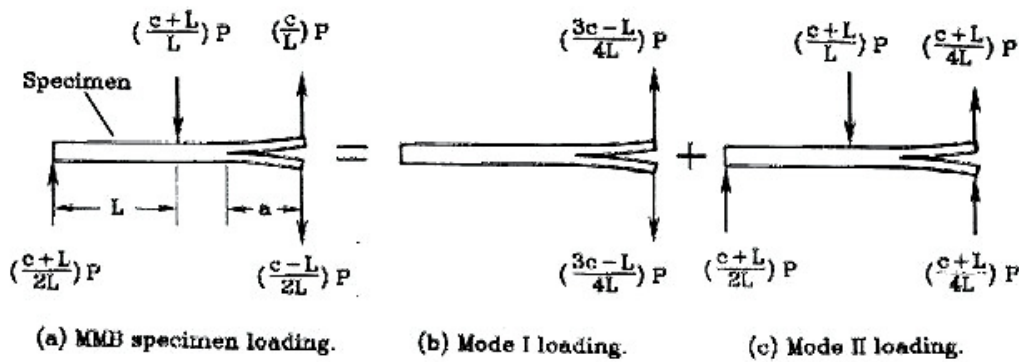


Figure 6.18 : Superposition of Mode-I and Mode-II loadings for Mixed-Mode case

For FE simulations, the expressions used for the load components are as follows,

$$P_I = \frac{1}{4}(3c - L) \frac{P}{L} \quad (6.01)$$

$$P_{II} = (c + L) \frac{P}{L} \quad (6.02)$$

Here,  $c$  is the distance from the center of the span of the testing length (i.e.,  $2L$ ) to the point of the applied load in the horizontal direction. Note that, both  $c$  and  $L$  are illustrated clearly in Figure 6.03. Under the assumption of cylindrical bending and mid-plane symmetry, the Mode-III strain energy release rate (i.e.,  $G_{III}$ ) in the MMB specimen is zero, and the total strain energy release rate for the MMB test is,

$$G_{TOT} = G_I + G_{II} \quad (6.03)$$

The Mixed-Mode fracture toughness with a mode mixity of  $\phi = G_{II} / G_{TOT} = 0.5$  was reported equal to  $0.447 [N/mm]$ . Note that, here mode-ratio (i.e.,  $\phi$ ) is taken with respect to Mode-II. Several FE simulations were first carried out to identify the value ' $c$ ' which corresponds to the mode-mixture of 0.5 and found to be equal to  $37 [mm]$ . Subsequently, the fatigue simulations for the selected Mixed-Mode were then carried out on that basis. Note that, for the MMB test a force loading condition was selected instead of imposed displacement condition.



Parameters  $C$ ,  $\beta$  and  $\lambda$  were calibrated for the new test and equal to values given below.

$$C = 1.5E-1 \quad \beta = 3.7 \quad \lambda = 1$$

Figure 6.19 shown hereunder includes the simulated results with experimental results.

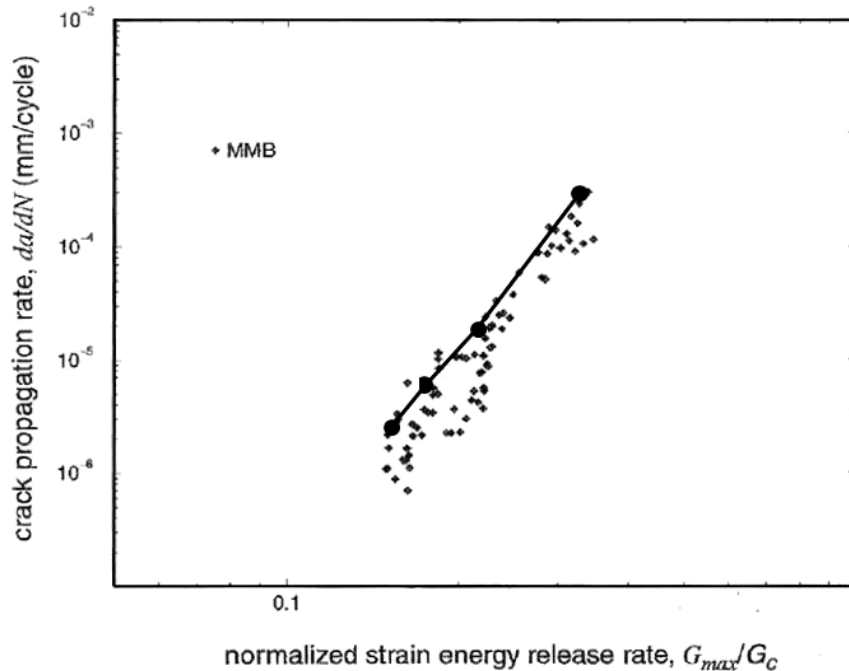


Figure 6.19 : Normalized Paris Plot for simulated and experimental results of MMB fatigue tests

The simulated results in the normalized Paris Plot are in good agreement with the experimental results. Once more, the Mixed-Mode test case is also intended for analyzing only the linear region of the Paris Plot. In addition, results of the analysis on evolution of crack extension with number of loading cycles are illustrated in Figure 6.20.

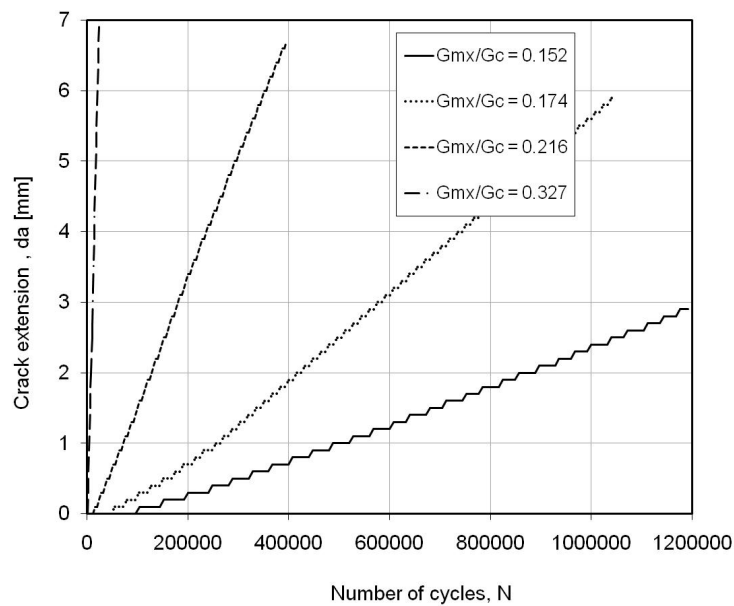


Figure 6.20 : Evolution of crack extension for different  $G_{max}/G_{TOT-0.5C}$  ratios in MMB fatigue test

From above figure, it can be concluded that, crack growth take place in linear manner with increasing number of loading cycles for the MMB test. Similar to 3ENF test, computation of crack growth rate becomes straight forward. The values of the constants;  $B$  and  $m$  of the Paris Law are found using the same method as explained earlier and the plot is given in Figure 6.21. The results obtained are also given below.

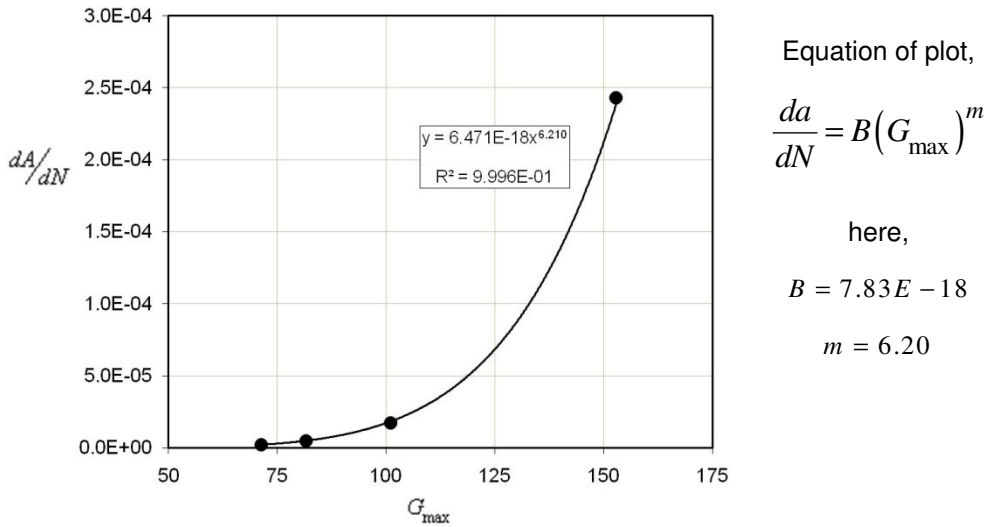


Figure 6.21 : Evolution of crack growth rate with  $G_{max}$  in MMB fatigue test

### 6.3.6 Mixed-Mode Fatigue Failure Prediction

The results of all numerical simulations performed for each mode of failure are found to be in good agreement with corresponding experimental data. Therefore, it proves the effectiveness of the proposed fatigue damage model in predicting delamination due to fatigue. Now, next task would be to propose a methodology to determine the parameters for the fatigue model for any given mode-ratio.

In order to achieve this goal one needs to develop a relationship between parameters  $B$  and  $m$  of the Paris law with mode-ratio,  $\phi$ . The idea here would be to use the known values of the parameters to interpolate the unknown values for a given mode-ratio. Many researchers have worked on this problem for decades and have proposed both monotonic and non-monotonic relationships with varying degree of accuracy with respect to their particular case studies. The mode-ratio is intimately related to type of the material and therefore the parameter relationship with mode-ratio differs significantly from GFRP to CFRP materials. Note that, the material HTA/6376C under consideration in this case study is a CFRP composite. Figure 6.22 and Figure 6.23 given below illustrate a compendium of the both monotonic and non-monotonic relations developed for CFRP materials [Blanco N. et al, 2004].

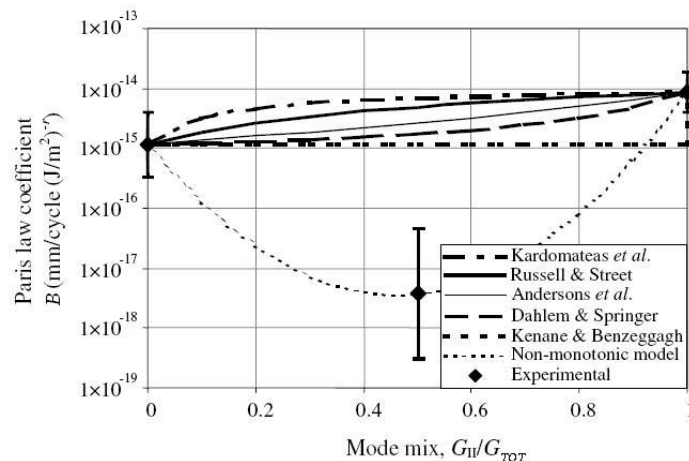


Figure 6.22 : Monotonic and non-monotonic relations for Paris law coefficient

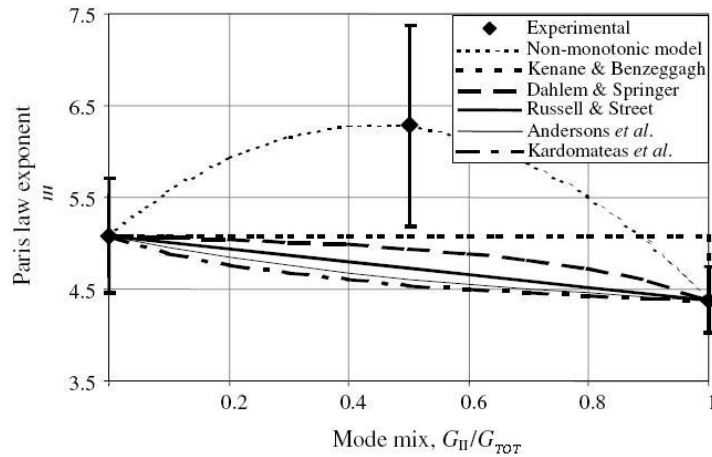


Figure 6.23 : Monotonic and non-monotonic relations for Paris law exponent

From the two figures, it is evident that the non-monotonic relation provides a better approximation to predict the evolution of parameters compared to the monotonic approaches. Therefore, present study focuses on application of non-monotonic relations to obtain new values for parameters of Paris law.

Recently in literature, a non-monotonic equation was reported [Tumino D. and Cappello F., 2007] to relate fatigue damage parameters and mode-ratio. To establish this non-monotonic relation, the suggested procedure requires at least three fatigue tests; fatigue in pure Mode-I, in pure Mode-II and in Mixed-Mode with a fixed value of  $\phi$  (i.e.,  $\phi_{mix}$ ). The fatigue tests shall provide values of  $B_I$ ,  $B_{II}$ ,  $B_{mix}$ ,  $m_I$ ,  $m_{II}$  and  $m_{mix}$  for the calibrated fatigue parameters  $C_I$ ,  $C_{II}$ ,  $C_{mix}$ ,  $\beta_I$ ,  $\beta_{II}$  and  $\beta_{mix}$ . Thereafter, the following equations will be used to compute the new values of  $B$  and  $m$  for any given mode-ratio. Note that, here fatigue damage parameter,  $\lambda$  will be taken as a fixed constant for all three failure modes to reduce the complexity of the formulation.

$$\beta = A_{1\beta}\phi^2 + A_{2\beta}\phi + A_{3\beta} \quad (6.04)$$

$$\ln(C) = A_{1C}\phi^2 + A_{2C}\phi + A_{3C} \quad (6.05)$$

Where,

$$A_{1\beta} = \frac{\beta_{II} - \beta_{mix} + (\beta_{II} - \beta_I)\phi_{mix}}{\phi_{mix} - \phi_{mix}^2} \quad (6.06)$$

$$A_{2\beta} = \frac{\beta_{mix} - \beta_I - A_{1\beta}\phi_{mix}^2}{\phi_{mix}} \quad (6.07)$$

$$A_{3\beta} = \beta_I \quad (6.08)$$

$$A_{1C} = \frac{\ln(C_I) - \ln(C_{mix}) + (\ln(C_{II}) - \ln(C_I))\phi_{mix}}{\phi_{mix} - \phi_{mix}^2} \quad (6.09)$$

$$A_{2C} = \frac{\ln(C_{mix}) - \ln(C_I) - A_{1C}\phi_{mix}}{\phi_{mix}} \quad (6.10)$$

$$A_{3C} = \ln(C_I) \quad (6.11)$$

Finally, the new values of  $C$  and  $\beta$  for a given mode-ratio will be used to find respective  $B$  and  $m$  values of the Paris law. In this study the same approach will be used to evaluate the performance of the proposed fatigue damage model for different mode-ratios where experimental data is no longer available. Table 6.06 given below provides a summary of the data and results of the study.

Fracture Mode (Test)	$\phi$	$C$	$\beta$	$\lambda$	$B$	$m$
MODE-I (DCB)	0	$2.0E-2$	3.5	1	$3.91E-15$	4.84
Mixed-Mode (MMB)	0.5	$1.2E-1$	3.35	1	$7.83E-18$	6.20
MODE-II (3ENF)	1	$7.0E-3$	2.2	1	$1.36E-14$	4.23

Table 6.06 : List of values of  $C$ ,  $\beta$ ,  $\lambda$ ,  $B$  and  $m$  for mode-ratios 0, 0.5 and 1

Now, evolutions of the fatigue model parameters,  $C$  and  $\beta$ ; with the mode-ratio can be illustrated as given in Figure 6.24 and 6.25 respectively.

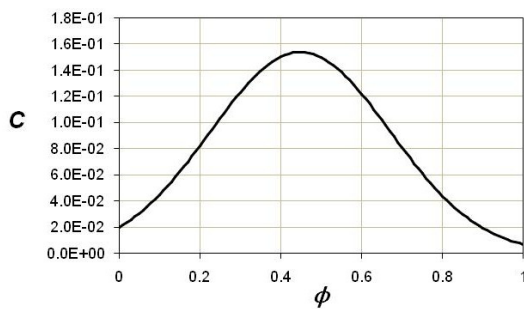


Figure 6.24 : Evolution of  $C$  vs.  $\phi$

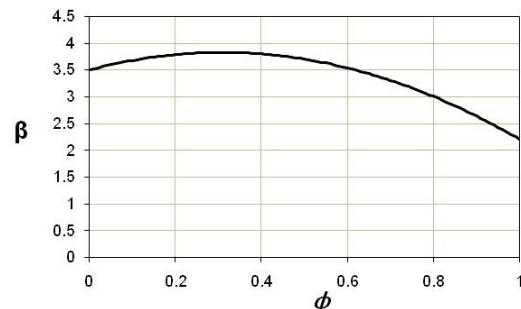


Figure 6.25 : Evolution of  $\beta$  vs.  $\phi$

Mode-ratios; 0.25 and 0.75 were the selected for the investigation. Figure 6.26 given below illustrate the normalized Paris plots developed using the predicted  $C$ ,  $\beta$  and  $\lambda$  values.

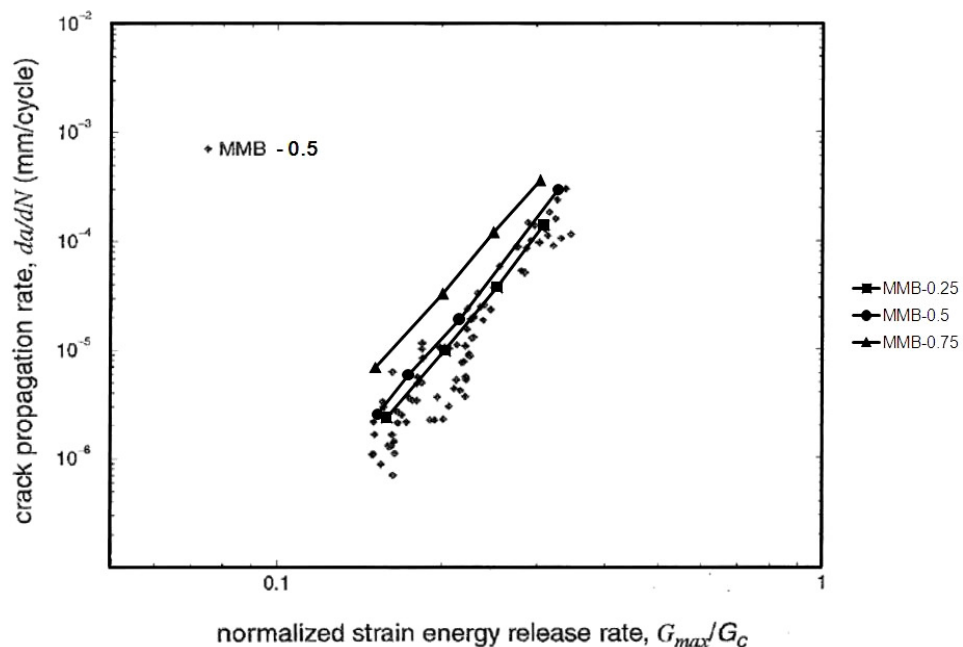


Figure 6.26 : Normalized Paris Plot for mode-ratios 0.25, 0.50 and 0.75

Note that, the new curves for MMB-0.25 and MMB-0.75 are also rectilinear and are roughly parallel to MMB-0.5 curve. This suggests that, the fatigue model is working in a robust manner. In addition, the results shown above give an insight to the change of the fatigue behaviour for different mode ratios. The curves tend to translate in the upward direction with increasing mode-ratio and interestingly, gap between each translation has also increased when the mode-ratio is doubled.

The coefficient and exponent of the Paris Law were found using the same method as explained earlier. Table 6.07 given below includes the with the predicted  $C$ ,  $\beta$  and  $\lambda$  values with their corresponding  $B$  and  $m$  values for each mode-ratio.

	$\phi$	$C$	$\beta$	$\lambda$	$B$	$m$
MMB-0.25	0.25	$1.03E-1$	3.81	1	$5.63E-17$	6.12
MMB-0.75	0.75	$6.11E-2$	3.16	1	$1.90E-17$	5.70

Table 6.07 : List of values of  $C$ ,  $\beta$ ,  $\lambda$ ,  $B$  and  $m$  for mode-ratios 0.25 and 0.75

In addition, Tumino and coworkers reported an empirical formula to evaluate the critical energy release rate for a given mode-ratio. The equation has the form given below.

$$G_{\phi C} = G_{IC} + (G_{IIC} - G_{IC})\phi^2 \tag{6.12}$$

Table 6.08 includes a comparison of the critical energy release rates calculated with the formula and by numerical means with simulations.

$\phi$	Analytic	Numeric	Difference %
0.25	0.306	0.343	10.6
0.50	0.446	0.468	4.8
0.75	0.677	0.707	4.2

Table 6.08 : Comparison of fracture toughness calculated by analytical and numerical methods

Evolutions of  $B$  and  $m$  of with the mode-ratio are illustrated in Figure 6.27 and 6.28 respectively.

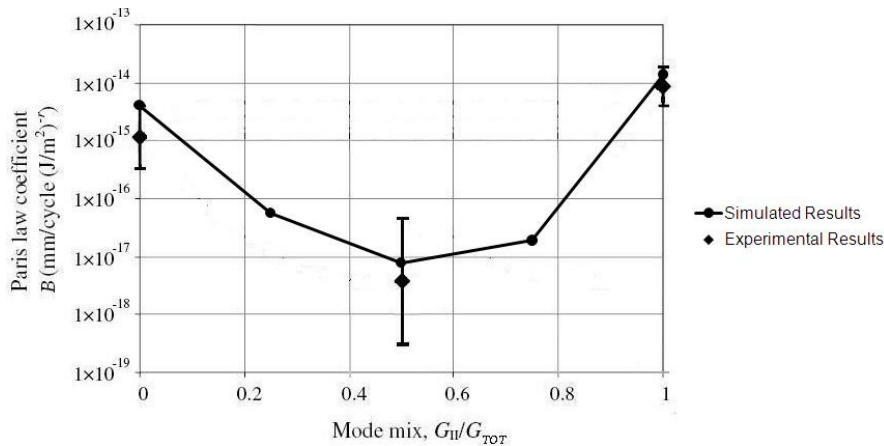


Figure 6.27 : Comparison of experimental / simulated results of  $B$  for different mode-ratios

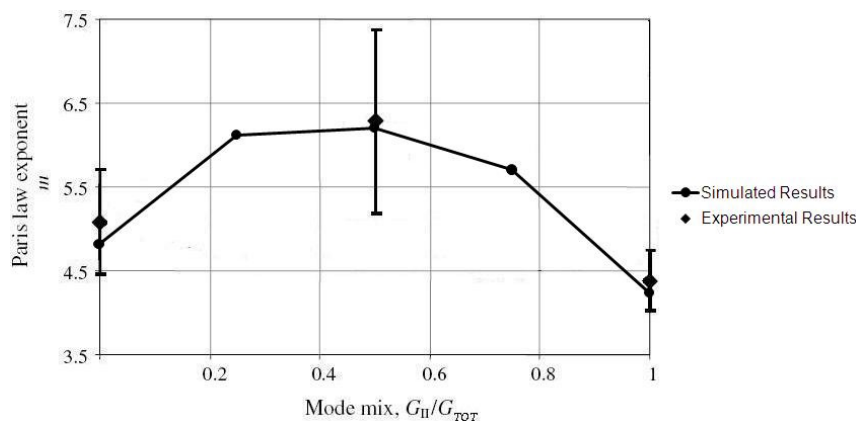


Figure 6.28 : Comparison of experimental / simulated results of  $m$  for different mode-ratios

As illustrated by the two figures;  $B$  and  $m$  values corresponding to mode-ratios 0.25 and 0.75 are in good agreement with the experimental data reported by Blanco N. et al, 2004. Note that,  $C$ ,  $\beta$  and  $\lambda$  values for the fatigue damage model were derived using a non-monotonic relation and therefore the results expected should also be non-monotonic. The key idea behind the investigation was to verify the effectiveness of the methodology. Note that, after calibrating the fatigue model (with three fatigue tests: pure Mode-I, pure Mode-II and a Mixed-Mode) it can be used to evaluate fatigue characteristics for any given mode-ratio. In practice, most composite structure experience Mixed-Mode failure loads more often than not. Moreover, there are situations where amplitude of the Mixed-Mode load would also vary with time. Therefore, the fatigue damage model needs to be versatile to handle such loading conditions. Importantly, the proposed model could easily be adapted for a given situation without many complications and has proven to be a robust tool.

## 6.4 Effects of Fatigue Model Parameters

Since affect of parameters,  $k_1^0$ ,  $k_2^0$  and  $\alpha$  has already been already discussed under the static part of the study, the following sections are devoted to understanding the affect of parameters;  $\Delta N$ ,  $C$  and  $\beta$  of fatigue model.

### 6.4.1 Influence of Parameter - $\Delta N$

To justify the fact that increment of number of cycles (i.e., jump cycles) has no effect on the final result, an investigation was carried out for different  $\Delta N$  values for each failure mode. Some selected results of the investigation are given hereunder. Figure 6.29 and Figure 6.30 illustrate evolutions of delamination length for 3ENF and MMB-0.75 tests with  $G_{\max} = 0.2G_C$ .

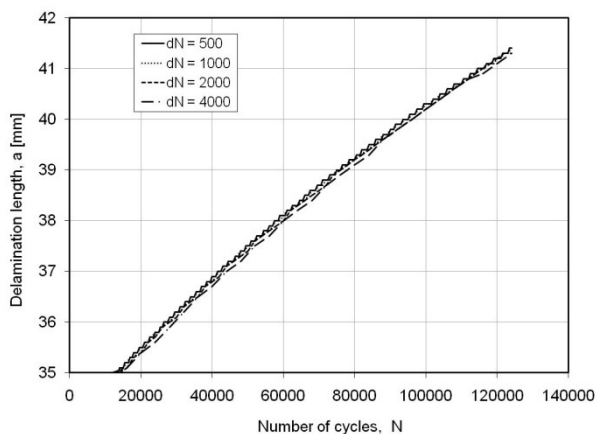


Figure 6.29 : Evolution of delamination length for different jump cycles in 3ENF fatigue test

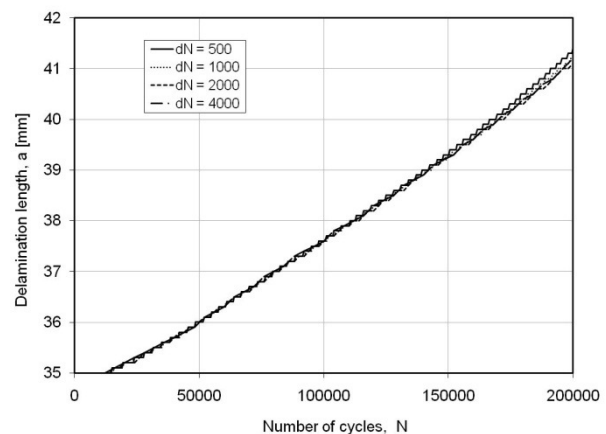


Figure 6.30 : Evolution of delamination length for different jump cycles in MMB fatigue test

By analyzing the results, it can be concluded that, evolution of the delamination length is independent of the selection of the jump cycles. Importantly, as  $\Delta N$  decreases computation time increases significantly. For higher values of  $\Delta N$  the delamination rate grow rapidly and proper analysis of the evolution of energy release rate and other parameters would be difficult. Therefore, a compromise should be made between the accuracy and computational cost. Here, the simulation results were obtained with element size equal to 0.1. Note that, element sizes lesser or equal to 0.2 produced converging results.

### 6.4.2 Influence of Parameters - $C$ & $\beta$

As for the fatigue model,  $C$  and  $\beta$  are the key parameters that influence the numerical results. The model was calibrated accordingly by adjusting the values of the two parameters with the aid of experimental results. For simplicity,  $\lambda$  was taken as a constant throughout the study. Given hereunder are the results of the investigation carried out by varying one parameter while keeping the other one constant. Figure 6.31 and Figure 6.32 given below illustrate the influence of parameter,  $C$  for  $\beta=2.2$  and parameter,  $\beta$  for  $C=7.0E-3$  respectively considering the 3ENF test.

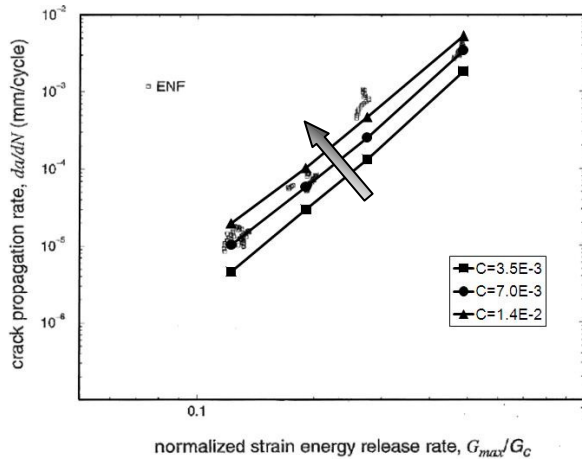


Figure 6.31 : Influence of  $C$  in 3ENF fatigue test

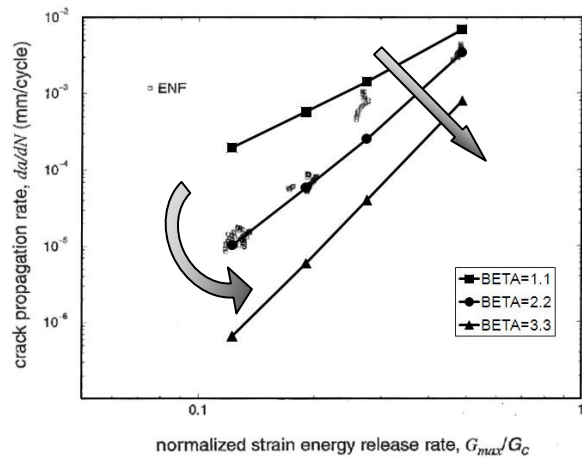


Figure 6.32 : Influence of  $\beta$  in 3ENF fatigue test

Here in Figure 6.30, as  $C$  increase the Paris plot curves tend to translate in the upward direction without exhibiting any rotation. It is also interesting to observe that when the value of  $C$  is doubled the translation distance of the curve is roughly similar. Next, in Figure 6.31 much more complicated behaviour was observed. In this case Paris plot curves tend to not only translate but also rotate at the same time for increasing value of  $\beta$ . On the contrary to parameter,  $C$  both translation and rotation takes place in the downward direction.

Prior to the beginning of the simulation campaign the user needs to first appreciate the probable values for each parameter through a trial-and-error analysis (FE inverse-optimization also possible). When the approximate values are known, the aforementioned behavioural pattern of the parameter can be taken in to account for fine tuning and arrive at a good approximation.

## **PART – IV**

### **Closure**

Section include one chapter, Chapter 7.  
Contains remarks regarding the work carried out on static and fatigue delamination modelling and simulations.



## Chapter 7

### General Conclusions

Failure simulation of a laminate structure, taking into account all the progressive degradation phenomena, leads to a time dependant three-dimensional non-linear problem. Especially in the past three decades many researchers have been working towards in solving this problem with different types of mathematical tools. In this respect, both Fracture and Damage Mechanics have grown from strength to strength in recent times in successfully predicting delamination phenomenon in laminated composites. In present study, efforts have been made to further enhance an existing damage model to predict delamination of unidirectional composite laminates at both static and fatigue loading conditions. The damage model is formulated within the framework of Damage Mechanics, but is also essentially linked to Fracture Mechanics, as the damage model parameters (both of static and fatigue models) are determined using Fracture mechanics test results. Discussed hereunder are some of the key findings of the present study. The conclusions are based on the results obtained from the FE modelling(s) presented earlier. Note that, all the simulations in the present work have been performed in CAST3M finite element software developed by CEA/DMT. The non-linear material models were introduced in to the finite element frame work of CAST3M using the user-defined subroutine, known as UMAT (Source - ABAQUS).

#### 7.1 Delamination under Quasi-Static Loading

The work performed under static study consists two major parts. At first, the existing local damage model was thoroughly studied and effort was put into FE implementation of the model in all dimensional spaces. A complete description on identification of model parameters and their influence on model's behaviour were presented. Secondly, a nonlocal damage model was proposed. Here, the proposed model is based on existing theories on nonlocal modelling for homogeneous monolithic materials. Inherent problem in nonlocal modelling is their high computational cost. Therefore, the proposed FE implementation procedure was further optimized to reduce the computational time significantly.

Under the study performed on the local model, much attention was given on identification of model parameters. Many parameters were basically identified by Fracture Mechanics test results. However, some parameters such as exponent of material function ( $n$ ) and interface rigidities were based on previous work published on similar studies. Therefore, a separate analysis was performed on appreciating their influence on damage model's behaviour. Here, it was found that parameter  $n$  has a notable influence on the maximum stress attained. With increasing  $n$ , both stiffness and maximum stress value seem to increase further. The interface rigidities are determined by formulas derived using the instability criteria. Finally, the initial rigidities are expressed in terms of critical stress for the interface. Generally, in most publications attention devoted to correct identification of critical stress is given less importance. However, investigation results show that simulation results are considerably sensitive to this parameter.

Simulation results corresponding to local damage model for all dimensional spaces were found to be in good agreement with the experimental results. For both Mode-I (DCB) and Mode-II (3ENF) simulations, it was necessary to consider the contact condition between laminate arms. Note that, interface elements (i.e., in CAST3M) are not modelled to bear compressive forces, hence upon compression or even flexure the two meshes tend to overlap each other and give erroneous numerical results. Results on convergence analysis also helped to identify the optimum parameters for the simulations. Through, 3D modelling it was possible to identify the actual shape of the crack front. In addition, the stress field over the 2D interface provided an insight on the size of the process zone associated to each failure mode. For DCB test it was observed that crack initiation in the center

of the specimen happens much earlier than close to outer edges. Therefore, crack front is no longer a straight line but a curved one. In case of 3ENF test, the crack growth at both center and edges happen relatively at the same time. However, it was noted that initiation of the crack still happens at the center of the specimen. As it was evident from the stress field plots, the process zone ahead of crack front in 3ENF test is much larger than the process zone in DCB test. This phenomenon can be explained by the slow decay of the stress field associated to Mode-II failure mode. Thus, it partly explains the reason for the uniform crack extension observed over the width of 3ENF test specimens.

Under the study performed on nonlocal model, a detailed analysis was made on understanding its behaviour. Here, the implementation of the proposed model was done using the PERSO1 personal procedure in CAST3M. In the present study, the damage energy variable was used to regularize the evolution of all the other variables (i.e., damage variables, stress, strain, etc). In the original formulation of the nonlocal model, averaging of the variables is performed at each gauss point of all the elements. As the number of elements increase, the number of variables that is to be averaged would also increase. In addition, averaging of variables at each point is normally performed by considering all the points in the interface. When the number points increase, the number of points that is to be considered for the averaging process will automatically increase. In fact, this is the main reason which is responsible for the notable increase of the computational time. However, in the analysis it was noted that only some points in the neighbourhood of the interested point are responsible for making a significant contribution to the averaging process. Similarly, few other redundant computations were also identified. Finally, by considering all these facts several modifications were introduced to the original formulation to reduce computational cost. Interestingly, the ratio of the computational time between the original and the modified model was found to be in the range between 4 and 6. Here, results obtained from both models were essentially the same. Therefore, this proves the effectiveness of the simulation methodology introduced with the proposed nonlocal model.

Note that, simulated results are in good agreement with the experimental results. Interestingly, the convergence of the simulated results needed to be checked with both element size and increment size. Here, converged results of both local and nonlocal models are close to each other. However, in case of nonlocal model, an increase of the maximum force can be seen. This is mainly, due to the added viscosity introduced by the nonlocal model. By performing averaging we ultimately delay the evolution of the variables. As a consequence we see an increase of the maximum force value. This phenomenon becomes more and more severe as we continue to increase the magnitude of the internal length scale. This is obviously true because by increasing the internal length scale we are ultimately increasing the size of the averaging zone. Importantly, the internal length scale should not be too small compared to the element size. If it goes below the critical value, then the nonlocal model will no longer be capable of regularizing the evolutions of the variables. Identification of the internal length scale is still an open question and needs rigorous analysis with more experimental data. Some effort was also put into predict size effect phenomenon using the proposed nonlocal model. As expected, the model was able to capture the decrease of the laminate strength with increasing size.

## 7.2 Delamination under Fatigue Loading

The work performed under fatigue study includes FE implementation of a newly proposed fatigue damage law. At first, an overview was made on existing fatigue theories for isotropic materials (i.e., metals). Fatigue behaviour of metals is usually characterized by the Paris law based on stress intensity factor. However, evaluation of stress intensity factors is problematic for composite materials. Therefore, the Paris law based on strain energy release rate was considered in the present study.

Here, the existing local model was used as a platform to build the new fatigue model. The derivation and FE implementation of this model was explained in details in Chapter 7. The new fatigue model is capable of carrying out FE simulations on composite materials undergoing different types of loading conditions. Range of loading conditions which it can handle include pure Mode-I, pure Mode-II and any given Mixed-Mode loading condition. In literature, there are several models that extend cohesive laws for monotonic loading into forms suitable for cyclic loading. But, most of these extended cohesive models determine the accumulated damage on a cycle-by-cycle basis and as the number of cycles grows the computation becomes much more intractable. Therefore, for high-cycle fatigue, the damage

evolution that results from cyclic loads needs to be formulated as a function of the number of cycles and displacement jumps. The newly proposed model is basically based on this concept.

Here in the fatigue damage law, the total damage in the interface is considered to be sum of the damage caused by quasi-static load and the damage that results from the cyclic loads. The existing damage model was systematically modified to incorporate these damage effects coming from both static and cyclic parts of the loading. Similarly to the case static damage model, the fatigue model parameters were also needed to be identified using Fracture Mechanics test results. The four main parameters of the fatigue model are  $C$ ,  $\lambda$ ,  $\beta$  and  $\Delta N$ . As it was shown later, the choice of the number of increment cycles does not affect the final result of the computation. Therefore, the user is free to select a value for  $\Delta N$ . But it was noted that, for large  $\Delta N$  the delamination process will be rapid and proper identification of evolution of certain parameters (i.e., energy release rate) may become difficult. However, if  $\Delta N$  is too small, the computational time would increase significantly. An optimum value for  $\Delta N$  was selected depending on the problem in hand (i.e., type of loading condition, maximum value of the cyclic load envelope, etc). In the present study, identification of parameter values for  $C$ ,  $\lambda$  and  $\beta$  was achieved through a trial-and-error approach. However, for future studies parameter identification through FE inverse-optimization approach is recommended.

The simulated results were found to be in good agreement with the experimental results. Importantly the fatigue model was able to reproduce the linear part of the Paris plot with good accuracy. Note that, in the present study more focus was given to identifying the behaviour of the composite material in cyclic load ranges corresponding to the linear region of the Paris plot. For simplicity, parameter  $\lambda$  was kept constant while parameters  $C$  and  $\beta$  were used as variables. To help the identification process, a separate analysis was performed on understanding the influence of parameters  $C$  and  $\beta$ . According to investigation results, it was seen that as  $C$  increase the Paris plot curves tend to translate in the upward direction without exhibiting any rotation. However, in case of increasing value of  $\beta$  the Paris plot curves tend to not only translate but also rotate at the same time. On the contrary to parameter  $C$  both translation and rotation takes place in the downward direction for parameter  $\beta$ . Finally, identified behavioural pattern of these parameters were taken in to account to calibrate the fatigue damage model using the available experimental data (i.e., data corresponding to pure Mode-I and Mode-II).

Evolution of the energy release rate with increasing crack length was also determined for each pure mode case (i.e., Mode-I and Mode-II). This was achieved by performing numerical integration over the local stress and strain variables at each gauss point of the interface elements. All evolutions corresponding to medium to small cyclic load levels were in good agreement with the experimental results. However, for high cyclic load levels the simulated results show small deviation from the experimental results. The cause here is believed to be associated to the large deformations of the laminate arms at high cyclic load levels.

In the final part of the study, fatigue model was tested for different Mixed-Mode ratios. The idea was to develop a methodology to identify model parameters without the aid of any experimental data. Here, a non-monotonic relation between each parameter was established using results of three different Fracture Mechanics tests. The parameters identified by this approach were then used to develop the Paris plot for for different mode-ratios. Importantly, the simulated results were in good agreement with the reported experimental results.

In conclusion, the fatigue model seems to be operating effectively and proves to be a versatile model in determining delamination phenomena for different loading conditions.

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